

1 **Biodegradable active food packaging structures based on hybrid cross-linked**  
2 **electrospun polyvinyl alcohol fibers containing essential oils and their application in**  
3 **the preservation of chicken breast fillets.**

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26 **ABSTRACT**

27 Active food packaging materials produced through electrospinning of polyvinyl alcohol  
28 (PVOH) containing essential oils (EOs) from two broadly used spices (*Laurus nobilis* –  
29 LEO- and *Rosmarinus officinalis* -REO-) were developed and applied to chicken breast  
30 fillets with the aim of prolonging their shelf-life. Citric acid (CA) was successfully  
31 incorporated as a natural cross-linker and the developed electrospun structures were heat-  
32 treated to promote both the crosslinking and the crystallization of the PVOH matrix.  
33 Initially, the morphology, water solubility, physicochemical and thermal properties of the  
34 developed structures were evaluated. Then, the antioxidant and antibacterial efficiency of  
35 PVOH-EOs hybrid structures were evaluated when directly applied onto an inoculated  
36 side of chicken breast fillets. These annealed active food packaging structures containing  
37 EO and CA exhibited improved water resistant and thermal stability with respect to their  
38 non-crosslinked counterparts. Although part of the EOs content was degraded after the  
39 annealing process, the remaining amount in the PVOH samples inhibited the lipid  
40 oxidation process up to 68% and displayed enhanced antimicrobial effectiveness when  
41 applied onto chicken breast fillets, having a beneficial effect on both the pH and color  
42 parameters during storage.

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47 **Keywords**

48 Active packaging; electrospinning; biodegradable materials; food coatings

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## 52 **1. INTRODUCTION**

53 Biodegradable packaging materials have recently gained attention in the food industry  
54 due to consumer's awareness of associated environmental benefits. Among them,  
55 polyvinyl-alcohol (PVOH) is a water soluble petroleum-based polymer with very good  
56 optical properties, non-toxicity, biocompatibility, biodegradability and good film-  
57 forming ability to be used as food packaging (Aslam, Kalyar, & Raza, 2018; Shi et al.,  
58 2008; Tian, Yan, Rajulu, Xiang, & Luo, 2017). These intrinsic characteristics make  
59 PVOH an interesting material for the development of active packages or coatings of great  
60 interest in the food area. The development of active materials aiming at maintaining or  
61 enhancing the quality and safety of packaged food through the incorporation of  
62 antimicrobial and/or antioxidant natural compounds is, in fact, an active research area  
63 (Chen et al., 2018; Fang, Zhao, Warner, & Johnson, 2017; Kwon, Chang, & Han, 2017;  
64 Neo et al., 2013). However, the most widely used processing methods for the  
65 development of packaging materials require high temperatures which could compromise  
66 the active properties of many of antimicrobials and antioxidant compounds, as most of  
67 them are thermosensitive (i.e. essential oils) (Atarés & Chiralt, 2016; M. Ramos, Jiménez,  
68 Peltzer, & Garrigós, 2012).

69 To counteract this problem, electrospinning has been lately proposed as an alternative to  
70 develop active food packaging materials (Estevez-Areco, Guz, Candal, & Goyanes, 2018;  
71 Fabra, Lopez-Rubio, & Lagaron, 2016; Lin, Mao, Sun, Rajivgandhi, & Cui, 2019;  
72 Pinheiro Bruni et al., 2020). By means of this technique, encapsulation structures can be  
73 developed by applying electrostatic forces between a solution of the biopolymer and a  
74 grounded collector without the need of using high temperatures for drying the materials,  
75 as the solvent is evaporated during the flight of the solution towards the collector due to

76 the whipping of the biopolymer caused by the high voltage application (Anu Bhushani &  
77 Anandharamakrishnan, 2014; Zhang et al., 2019). Therefore, this technology provides  
78 several advantages in preserving the functionality of the active compound encapsulated  
79 within the electrospun fibers.

80 As active compounds, essential oils (EOs) have attracted extensive research, because  
81 apart from being a natural product, they have also demonstrated numerous biological  
82 activities like, for instance, antimicrobial, antioxidant or anti-inflammatory amongst  
83 others (Brahmi et al., 2016; Gómez-Estaca, López de Lacey, López-Caballero, Gómez-  
84 Guillén, & Montero, 2010; Yen, Hsieh, Hsieh, Chang, & Wang, 2015). Specifically,  
85 essential oils from bay (*Laurus nobilis*) and rosemary (*Rosmarinus officinalis*) have  
86 received much attention because of their antimicrobial and antioxidant properties  
87 (Göksen, Fabra, Ekiz, & López-Rubio, 2020; Ojeda-sana, Baren, Elechosa, Juárez, &  
88 Moreno, 2013; C. Ramos et al., 2012).

89 However, one of the drawbacks of PVOH as a matrix to include the essential oils is its  
90 high water solubility which limits its application in real foods with high water activities.  
91 To overcome this problem, several strategies have been carried out in the literature,  
92 including mixtures with other polymers (Çay, Miraftab, & Perrin Akçakoca Kumbasar,  
93 2014; Peresin et al., 2014; Yang, Li, & Nie, 2007) and the use of cross-linking agents (Shi  
94 et al., 2008) in order to improve the functionality and applicability of PVOH-based  
95 materials. Crosslinking can be achieved applying either physical (heat treatment, gamma  
96 irradiation, ultraviolet) or chemical agents (glutaraldehyde, formaldehyde) (Mansur,  
97 Sadahira, Souza, & Mansur, 2008; Miraftab, Saifullah, & Çay, 2015). However, most of  
98 these treatments have been reported to be harmful. For instance, migration of metal ions,  
99 used as cross-linkers, could exert cytotoxic effects (Lee & Mooney, 2001). Other  
100 crosslinking chemical agents such as glutaraldehyde are highly toxic and are prone to

101 leach out from the packaging material to the food, thus compromising food safety (Lin,  
102 Gu, & Cui, 2018). As an alternative, citric acid (CA), a food-grade compound (Suganthi  
103 et al., 2018), has been proposed in this work as a crosslinking agent to keep the integrity  
104 of the PVOH materials once in contact with high water activity food products (Stone,  
105 Gosavi, Athauda, & Ozer, 2013).

106 This proof-of-concept study proposes a new route for the development of novel active  
107 packaging structures of interest in food preservation. Therefore, based on the  
108 considerations outlined above and potential applications of active films, the objectives of  
109 the present work were to develop PVOH active films by means of electrospinning and to  
110 investigate the influence of the crosslinking treatment on the morphology, thermal and  
111 physicochemical properties of the developed structures. Finally, their application as a  
112 potential antimicrobial and antioxidant food packaging material has been evaluated in  
113 chicken breast fillets.

114

## 115 **2. MATERIALS AND METHODS**

### 116 **2.1. Materials**

117 *Laurus nobilis* essential oil (LEO) and *Rosmarinus officinalis* essential oil (REO) were  
118 obtained according to Göksen et al., (2020). Specifically, the composition of REO and  
119 LEO was previously determined (Göksen et al., 2020), and the main compound found in  
120 both essential oils was 1,8 cineole (55.80 % and 69.87%, respectively).

121 Polyvinyl alcohol (PVOH) was purchased from Plásticos Hidrosolubles S.L. (Valencia,  
122 Spain) and citric acid (CA) was obtained from Sigma-Aldrich (St. Louis, Mo., U.S.A).  
123 The strain of *Listeria monocytogenes* CECT 4032 (NCTC 11994) was obtained from the  
124 Spanish Type Culture Collection (CECT, Valencia, Spain). Oxford-Listeria-selective  
125 agar base was purchased from Thermo Scientific-Oxoid (Basingstoke, UK). Fresh

126 chicken breast slices were purchased from a local supermarket in Valencia (Spain) and  
127 immediately transported to the laboratory and kept under refrigerated conditions.

128

## 129 **2.2. Preparation of electrospinning solutions**

130 PVOH solutions were prepared by dissolving 14 g PVOH powder in 100 mL distilled  
131 water at 60 °C under magnetic stirring at 300 rpm. Aqueous solutions containing CA were  
132 similarly prepared and then, 5 g of CA were incorporated into the PVOH solution under  
133 stirring for another 2 h. Aqueous solutions containing LEO or REO essential oils (EOs)  
134 were prepared by adding 10 g of each EOs into the as-prepared PVOH aqueous solution  
135 and stirred for 2 h to obtain homogeneous solutions. Compositions and nomenclature of  
136 each formulation are compiled in Table 1.

137

138 **Table 1.** Composition and properties of electrospinning solutions.

Sample code	%PVA(w/v)	%Citric acid (w/w)	%LEO (w/w)	%REO (w/w)
PVA	14	0	0	0
PVAc	14	5	0	0
PVA/LEO	14	0	10	0
PVAcLEO	14	5	10	0
PVA/REO	14	0	0	10
PVAcREO	14	5	0	10

139 PVA: Polyvinyl alcohol; c: citric acid; LEO: *Laurus nobilis* essential oil; REO:  
140 *Rosmarinus officinalis* essential oil

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143 **2.3. Characterization of electrospinning solutions**

144 The Wilhemy plate method was used to measure the surface tension of the biopolymer  
145 solutions using the in EasyDyne K20 tensiometer (Krüss GmbH, Hamburg, Germany).  
146 The viscosity of the electrospinning solutions was measured using a rotational viscosity  
147 meter (Visco Basic Plus L) from Fungilab S.A. (San Feliu de Llobregat, Spain) using  
148 spindle no TL1 at 12 rpm. The electrical conductivity and pH of the solutions were  
149 determined using a conductivity meter (XSCon6) and pH meter (XS pH50) from Labbox  
150 (Barcelona, Spain). All measurements were done in triplicate at  $25 \pm 2$  °C.

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153 **2.4. Electrospinning process and post-treatment of the fibers**

154 The electrospinning solutions were loaded into 5 mL plastic syringes connected to an 18-  
155 gauge stainless steel nozzle with horizontal configuration. This process was carried out  
156 by an electrospinning apparatus including high-voltage (0–30 kV) power supply  
157 (Acopian, USA), a syringe pump (KD Scientific, USA) and a collector plate. The flow  
158 rate of the solutions, applied voltage and tip-to-collector distance were set to 0.15–0.20  
159 mL/h, 18 kV and 10 cm, respectively, based on preliminary optimization trials. The  
160 electrospinning process was carried out under ambient conditions (20 °C and 58% RH).  
161 After the electrospinning process, the samples were cured in an oven at 170 °C for 10 min  
162 to foster materials' crosslinking with/without CA. The temperature and time were fixed  
163 based on screening experiments.

164

165 **2.5. Scanning electron microscopy (SEM)**

166 The morphology of the nanofibers was analyzed using a scanning electron microscope  
167 (SEM) (Hitachi S-4800, Matsuda, Japan) at an accelerating voltage of 10 kV and working

168 distance of 8-16 mm. The samples were coated with gold-palladium sputtering under  
169 vacuum (Pinheiro Bruni et al., 2020). After analysis, the diameters of the fibers from SEM  
170 micrographs were measured using ImageJ software (Image J, NIH, Maryland, USA).

171

## 172 **2.6. Fourier transform infrared spectroscopy (FT-IR) analysis**

173 Electrospun fiber mats were characterized using FT-IR spectrometer (Bruker  
174 Rheinstetten, Germany). The scans were recorded in the spectral range from 650 to 4000  
175  $\text{cm}^{-1}$ . All spectra were collected with a resolution of  $4 \text{ cm}^{-1}$  by averaging 16 scans.

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177

## 178 **2.7. Thermal properties of electrospun fibres**

179 Thermal properties of the electrospun mats were characterized using differential scanning  
180 calorimetry (DSC) (Perkin Elmer, Inc., DSC 7) and thermogravimetric analysis (TGA)  
181 (TA Instruments model Q500 TGA). For TGA, the samples (approximately 10 mg) were  
182 heated up from  $25 \text{ }^\circ\text{C}$  to  $600 \text{ }^\circ\text{C}$  at a heating rate of  $10 \text{ }^\circ\text{C min}^{-1}$  under dynamic nitrogen  
183 atmosphere. TGA curves express the weight of the sample as a function of temperature.

184 For DSC analysis, samples (between 2-5 mg) were heated of  $10 \text{ }^\circ\text{C min}^{-1}$  from 20 to 290  
185  $^\circ\text{C}$  under nitrogen gas and using an empty pan as a reference. Indium was used for  
186 calibration and the thermograms from an empty pan were used for correcting the slope of  
187 the thermal scans from the samples. All tests were carried out, at least, in duplicate.

188 The degree of PVOH crystallinity was calculated from the ratio:

$$189 \chi_c(\%) = (\Delta H_m / \Delta H_m^0) \times 100 \quad (\text{Equation 1})$$

190 where  $\Delta H_m^0 = 138.6 \text{ J g}^{-1}$  is the melting enthalpy for a perfect 100% PVOH crystal  
191 (Jiang, Qiao, & Sun, 2006).

192



193 **2.8. Challenge tests**

194 Fresh chicken breast fillets (0.80 g carbohydrate, 22.18 g protein, 1.54 g fat content per  
195 100 g) were purchased from a local market in Valencia (Spain) and transferred to the  
196 laboratory within 1 h and kept under refrigerated conditions. These fillets were aseptically  
197 cut into pieces of 10 g and covered with the electrospun structures. The prepared samples  
198 were then placed in sterile petri dishes and petri dishes were sealed with parafilm and  
199 stored at 4 °C up to 7 days. The physicochemical and microbiological analyses were  
200 performed, in triplicate, at 0, 1, 4 and 7 days of storage. The antioxidant analysis was  
201 carried out, in triplicate, at the beginning and at the end of the storage time. Four different  
202 groups were prepared: uncovered fillets (control) and fillets covered (directly contacting  
203 the pieces of chicken) with the cross-linked fibers (PVOHc, PVOHcLEO, PVOHcREO).  
204 The pH and color parameters of the samples were measured at each storage time. Briefly,  
205 chicken breast samples (10.0 g) were homogenized with 90 mL of distilled water for 1  
206 min using a homogenizer (MICCRA GmbH, Heitersheim, Germany). Then, the  
207 homogenate was used for determination of pH value (Gonzales-Fandos, Herrera, & Maya,  
208 2009). Color parameters (CIE L\*, a\* and b\*) of chicken breast samples were determined  
209 using Chroma Meter-CR 400 (Konica Minolta Japan).

210 The lipid oxidation was evaluated by means of the thiobarbituric acid reactive substance  
211 (TBARS). The TBARS value was evaluated with the 2-thiobarbituric acid (TBA)  
212 distillation method described by Fontes-Candia, Erboz, Martínez-Abad, López-Rubio, &  
213 Martínez-Sanz, (2019) with a slight modification. Briefly, 10.0 g of chicken breast meat  
214 were homogenized with 30 mL of water. The homogenate was transferred to a distillation  
215 flask (500 mL) with 65 mL of water and then the pH was adjusted to 1.5 with using HCl  
216 (4N). Ethanolic propyl gallate (10%, 1 mL), EDTA (10%, 1 mL), and a drop of anti-  
217 foaming agent were added to mixture. The flask was connected to a Soxhlet apparatus

218 and the mixture was boiled until 50 mL of distillate were gathered. After distillation, 5  
219 mL of TBA reagent (0.02M TBA in 90% acetic acid) were added to 5 mL of the distillate  
220 and placed in a boiling water bath for 40 min of reaction. As a control, 5 mL of distilled  
221 water were used instead of the distillate. After cooling at room temperature, the  
222 absorbance was measured at 535 nm in a spectrophotometer (Santa Clara, CA, USA).  
223 TBARS value was calculated by multiplying the absorbance readings by a factor of 7.8  
224 and expressed as mg malondialdehyde (MDA)/kg meat. The inhibition of lipid oxidation  
225 was determined after 7 days as follows:

$$226 \quad \text{Inhibition (\%)} = \frac{(TBARS_{\text{blank}} - TBARS_{\text{electrospun}})}{(TBARS_{\text{blank}} - TBARS_{\text{day 0}})} \times 100 \quad (\text{Equation 2})$$

227

228 In this equation,  $TBARS_{\text{day 0}}$  corresponds to the TBARS in the uncovered fresh chicken  
229 breast meat at day 0 and  $TBARS_{\text{blank}}$  and  $TBARS_{\text{day 7}}$  refer to the TBARS in the control  
230 sample (uncovered fresh chicken breast) and in the different meat samples covered with  
231 the electrospun materials after 7 days of refrigerated storage, respectively.

232 The antimicrobial assays were carried out as follows: chicken breast fillets (10 g) were  
233 sterilized by immersion in ethanol 70% for 5 min. Before packing, the samples were  
234 soaked in the bacteria suspension of *L. monocytogenes* (approximately  $10^8$  CFU/mL) and  
235 left under a laminar flow safety cabinet for 15 min for attaching inoculums onto chicken  
236 breast meat surface. Later on, the samples were packaged individually with the nanofibers  
237 (5x5 cm) previously sterilized with UV light for 30 min. Packaged and control samples  
238 (unpackaged) were placed into sterile petri dishes and sealed with parafilm and then  
239 stored at 4 °C. *L. monocytogenes* presence in the different samples was evaluated by plate  
240 counting at different time intervals (0, 1, 4 and 7 days) during cold storage.

241

## 242 **2.9. Statistical Analysis**

243 All results were presented as the average  $\pm$  standard deviation. Statistical analysis was  
244 performed by analysis of variance (ANOVA) with SPSS software (version 17.0; IBM  
245 Corp., Armonk, NY) using Tukey's test.

246

### 247 **3. RESULTS AND DISCUSSION**

#### 248 **3.1. Development and characterization of active fibers containing LEO or REO.**

249 Initially, the physicochemical properties of the electrospinning solution were evaluated  
250 and correlated with the morphology of developed fibers. Different sizes and morphologies  
251 can be obtained through electrospinning processing of biopolymer solutions depending  
252 on solution properties (surface tension, viscosity, electrical conductivity) and process  
253 parameters. In this work, CA was added as a PVOH crosslinking agent (in combination  
254 with an annealing process required for effective crosslinking) in order to avoid the  
255 disintegration of electrospun fibers once they would be in direct contact with food, since  
256 most of them present high water activity.

257 PVOH-based solutions with and without CA or EOs were initially characterized and the  
258 results are compiled in Table 1S of the Supplementary Material. pH values of the PVOH  
259 solutions were around 5.3 at room temperature and they significantly decreased ( $p < 0.05$ )  
260 with the incorporation of CA due to its acidic nature. The addition of CA resulted in a  
261 slight increase in the viscosity values of the biopolymer solution, in agreement to that  
262 previously reported by Esparza, Ullah, Boluk, & Wu, (2017), although it did not  
263 significantly affect the average diameters of the fibers obtained. In general, the  
264 incorporation of EOs made the fluid solutions more viscous (greater apparent viscosity  
265 values) than the neat PVOH and PVOHc solutions (see Table 1S) which, surprisingly,  
266 resulted in thinner fibers (see Table 2S in the Supplementary Material) thus suggesting

267 that, in this case, the diameter of the fibers was mainly governed by the electrical  
268 conductivity of the electrospinning solutions.

269 Concerning the conductivity values, the addition of CA solution provoked a sharp  
270 increase ( $p < 0.05$ ) in electrical conductivity of the PVOH solution ( $1081 \mu\text{S}/\text{cm}$ ), probably  
271 ascribed to the ionic nature of this carboxylic acid in solution. A similar trend was  
272 observed in PVOH-EOs solutions containing CA, favoring the formation of slightly  
273 thinner fibers during electrospinning as the greater conductivity facilitates jet stretching,  
274 as inferred from Table 2S and Figure 1.

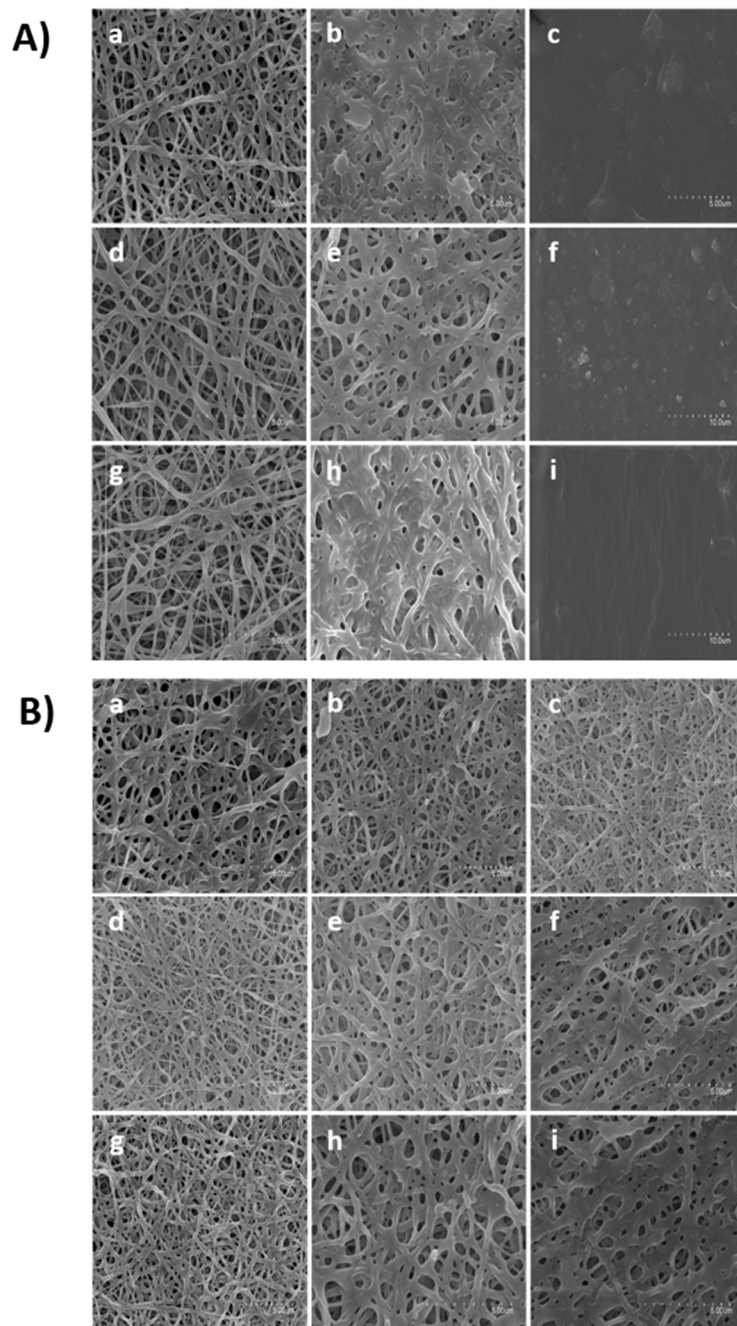
275 Regarding the surface tension values, PVOH-aqueous solutions showed surface tension  
276 values significantly lower than those of the solvent ( $72 \text{ mN}/\text{m}$ ). Likewise, the  
277 incorporated EOs behaved as surfactants, further decreasing the surface tension values of  
278 the solutions. A slight increase in surface tension was observed upon CA addition when  
279 compared with their counterparts prepared without the natural crosslinking agent. As  
280 deduced from the solution properties, surface tension did not play a major role on the  
281 morphology of the fibers which were mainly governed by the increase in the electrical  
282 conductivity of the CA-containing solutions, favoring the formation of thinner fibers.

283 The SEM images of PVOH-based fibers (with and without CA and EOs) are displayed in  
284 Figure 1. Table 2S of the Supplementary Material gathers the average diameter of the  
285 electrospun fibers before and after the annealing process used to promote the PVOH-CA  
286 crosslinking. The size measurements from the SEM micrographs revealed that the  
287 average diameter of fibers decreased with the incorporation of both essential oils  
288 (LEO/REO) and CA, mainly ascribed to the increase in electrical conductivity.  
289 Interestingly, a significant increase ( $p < 0.05$ ) in the average diameter was clearly observed  
290 in PVOH and PVOH/EOs films after the annealing process. In contrast, this increase was  
291 significantly lower if the electrospun coatings had CA, suggesting that interactions

292 between the PVOH and CA preserved the morphology and porous structure of the  
293 coating, and thus the integrity of the fibers, to a greater extent. The increase in the average  
294 diameter of the electrospun fibers in annealed films prepared without CA can be attributed  
295 to a partial melting of the PVOH matrix and subsequent aggregation of the PVOH-based  
296 fibers during the heat treatment, as it is clearly observed in the images B, E and H of Fig.  
297 1A.

298 As most food products present high water activity, the integrity of the PVOH-based  
299 electrospun fibers upon water immersion was also evaluated and their morphology after  
300 immersion and subsequent drying was analyzed by SEM in order to evaluate the effect of  
301 crosslinking on water resistance. As expected, when CA was not used, the fibrillar  
302 morphology of the electrospun PVOH/EOs films was completely lost after water  
303 immersion (Images C, F and I of Fig. 1A), as previously reported by other authors (Çay  
304 & Mirafteb, 2013; Destaye, Lin, & Lee, 2013). In contrast, the cross-linked films kept  
305 their fibrillary morphology (although a certain swelling was observed as deduced from  
306 the increase in the average fiber diameter- see Table 2), providing a better stability and  
307 evidencing the efficacy of the annealing process in promoting the esterification reaction  
308 between carbonyl groups in CA and hydroxyl groups in PVOH (see Figure 1B). In  
309 general, crosslinking reduces the interstitial spaces between the biopolymer chains, thus  
310 reducing molecular motion and preventing extensive swelling of the electrospun fibers  
311 (Çay & Mirafteb, 2013; Mirafteb et al., 2015; Santiago-Morales, Amariei, Letón, &  
312 Rosal, 2016).

313



314

315 **Figure 1.** SEM images of: **A)** PVOH, PVOH/LEO and PVOH/REO nanofibers and **B)**  
 316 PVOHc, PVOHcLEO and PVOHcREO nanofibers, before annealing (a, d, g), after  
 317 annealing (b, e, h) and after water immersion (c, f, i).

318

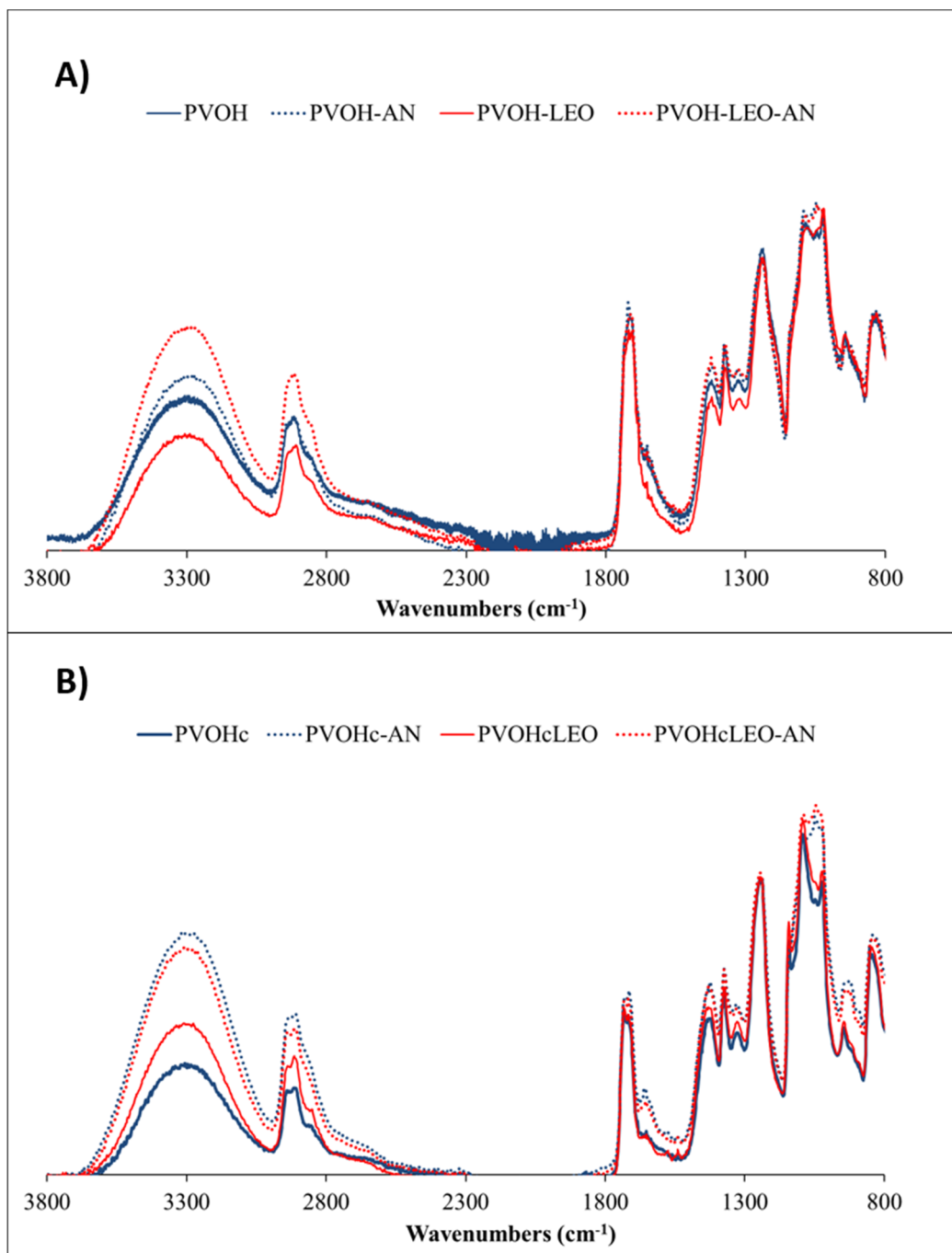
319 Infrared spectroscopy was used to evaluate the changes in the molecular structure of the  
 320 different electrospun mats obtained. As very similar spectra were obtained for the mats

321 with the two different EOs, only the ones containing LEO are shown for clarity. ATR-  
322 FTIR spectra of the neat and LEO-loaded PVOH or PVOH/CA electrospun films before  
323 and after the annealing process are gathered in Figure 2. The spectrum of neat PVOH, as  
324 previously described, showed an intense band between 3600 and 3200  $\text{cm}^{-1}$  ascribed to  
325 the stretching of O-H groups and other characteristic bands between 2840 and 3000  $\text{cm}^{-1}$ ,  
326 attributed to the stretching C-H from alkyl groups and between 1757-1710  $\text{cm}^{-1}$   
327 corresponding to the C=O stretching and C-O from acetate groups remaining from PVOH  
328 (Andrade, Barbosa-Stancioli, Mansur, Vasconcelos, & Mansur, 2008; Mansur et al.,  
329 2008).

330 For comparison purposes, the different spectra were normalized to the vibrational band  
331 at 1232  $\text{cm}^{-1}$ , assigned to the stretching vibration of C-O arising from alcohol and ester  
332 groups (Esparza et al., 2017). Upon EOs incorporation, no significant shifts were  
333 observed in the spectral bands of the neat PVOH or PVOH/CA independently of the final  
334 compositions of the films, thus, suggesting that EOs were not chemically interacting with  
335 the PVOH network. However, a decrease in the OH stretching band was seen upon EO  
336 incorporation.

337 Annealing is known to promote crystal formation in polymeric materials. From Figure  
338 2A, it can be clearly seen that annealing caused an increase in the OH stretching band  
339 centered at 3300  $\text{cm}^{-1}$ , probably indicating stronger hydrogen bonding interactions.  
340 Another remarkable change, was the intensity increase in the range from 1000-1100  $\text{cm}^{-1}$   
341 (C-O stretching in C-O-H groups and C-O-C groups), thus probably indicating that this  
342 area could arise from crystallizable chain segments in PVOH.

343



344

345 **Figure 2.** ATR-FTIR of: **A)** neat and LEO-loaded PVOH electrospun fibers and **B)** neat  
 346 and LEO-loaded PVOH/CA electrospun fibers before and after the annealing process  
 347 (indicated by -AN).

348

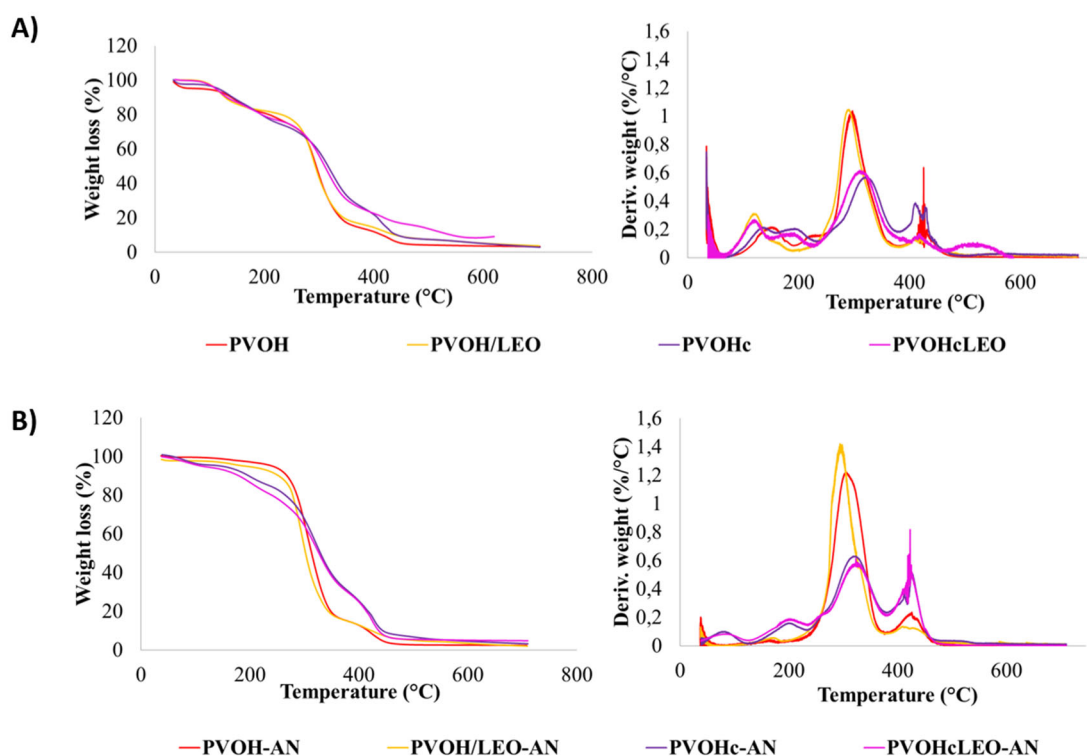
349 In contrast, additional spectral changes were observed in the samples containing CA  
 350 (Figure 2B). Similar to what was observed in the mats without CA, the heat treatment



351 caused an increase in the OH stretching band and in the range from 1000-1100  $\text{cm}^{-1}$ , both  
352 probably indicating increased molecular order in the PVOH chains. But apart from these  
353 changes, an increase in the bands from 1700 to 1750  $\text{cm}^{-1}$  arising from ester and  
354 carboxylic bonds was also observed in the heat-treated samples, thus confirming the  
355 crosslinking reaction. Presumably, during the heat treatment process, CA is first  
356 dehydrated to its anhydride and subsequently, the ester group is formed through the ring  
357 opening of the citric acid anhydride by the  $-\text{OH}$  groups present in the PVOH backbone  
358 and this process is repeated to complete the crosslinking (Nikfarjam, Taheri Qazvini, &  
359 Deng, 2014).

360 The thermal stability of the electrospun fibres was also assessed by thermogravimetric  
361 analysis. As very similar thermograms were obtained for the mats with the two different  
362 EOs, only the ones containing LEO are shown for clarity. Figure 3 shows the  
363 thermogravimetric curves of the neat and LEO-loaded PVOH or PVOH/CA electrospun  
364 films before and after the annealing process. Neat PVOH showed a first stage of mass  
365 loss (25 - 150  $^{\circ}\text{C}$ ), attributed to the loss of water molecules (both weakly and strongly  
366 interacting with the polymer). The second phase of mass loss (200 - 375  $^{\circ}\text{C}$ ) has been  
367 associated to the decomposition of the side chains of PVOH and the third stage ranging  
368 from 375 to 500  $^{\circ}\text{C}$  has been mainly related with polymer main chain degradation (Van  
369 Etten et al., 2014). Thus, the last two stages were responsible for the mass loss of around  
370 92%, which corresponded to the structural decomposition of PVOH.

371



372

373 **Figure 3.** TG and DTG curves of **A)** neat and LEO-loaded PVOH and PVOH/CA  
 374 electrospun fibers **A)** before and **B)** after the annealing process (indicated by -AN).

375

376 Considering that the neat LEO and REO had only one degradation peak ( $\sim 88^\circ\text{C}$ )  
 377 corresponding to the evaporation of the volatile compounds present in the essential oils  
 378 (Göksen et al., 2020), the shift to higher temperature ( $\sim 120^\circ\text{C}$ ) observed when the EOs  
 379 were incorporated within the PVOH matrices, suggest that the polymer increased their  
 380 thermal stability.

381 Interestingly, a new degradation peak at around  $\sim 185^\circ\text{C}$  (see Figure 3) appeared in the  
 382 samples containing CA, which probably corresponds to the degradation temperature of  
 383 neat CA, as it has been previously reported to occur in a single-step between 160 and 220  
 384  $^\circ\text{C}$  (Wyrzykowski, Hebanowska, Nowak-Wicz, Makowski, & Chmurzyński, 2011).

385 After the annealing process, a smoother peak was found at higher temperature ( $\sim 162^\circ\text{C}$ )  
 386 in PVOH-LEO-AN, evidencing that EOs were mostly degraded due to the annealing

387 treatment carried out at 170 °C, at a much higher temperature than the degradation  
388 temperature of both LEO and REO. It was also observed that the main degradation peak  
389 of PVOH was shifted to lower temperatures in PVOH-EOs-AN samples, indicating that  
390 the presence of LEO or REO detrimentally affected the degradation temperature of neat  
391 PVOH. Interestingly, a new degradation peak centered at ~ 89 °C was found in annealed  
392 samples containing CA, even in those prepared without EOs, suggesting the presence of  
393 less heat-stable molecules which were probably formed during the annealing, which  
394 degraded at this temperature.

395 Furthermore, one of the main changes that occurred after the annealing process was  
396 related to the stability of the PVOH macromolecules cross-linked with the CA. In fact,  
397 the thermal degradation during the second and third weight loss stage occurred more  
398 gradually and started at higher temperatures when CA was present, thus suggesting that  
399 the interactions between CA and PVOH had a protective impact on the thermal stability  
400 of the developed electrospun fibers. Thus, TGA results suggest that crosslinking had a  
401 positive effect on the thermal stability of the polymer material, while the inclusion of the  
402 essential oils had the opposite effect.

403 The degree of crystallinity has important effects in the physical properties of biopolymers.  
404 Thus, the thermal properties of the PVOH and PVOH/CA films and the active films  
405 containing EOs were also investigated by DSC analysis. Table 2 compiles the temperature  
406 of melting ( $T_m$ ), as well as the melting enthalpy ( $\Delta H_m$ ) and crystallinity ( $X_c$ ) obtained  
407 from the first heating run.  $\Delta H$  can be used to estimate the crystallinity of a polymer  
408 sample.

409 Interestingly, just after electrospinning, i.e. without an annealing process, the fibers  
410 obtained were amorphous and no thermal transition was seen through DSC. In contrast,  
411 the annealing process promoted crystal development, which was different depending on

412 the initial composition of the fibers. As seen in Table 2, the presence of EOs, increased  
413 the melting point of the annealed fibers, indicating that the crystals present in the materials  
414 were bigger or more perfect. This could be ascribed to a plasticization effect of the oils,  
415 facilitating chain mobility and packing during the annealing process, fact which led to  
416 higher melting temperature and higher melting enthalpy. In contrast, Lan et al., (2019)  
417 stated that the addition of d-limonene in PVOH matrices hindered crystallization of the  
418 PVOH molecules, thus resulting in lower melting temperature and enthalpy which was  
419 ascribed to the d-limonene impeding hydrogen bonding not only within PVOH molecules  
420 but also between PVOH molecular chains. Similarly, Hernández-López et al., (2019)  
421 reported an  $\Delta H$  decrease when pine essential oil was added to PLA-based composite  
422 fibers, indicating that the incorporation of the dispersed phase into the matrix reduced the  
423 crystallization of the biopolymer matrix. It should be stressed, however, that in the  
424 thermal data provided in the mentioned works was obtained after erasing the thermal  
425 history of the materials, while in our case the thermal parameters were obtained during  
426 the first heating ramp. Regarding the effect of citric acid incorporation, similarly no  
427 transition was observed before annealing, while after this heat treatment crosslinking was  
428 promoted, thus limiting the ability of polymeric chains to reorganize, thus, resulting in  
429 lower melting points, although similar crystallinity values. This implies that although the  
430 amount of crystals formed during annealing was similar in the samples with or without  
431 citric acid, they were smaller or less perfect when crosslinking took place.

432

433 **Table 2.** Parameters from DSC curves of annealed electrospun fibers.

<b>Sample</b>	<b>Peak Temp. (°C)</b>	<b>Melting Enthalpy (J/g)</b>	<b>Onset Temp. (°C)</b>	<b>End Temp. (°C)</b>	<b>Xc (%)</b>
PVA	178.4±1.1 <sup>a</sup>	16.1±1.4 <sup>a</sup>	160.2±1.8 <sup>a</sup>	191.0±1.0 <sup>a</sup>	11.6±1.0 <sup>a</sup>

PVA/LEO	189.7±6.5 <sup>b</sup>	26.6±10.2 <sup>a</sup>	158.4±5.1 <sup>a</sup>	205.8±1.0 <sup>b</sup>	19.2±7.4 <sup>a</sup>
PVA/REO	194.6±0.4 <sup>b</sup>	24.2±11.1 <sup>a</sup>	173.4±1.3 <sup>b</sup>	207.9±8.7 <sup>b</sup>	17.4±8.0 <sup>a</sup>
PVAc	166.1±0.7 <sup>c</sup>	16.8±2.3 <sup>a</sup>	145.5±4.4 <sup>c</sup>	181.2±3.4 <sup>c</sup>	12.6±1.6 <sup>a</sup>
PVAcLEO	184.5±1.0 <sup>b</sup>	29.4±10.3 <sup>a</sup>	155.9±6.8 <sup>ac</sup>	201.4±2.6 <sup>b</sup>	21.2±7.4 <sup>a</sup>
PVAcREO	179.2±2.0 <sup>a</sup>	18.9±5.8 <sup>a</sup>	154.0±1.8 <sup>a</sup>	194.7±2.8 <sup>a</sup>	13.6±4.2 <sup>a</sup>

434 Mean values ± standard deviations (n=3). Different letters at column denote significant  
435 differences (p<0.05).

436

### 437 **3.2. Challenge tests**

438 Challenge tests on chicken breast fillets were carried out to ascertain the antioxidant and  
439 antimicrobial effectiveness of the active fiber mats on real food samples. Although TGA  
440 analysis revealed that EOs were mostly degraded during the annealing process, the  
441 functional properties of the developed electrospun fibers were tested to determine if the  
442 LEO or REO remaining after the annealing process had antioxidant and antimicrobial  
443 properties. In fact, previous works carried on EOs loaded-casting films revealed  
444 antilisterial activity even though the losses of volatile compounds during the film drying  
445 ranged between 39 and 99 %, depending on the EOs and EOs:biopolymer ratio (Sánchez-  
446 González, Vargas, González-Martínez, Chiralt, & Cháfer, 2011).

447 Initially, the physicochemical quality of the packaged and unpackaged samples was  
448 evaluated in terms of pH and color parameters. Table 3S from the Supplementary Material  
449 shows the changes in the pH values during the storage in the different chicken samples  
450 covered by electrospun structures. The initial pH value of the fillet was ~ 5.97. The pH  
451 development in the chicken breast fillets can be affected by both the microbial growth  
452 and the oxidation process which occurred to a different extent in the different packaged  
453 samples. As expected, the pH of unpackaged samples slightly increased during the storage

454 time, ranging between 5.97 and 6.81. This agrees with the fact that foods stored under  
455 aerobiosis and rich in proteins and free aminoacids, such as chicken breast fillets, present  
456 a pH increase as the number of microorganisms that cause spoilage increases (Křížek,  
457 Vácha, Vorlová, Lukášová, & Cupáková, 2004; Ntzimani, Paleologos, Savvaidis, &  
458 Kontominas, 2008), as it will be detailed below. Furthermore, the proteolytic activity also  
459 results in the production of basic compounds which could contribute to the increase in pH  
460 (Vinci & Antonelli, 2002). In contrast, the pH of the packaged samples did not vary  
461 significantly during the storage time or it was even slightly decreased, confirming that  
462 these samples were better protected.

463 Table 3 shows the changes in the color parameters ( $L^*$ ,  $a^*$  and  $b^*$  values) of the food  
464 samples during storage at refrigeration conditions (4 °C).  $L^*$  values decreased during  
465 storage in all the samples, indicating that the chicken fillets became darker, change that  
466 was more accentuated in unpackaged samples. Regarding the sample redness ( $a^*$  values),  
467 it increased during storage time in unpackaged samples. In contrast, it was significantly  
468 lower in packaged chicken breast fillets and there was a trend to decrease during the  
469 storage time in those prepared with electrospun fibers containing EOs, suggesting some  
470 interaction between the essential oils and the meat pigments, thus affecting redness. The  
471 sample yellowness ( $b^*$  values) increased to a greater extend in unpackaged chicken breast  
472 fillets which could be related with the greater lipid oxidation of these samples as it will  
473 be detailed bellow.

474

475

476

477

478 **Table 3.** Color measurement of L\*, a\* and b\* values during storage of packaged chicken  
 479 breast samples

		Storage time (day)			
		0	1	4	7
L*	Control	49.71±1.05 <sup>a1</sup>	49.23±1.05 <sup>a1</sup>	42.75±1.50 <sup>a2</sup>	43.41±0.82 <sup>a2</sup>
	PVOHc	50.08±0.90 <sup>a1</sup>	49.20±1.39 <sup>a1</sup>	47.95±1.05 <sup>b1</sup>	47.15±1.03 <sup>b1</sup>
	PVOHcLEO	55.44±0.79 <sup>b1</sup>	56.09±2.08 <sup>b1</sup>	54.18±0.57 <sup>c1</sup>	51.89±1.35 <sup>c2</sup>
	PVOHcREO	54.60±1.68 <sup>b1</sup>	53.75±1.35 <sup>ab1</sup>	51.54±0.47 <sup>c1</sup>	51.45±0.31 <sup>c1</sup>
a*	Control	2.11±0.05 <sup>a1</sup>	2.37±0.09 <sup>a2</sup>	2.43±0.10 <sup>a23</sup>	2.58±0.06 <sup>a3</sup>
	PVOHc	1.85±0.10 <sup>b1</sup>	2.03±0.11 <sup>b1</sup>	1.91±0.05 <sup>b1</sup>	2.02±0.11 <sup>b1</sup>
	PVOHcLEO	1.87±0.07 <sup>b1</sup>	1.71±0.06 <sup>c2</sup>	0.90±0.06 <sup>c2</sup>	1.12±0.09 <sup>c1</sup>
	PVOHcREO	1.93±0.03 <sup>ab1</sup>	1.88±0.11 <sup>bc1</sup>	0.87±0.08 <sup>c2</sup>	1.04±0.18 <sup>c1</sup>
b*	Control	7.75±0.41 <sup>a1</sup>	8.34±0.33 <sup>a1</sup>	9.36±0.46 <sup>a2</sup>	10.18±0.34 <sup>a2</sup>
	PVOHc	7.69±0.23 <sup>a1</sup>	8.07±0.25 <sup>a1</sup>	8.67±0.22 <sup>a12</sup>	8.84±0.43 <sup>b2</sup>
	PVOHcLEO	7.82±0.12 <sup>a1</sup>	8.56±0.69 <sup>a12</sup>	8.73±0.25 <sup>a2</sup>	8.97±0.18 <sup>b2</sup>
	PVOHcREO	7.54±0.33 <sup>a1</sup>	7.75±0.34 <sup>a1</sup>	8.64±0.30 <sup>a2</sup>	8.75±0.48 <sup>b2</sup>

480 Mean values ± standard deviations (n=3).

481 Different letters in the same column show significant differences (p<0.05) among  
 482 samples.

483 Different numbers in the same file show significant differences (p<0.05) during the  
 484 storage time.

485

486 Lipid oxidation values during chilled storage of the chicken breast fillets are gathered in  
 487 Table 4 for the samples covered with the different electrospun structures. Samples  
 488 packaged with PVOHc structures inhibited lipid oxidation around 19%, whereas samples  
 489 packaged with EOs-loaded electrospun fibers prevented lipid oxidation to a greater  
 490 extent, reaching a ~43 and ~65 % of lipid oxidation inhibition for PVOH films prepared  
 491 with REO and LEO, respectively. The protective effect observed for PVOHc electrospun  
 492 coatings can be ascribed to the presence of non-cross-linked (free) CA in the chicken  
 493 samples, which may contribute to improve food quality and shelf-life (Doležalová,  
 494 Molatová, Buňka, Březina, & Marounek, 2010; Gonzales-Fandos et al., 2009).  
 495 Interestingly, this effect was enhanced with the incorporation LEO and REO, indicating  
 496 that even most of the EOs were degraded during the annealing process, the remaining  
 497 compounds had antioxidant properties as well as antimicrobial activity, as it will be  
 498 detailed bellow.

499

500 **Table 4.** 2-Thiobarbituric acid reactive substances (TBARS) and estimated lipid  
 501 oxidation inhibition in fresh chicken breast fillets (day 0) and in the packaged samples  
 502 after 7 days of storage.

		TBARS (mg MDA/kg)	Lipid oxidation inhibition (%)
Day 0	Control	0.29±0.04 <sup>a</sup>	-
	Control	0.91±0.03 <sup>b</sup>	0
Day 7	PVAc	0.79±0.06 <sup>b</sup>	18.71±0.65
	PVAc-LEO	0.51±0.02 <sup>c</sup>	64.90±2.35
	PVAc-REO	0.64±0.04 <sup>d</sup>	42.95±6.34

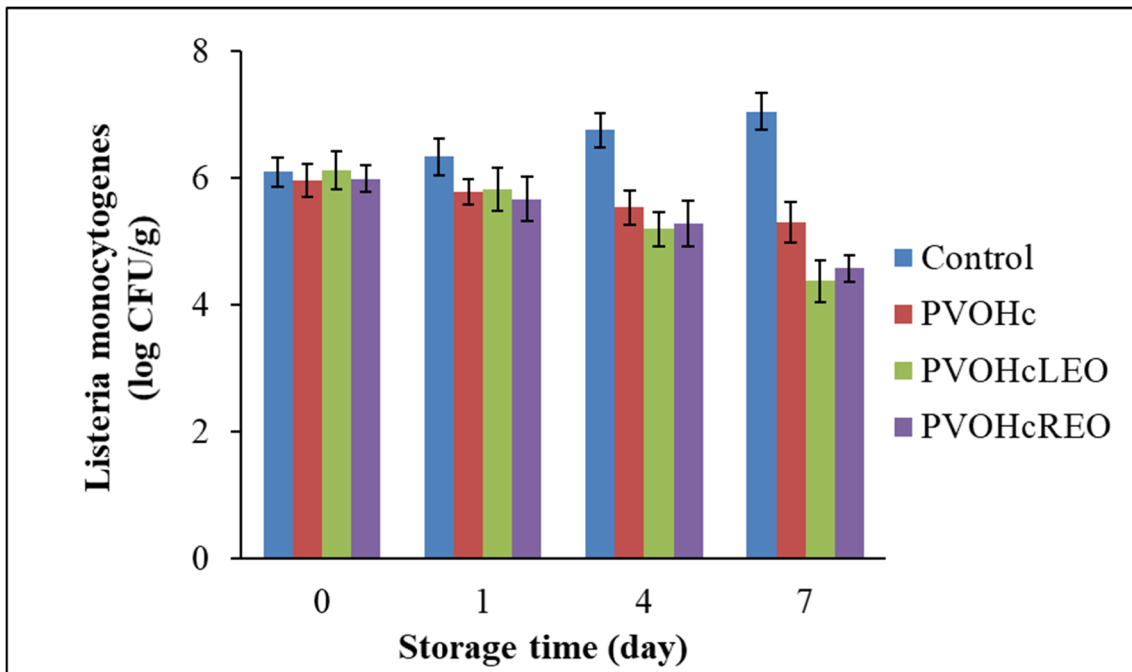
503 Different letters in the same column show significant differences (p<0.05).

504



505 Regarding the antimicrobial activity, viable counts of *L. monocytogenes* after 7 days of  
506 incubation at 4 °C on chicken breasts are displayed in Figure 4. An increase of about 1  
507 log in the counts of pathogenic bacteria was observed in the chicken breast controls after  
508 7 days of storage. In contrast, samples with PVOHc did not significantly increase the  
509 viable counts while a growth inhibition was observed in the chicken breasts packaged  
510 with PVOHc/EOs electrospun structures, exhibiting up to 1 log decrease with respect to  
511 the initial values and with the PVOH/LEO having slightly more antimicrobial effect than  
512 REO.

513 Therefore, in terms of antioxidant and antimicrobial growth, the use of EOs-loaded  
514 PVOH films significantly prolonging the shelf life of chicken breast fillets, being LEO  
515 more efficient than REO. These differences can be ascribed to the presence of the most  
516 active compound, 1,8 cineole, which was present in greater amounts in LEO (~ 69.87 %)  
517 than in REO (~55.80 %) (Göksen et al., 2020).



518  
519 **Figure 4.** Changes in *L. monocytogenes* (log CFU/g) during the storage of packaged  
520 chicken breast samples.

521

#### 522 **4. CONCLUSIONS**

523 Active cross-linked electrospun PVOH fibers were prepared containing two different  
524 EOs. Citric acid (CA) was used as a cross-linker together with an annealing process with  
525 the aim of maintaining the integrity of the fibers in contact with high moisture foods. The  
526 average diameter of the fibers decreased with the incorporation of both EOs and CA,  
527 mainly ascribed to the increase in electrical conductivity. The morphology and porous  
528 structure of the electrospun fibers containing EOs and PVOH was successfully  
529 maintained to a greater extent in annealed samples after being immersed in water,  
530 evidencing the efficiency of the crosslinking process. In contrast, this effect was not  
531 observed in the annealed electrospun samples prepared without CA.

532 Furthermore, physicochemical and thermal stability of annealed electrospun PVOH food  
533 packaging structures were successfully improved by the incorporation of CA. The  
534 materials were tested to improve preservation of chicken breast samples, even though the  
535 annealing process probably degraded most of the EOs, as observed through TGA. These  
536 active packaging coatings containing LEO and REO enhanced the shelf-life of chicken  
537 breast fillets, reducing the lipid oxidation process and reducing *Listeria* counts during  
538 cold storage. Therefore, this work provides a simple method to obtain PVOH-coatings  
539 with suitable integrity for use in the active preservation of fresh food products like meat.

540

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548

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