

1 **Use of nanosilica- or nanolime-added TEOS to consolidate cementitious materials in**  
2 **heritage structures: physical and mechanical properties of mortars**

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10 **ABSTRACT**

11 Tetra-ethoxysilane or tetraethyl orthosilicate (TEOS), also known as ethyl silicate, traditionally  
12 applied to consolidate stone, has recently begun to be used on Portland cement mortars and  
13 concrete with promising results. TEOS not only fills the pores in the substrate, but reacts with  
14 the cement favouring the precipitation of new C-S-H gels that densify and strengthen the  
15 materials. This study explored the effectiveness of new TEOS-based treatments bearing  
16 nanosilica (NS) or nanolime (NC) in and their compatibility with cement materials found in the  
17 built heritage, given the participation of the various products in the pozzolanic reactions that  
18 may induce additional C-S-H gel. The physical and hydric properties (mechanical strength,  
19 porosity, surface gloss and colour, water vapour permeability and low pressure water  
20 absorption) of a Portland cement mortar were determined before and after applying the  
21 consolidants. Ethyl silicate, alone or in conjunction with nanolime (4:1), proved to most  
22 effectively raise material strength and improve water repellence while conserving the original  
23 colour and gloss.

24 **Keywords:** Portland cement; mortar; heritage; consolidant; TEOS; nanosilica; nanolime

25 **1. Introduction**

26 Concrete attained noble architectural status in the twentieth century, when it began to be used on  
27 building exteriors and in sculpture [1]. While durable if properly designed, produced and  
28 maintained, in practice concrete often needs to be repaired [2,3], as it is subject to cracking or  
29 spalling attributable to a variety of causes [4,5]. European standard EN 1504 [6] defines the  
30 requirements that must be met by products and systems to protect and repair concrete structures  
31 in terms of the degree of deterioration (including products to be injected into the concrete,  
32 adhesives, products to protect reinforcement and surface treatments). However, intervention to

33 conserve cultural assets is subject to very specific principles [7,8] that reject overly aggressive  
34 treatments, recommend action only where and to the extent strictly necessary with products that  
35 alter neither the physical-chemical properties of the materials nor overall aesthetics and call for  
36 pre-testing, along with other requisites.

37

38 Material consolidation, understood to consist in the application of a product to a crumbling  
39 surface to improve cohesion, mechanical properties and the adherence of the altered layers to  
40 the healthy substrate, is a common practice in heritage stone conservation and restoration. Tetra-  
41 ethoxysilane (TEOS), also known as ethyl silicate ( $\text{Si}(\text{OEt})_4$ ), is one of the most widely used  
42 consolidation products [9] for its compatibility with and stability in [10] such substrates.

43

44 It hydrolyses in the presence of water (moisture in the atmosphere or in the pores of the  
45 material), generating silanol groups (Si-OH) that polymerise to form siloxane bonds (Si-O-Si)  
46 and ultimately inducing the precipitation of a silica gel that raises stone strength.

47 <sup>PL</sup>

48 This gel fills the pores and adheres to substrates with a Si base, also by forming siloxane (Si-O-  
49 Si) bonds [11]. It consequently adheres poorly to non-siliceous materials (such as limestone or  
50 marble). The treatment layer may also crack during drying on a number of types of substrates.  
51 Strategies developed to prevent this latter difficulty envisage combining the material with  
52 surfactants [12], additives [13] or nanoparticles [14,15,16,17].

53

54 Recent studies on the consolidation of cement-based materials, less intensely researched to date  
55 [18,19,20], have assessed the effect of TEOS on the physical properties of mortars with  
56 promising results: mechanical strength and durability (reduction of chlorides and carbonation  
57 depth) were enhanced with no significant alteration of surface colour or gloss. Nonetheless,  
58 TEOS has been shown to penetrate compact materials in depth [19] and in early age (1 day)  
59 cements the formation of alcohols during TEOS hydrolysis retards hydration and favours the  
60 formation of a more porous, weaker structure [21].

61

62 Nanosilica (nano- $\text{SiO}_2$ , NS)-based surface consolidating treatments [22], on the grounds of the  
63 reactivity of two of the main products of Portland cement hydration, portlandite ( $\text{Ca}(\text{OH})_2$ ) and  
64 C-S-H gel [10,20,23], and their ability to fill the pore systems, have been attempted, with  
65 varying success on hardened concrete and cement mortars, for whilst they can densify these  
66 materials, their penetration capacity is lower than in TEOS. In fact, NS compared with a variety  
67 of organic or inorganic surface treatments has been found to be less effective in reducing water  
68 absorption [24], chloride migration or  $\text{CO}_2$  penetration [19; 24] or in raising mechanical strength

69 [25]. Nonetheless, results are quite dependant on the microstructure of the coated materials  
70 since in more porous substrates nanoparticles are able to penetratete deeper and therefore act  
71 more effectively on reducing the migration of water and aggressive agents [26]. Like this,  
72 Sanchez et al. [27] by forcing the migration of nanosilica under the action of an electric on a  
73 cement mortar enhanced the surface of the material and thus the durability of the entire bulk and  
74 confirmed the sealing ability and pozzolanic activity of the nanoparticles [28].

75

76 Nanolime (nano- $\text{Ca}(\text{OH})_2$ , NC), traditionally used to consolidate carbonate-based materials  
77 [29], has also been tried recently as a surface treatment for porous cement mortar in conjunction  
78 with nanosilica (a coat of NS followed by a coat of NC) with promising results in terms of  
79 mechanical strength [25].

80

81 In the pursuit of the most ideal surface treatments to consolidate heritage cement assets by  
82 capitalazing the advantages of the aforementioned products (i.e. TEOS is more effective in  
83 filling pores smaller than 50 nm [22]; NS has a extremely high pozzolanic reactivity [22]); and  
84 their physicochemical compatibility and feasible synergy potential, this study explored the  
85 performance of TEOS and NS or NC blends on a cement mortar. In fact, recent studies have  
86 shown that nanoparticles can improve the performance of coatings on cement-based materials,  
87 by increasing their surface roughness and thus water repellency [30], or vice versa, the  
88 performance of nanoparticles can be enhanced by coatings like in a recent study of Hou et al.  
89 [31] in which nanosilica particles coated with polymethylhydrosiloxane enhanced the water  
90 proofing properties of a cement mortar.

91

92 The effectiveness and compatibility of the new explored consolidating blends were assessed in  
93 keeping with heritage intervention criteria and the possible changes in the physical and hydric  
94 properties of a Portland cement mortar: mechanical strength, porosity, surface gloss and colour,  
95 water vapour permeability and low pressure water absorption.

96

## 97 **2. Experimental**

### 98 *2.1. Materials and treatment application*

99 Portland cement (CEM I 42.5N/SR; Table 1) specimens of two sizes, 10x10x60 mm and  
100 60x70x10 mm, were prepared with a water/cement ratio of 1:2 and a cement/sand ratio of 1:5.  
101 The fines ( $\text{Ø} < 1$  mm) were removed from the standardised sand to favour mortar porosity. This  
102 is a way to simulate the porosity of a degraded mortar in need of consolidation. TEOS (CTS  
103 ESTEL 1000), used as a reference, and combinations of the product with two proportions of

104 nanosilica (BASF MEYCO MS 685) and nanolime (CTS NANORESTORE) (Table 2) were  
105 drip-applied to one side of specimens (previously cured in a humidity chamber at RH>95 % and  
106 a temperature of 21(±2) °C for 28 days) until apparent saturation, defined as the point when the  
107 surface remained moist for at least 1 min. Inasmuch as TEOS and NS could not be blended due  
108 to differences in the polarity of their respective solvents, NS was applied first, followed by  
109 TEOS. Manufacturer recommendations were followed to apply NC in conjunction with TEOS  
110 to prevent the appearance of carbonate veils on the surface, namely by placing a moist paper  
111 pulp poultice for 8 h on 9 g Japanese paper over the surface to be treated, all covered by  
112 polyethylene terephthalate film. In Table 3, the total amount of products consumed by the  
113 mortars either in one step (TEOS; TEOS and NC) or two (NS followed by TEOS) is listed.  
114 Used ratios were based on preliminary application tests and represent the volume of each  
115 product blended together before application for TEOS and NC mixes or individually applied for  
116 NS and TEOS treatments.

117 The physical properties of the specimens were characterised after curing the treatments for  
118 5 weeks at 21(±2) °C and relative humidity of 45(±5) %

119

## 120 *2.2. Methods*

121

122 Pore size distribution in the mortars before and after application of the treatments was found on  
123 a Mercury Micromeritics Autopore IV 9500 V1.05 porosimeter, applying a maximum intrusion  
124 pressure of 22 749.92 MPa, which corresponds to a pore diameter of 0.0067 µm. Since  
125 treatments were applied to one side of the specimens, porosity results of 1cm<sup>3</sup> fragments were  
126 just used for comparison purposes, in no case as absolute values.

127

128 Possible variations in water vapour permeability were determined to RILEM recommendations  
129 [33]. To ascertain material resistance to water absorption, Karsten tube tests were performed  
130 [34]. The latter test is particularly apt in studies of protective treatments for the historic heritage  
131 since can be carried out in situ.

132

133 After 28 days of initial curing, the untreated and treated 10x10x60 mm specimens were tested  
134 for compressive and flexural strength, respectively on an IBERTEST Autotest 200/10SW frame  
135 and a NETZSCH 6.111.2 device [35]. The resonant frequency signals generated in flexural  
136 mode vibration (Young's modulus or the dynamic modulus of elasticity) were recorded as  
137 specified in international standard UNE-EN ISO 12680-1: 2007 [36] on a GrindoSonic MK5  
138 “Industrial” analyser.

139

140 The chromatic coordinates ( $L^*$ , lightness, and hue,  $a^*$  (red+/green-) and  $b^*$  (yellow+/blue-))  
141 were measured in 60x70x10 mm mortar specimens before and after treatment with a Minolta  
142 CM 2500 D spectrophotometer. The total colour variations ( $\Delta E^*$ ) were calculated from the  
143 following equation ( $\Delta E^* = (\Delta L^*)^2 + (\Delta a^*)^2 + (\Delta b^*)^2)^{1/2}$ ) [37]. Surface gloss was determined on the  
144 same surfaces at an 85° angle using a Minolta Multi Gloss 268 glossimeter sensitive to  
145 variations in scanty glossy surfaces (0-160  $\mu\text{m}$  gloss units).

### 146 3. RESULTS AND DISCUSSION

#### 147 3.1 Porosity

148 The cement mortar prepared had an open porosity of 12.85 % (vol.), with pores of a diameter of  
149 over 10  $\mu\text{m}$  (entrained air pores) accounting for 24.78 % and capillary pores, the ones that  
150 decrease over time as cement reacts with capillary water, i.e., with a diameter of 0.01  $\mu\text{m}$ -10  
151  $\mu\text{m}$  (Table 4, Fig. 1), for 73.1 %. The remaining 2.2 % consisted of pores of C-S-H gel [38].

152 Whilst all the treatments lowered total mortar porosity, the ones with the highest TEOS content  
153 were particularly effective, reducing the value by 20 % to 32 % (NS+TEOS (1:1), TEOS and  
154 TEOS+NC (4:1)) (Table 4), while also filling entrained air pore more intensively.

155 TEOS and the TEOS+NC blends were most effective in lowering mortar capillary porosity,  
156 especially in respect of the smallest capillary pores (1-0.01  $\mu\text{m}$ ), and the only ones to reduce the  
157 mean pore size. Such pore structure refinement, which favours material durability by hindering  
158 the diffusion of aggressive ions, is the result of the physical pore filling (Hou et al. [22] reported  
159 that TEOS may penetrate pores under 50 nm) and the precipitation of C-S-H gel, the product of  
160 the pozzolanic reaction between TEOS and the cement matrix [10,20,21] and TEOS with  
161 nanolime [20].

162 The two combinations of TEOS and NS induced a substantial rise in the number of the smallest  
163 capillaries (1-0.01  $\mu\text{m}$ ), the outcome of the conversion of larger pores, and a considerable  
164 decline in gel pores (<0.01  $\mu\text{m}$ ). Unlike NC, the application of NS before ethyl silicate would  
165 not favour the formation of pore gels. The failure of NS to penetrate in the pore matrix, as  
166 illustrated by SEM micrography and Ca and Si elemental mapping on the mortar surface (Fig.  
167 2), would hamper subsequent access by TEOS which ultimately could fill larger pores.

#### 168 3.2 Water vapour permeability

169 The treatments lowered the water vapour transmission rate by 50 %-63 % (Table 5). Hou et al.  
170 [26] observed steeper declines in porosity in specimens made with higher w/c ratios in cement  
171 pastes treated with TEOS and a colloidal nano-SiO<sub>2</sub>.

172 At 59 %-63 %, the TEOS+NC blends induced the steepest declines, followed by the treatments  
173 with a higher ethyl silicate content, namely TEOS and NS+TEOS (1:9), with similar values (54  
174 %-56%) (Table 5). Generally speaking, the greater the pore structure refinement the greater was  
175 the reduction in permeability (with the exception of treatment NS+TEOS (1:9)). Whilst  
176 reductions of over 25 % are not recommended for heritage stone conservation [39], that  
177 threshold is not applicable to cementitious materials, in which the pores in the paste become  
178 more refined as the hydration reactions progress: the volume of capillary pores, which primarily  
179 govern the transport properties of cement-based materials [26], declines while the volume of gel  
180 pores rises. Material durability is in fact enhanced with such refinement, given the lesser  
181 possibility of expansion or contraction due to moisture gradients or acidic gas migration [26].

### 182 3.3. *Low pressure water absorption*

183 The possible changes in the surface characteristics of the mortars after application of the  
184 treatments, i.e., TEOS-prompted water repellence [40] and pore filling, were determined with  
185 the low pressure water absorption test.

186 The water absorbed by the specimens through Karsten tubes before and after applying  
187 TEOS+NS or TEOS+NC are shown in Figure 3, while Table 6 gives the values for each product  
188 of the two coefficients traditionally used to determine water repellence in historic materials on  
189 the grounds of those curves. All the products applied induced substantial declines in water  
190 absorption, ranging from 62% to 83%. De Witte et al. [41] deemed that a material is resistant to  
191 water penetration when amount of water absorbed measured by a Karsten tube is greater than or  
192 equal to 0.5 cm<sup>3</sup> between 5 and 15 minutes after starting the test (period of time in which the  
193 absorption rate is highest). Further to that criterion, the untreated mortar would itself be  
194 rainwater-resistant (Table 6). However, according to the Nwaubani and Dumbelton's [42]  
195 threshold value ( $\leq 0.35$  kg/m<sup>2</sup>h) for assessing water repellency, only the mortars treated with  
196 NS+TEOS (1:9) and TEOS+NC (4:1), the combinations with the highest TEOS content, would  
197 be effectively protected.

198 As these treatments induced consolidation rather than water repellence strictly speaking, they  
199 proved to be highly effective in terms of lowering water ingress and with it the amount of any

200 dissolved aggressive ions, observation reported by other authors after the application of only  
201 TEOS [18,26,40].

202

### 203 *3.4 Colour and gloss*

204 The values of the 5 week chromatic parameters ( $\Delta E^* = (\Delta L^*)^2 + (\Delta a^*)^2 + (\Delta b^*)^2)^{1/2}$ ) and gloss  
205 ( $\Delta G$ ) in the mortars are given in Table 7.

206 With the exception of the blend with a high nanolime content (TEOS+NC (1:4)), all the  
207 treatments darkened the mortar surface (decline in  $L^*$ ). In the products with a high TEOS  
208 content in particular, the colour change was perceptible to the naked eye, assuming a threshold  
209 of  $\Delta E^* > 3$  (for colour change in general [43]). Nonetheless, all but one of the colour variations  
210 were wholly acceptable for the purposes of ancient building material conservation ( $\Delta E^* \leq 5$ ;  
211 [43]). The exception was the change prompted by the NS+TEOS (1:9) blend, where  $\Delta E^* > 10.50$ .  
212 The results of applying TEOS separately were consistent with those reported by Pigino et al.  
213 [18] for Portland cement concretes, with  $\Delta E^*$  values of around 5 or even lower ( $\Delta E^* = 2.6$ ),  
214 depending on the water/cement ratios in the cement mixes. In an earlier study, Barberena-  
215 Fernandez et al. [24] found that separately, nanosilica and nanolime induced much lower,  
216 imperceptible ( $\Delta E^* < 2$ ) variations in colour in the same type of substrate. They also observed  
217 that nanosilica modified surface gloss slightly, albeit perceptibly to the naked eye, assuming the  
218  $\Delta G^* > 2$  threshold established by García and Malaga [44]. TEOS+NS-based treatments also  
219 affected surface gloss, more intensely at higher proportions of NS ( $\Delta G = 2.5-3.9$ ). Such  
220 variations may nonetheless be regarded as acceptable.

### 221 *3.5 Mechanical properties*

222 Further to the pre- and post-treatment mortar mechanical strength findings, at 37.71 MPa,  
223 untreated compressive mortar strength was slightly lower than specified in European standard  
224 EN 197-1 [45] for a 28 day, type I 42.5 cement (42.5 MPa-62.5 MPa). That was attributable to  
225 the low cement/sand ratio and the removal of fines during cement preparation to favour  
226 treatment penetration (Fig. 4). Flexural strength was 7.88 MPa (Fig. 4). Both types of strength  
227 rose after application of all the treatments without exceeding the limits recommended for the  
228 conservation of stone materials (treated specimen strength/untreated specimen strength  $< 1.5$ )  
229 [46].

230 The greatest rise in compressive (46 %) and flexural (36 %) strength was observed in the  
231 specimens with ethyl silicate (TEOS) as the sole component, which was consistent with the fact

232 that they also exhibited the steepest decline in total and entrained air porosity, corroborated by  
233 the excellent linear correlation between the latter and compressive strength (Fig. 5). The next  
234 strongest specimens were the ones treated with ethyl silicate-high blends: (TEOS+NC (4:1) and  
235 NS+TEOS (1:9)). The rise in compressive and flexural strength in the former was respectively  
236 36 % and 33 % and in the latter 39 % and 35 %.

237 The separate application not only of nanolime but also of nanosilica to the same substrate in a  
238 prior study by Barberena-Fernández et al. [25] prompted moderate (14 %) to low (3 %) rises in  
239 compressive strength. Nanosilica ineffectiveness was due either to surface carbonation, which  
240 would impede the pozzolanic reaction due to the lack of portlandite on the surface, or to  
241 treatment viscosity (Table 2), which would hamper its penetration into the substrate.  
242 Nonetheless, when used in appropriate proportions (1:5:1) [25], due to the pozzolanic reaction  
243 between the two types of nanoparticles, they induced greater rises than the combinations  
244 containing low concentrations of ethyl silicate studied here (NS+TEOS (1:1) and TEOS+NC  
245 (1:4)).

246  
247 The results of the acoustic resonance spectroscopic trials, a non-destructive method particularly  
248 apt for architectural heritage conservation studies, showed that at 31.5 GPa, Young's modulus  
249 or the dynamic modulus of elasticity was high in the untreated specimens (Fig. 6). The products  
250 that induced the steepest increase in material stiffness (rises in Young's modulus of 13%-18 %) were,  
251 logically, the same ones that prompted the highest rises in mechanical strength: those with  
252 more ethyl silicate in their composition, TEOS, TEOS+NC (4:1) and NS+TEOS (1:9), which  
253 filled the entrained air pores most effectively.

254

#### 255 **4. Conclusions**

256 This study explores the potential of blends of TEOS with different amount of NS or NC  
257 particles for the consolidation of cement-based historic assets. Of all these treatments studied,  
258 only calcium hydroxide nanoparticles combined with ethyl silicate (TEOS) (TEOS+NC (4:1))  
259 exhibited great affinity for conservation of cement mortar. Both, TEOS alone or combined with  
260 NC (4:1) raised mechanical strength by over 35 %, reduced the mean pore diameter and  
261 improved water repellence which would favour material durability by hindering the diffusion of  
262 aggressive ions. Furthermore, they did not perceptibly modify surface gloss or colour unlike the  
263 treatments with NS contrary, the presence of NS (NS+TEOS (1:9) and NS+TEOS (1:1)).

264 These results reveal the necessity of further exploring new blends that combine nanoparticles  
265 with either organic or inorganic coatings by optimizing the dosages of both components and the  
266 application procedures that ensure better penetration of the treatments.

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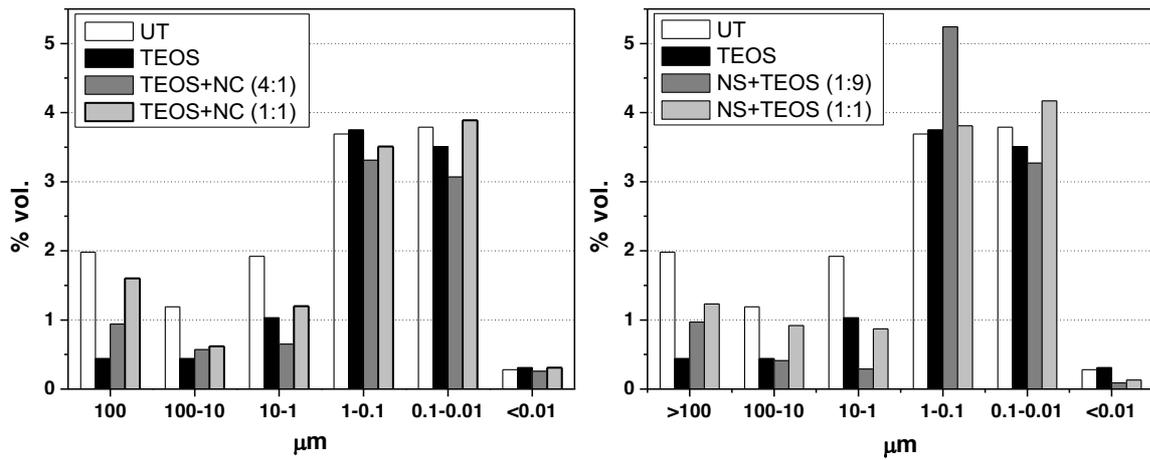
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431 Fig. 1. Pore size distribution (vol.%) in untreated (UT) and, ethyl TEOS-treated and TEOS+NC (left) and  
 432 TEOS+NS (right) -treated mortars.

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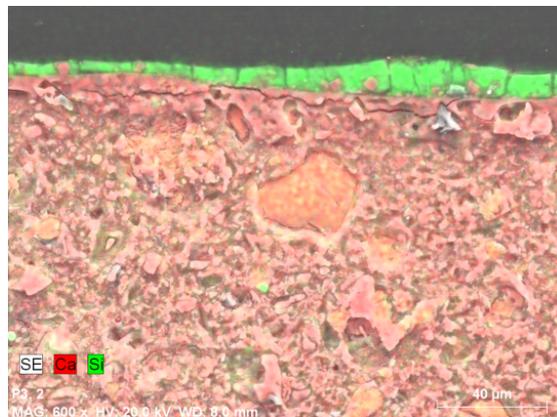
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454 Fig. 2. BSEM micrograph and Ca and Si mapping in nanosilica-coated mortar.

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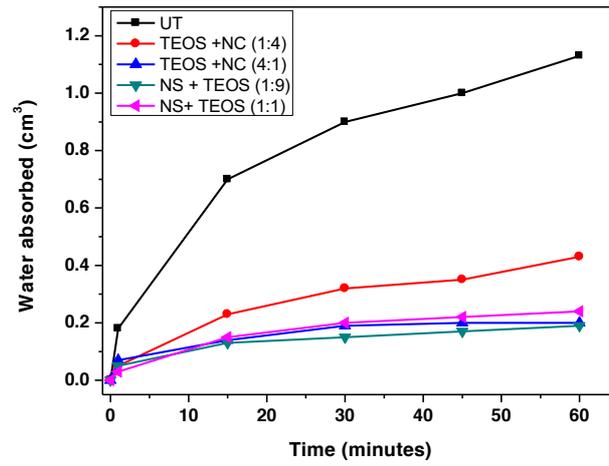


Fig. 3. Low pressure water absorption through Karsten tubes in treated and untreated specimens.

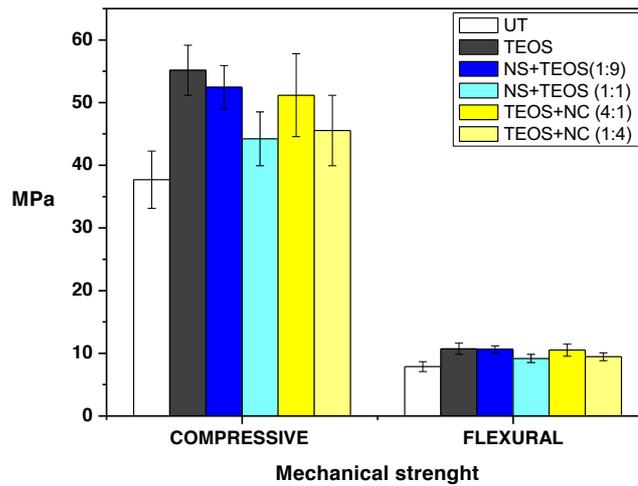
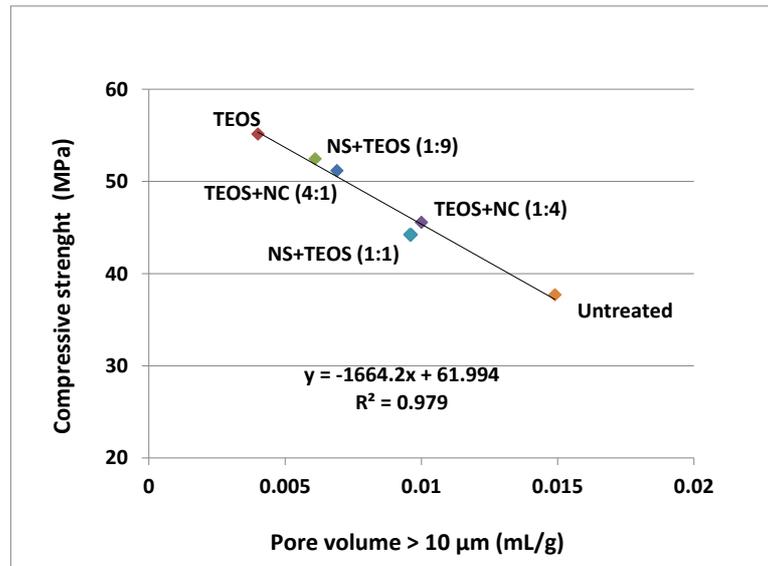


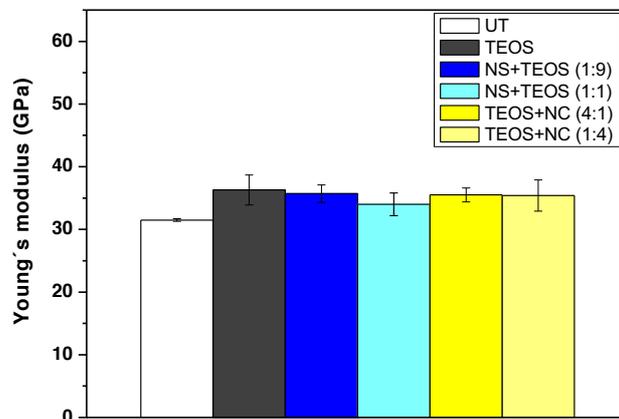
Fig. 4. Pre- and post-treatment compressive and flexural strength in cement mortar.

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544 Fig. 5. Linear correlation between compressive strength and entrained air pore volume in untreated and  
545 consolidant-treated mortar.

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562 Fig. 6. Young's modulus in pre- and post-treated cement mortar specimens.

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Table 1. XRF-based chemical analysis for cement CEM I 42.5N/SR.

(%)	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	Fe <sub>2</sub> O <sub>3</sub>	MnO	MgO	CaO	SO <sub>3</sub>	K <sub>2</sub> O	TiO <sub>2</sub>	LI*
Cement	19.01	4.06	4.73	0.03	1.64	62.4	3.19	0.66	0.23	3.02

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LI\*: loss on ignition at 1000 °C

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Table 2. Main physical-chemical characteristics of treatment products (manufacturers' information).

	ESTEL1000® (CTS)	NANORESTORE ® (CTS)	MEYCO MS 685® (BASF)*	
Acronym	TEOS	NC	NS	586
Composition	Tetraethyl orthosilicate	Calcium hydroxide nanoparticles	Silicon nanoparticles	587 588
Solvent	White spirit D40	Denaturalised isopropyl alcohol	Water	589
Density (g/cm <sup>3</sup> ) (20°C)	0.97	0.8	1.134 ±0.03	590 591 592
Viscosity (kg/(m s))	0.0049 (20°C)	0.00275 (25°C)	< 0.03 (20 °C)	593 594
Active principle wt%	75	0.5	40	595
Particle dimensions	-	< 100 nm	98.7 nm (mean)**	596 597
pH	-	7.2	10 ±1	598 599
Liquid characteristics	Colourless	Opaque white	Opaque white	600 601

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\* Presently registered as MASTERROC® MS 685

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607 Table 3. Amount of product consumed by the mortars.  
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Treatment	Applied together (cm <sup>3</sup> )		NS (1 <sup>st</sup> product) (cm <sup>3</sup> )		TEOS (2 <sup>nd</sup> product) (cm <sup>3</sup> )	
	a	b	a	b	a	b
TEOS	0.65	4.5	-	-	-	-
TEOS + NC (4:1)	0.49	3.37	-	-	-	-
TEOS + NC (1:4)	0.16	1.12	-	-	-	-
NS+ TEOS (1:9)	-	-	0.07	0.5	0.6	4.5
NS+ TEOS (1:1)	-	-	0.2	1.5	0.2	1.5

609 a= 10x10x60 mm; b= 60x70x10 mm  
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617 Table 4. Total porosity and mean pore size of treated and untreated cement mortars.  
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	Total porosity (vol. %)	Mean pore Ø (µm)
Untreated	12.85	0.0769
TEOS	9.47	0.0558
TEOS+NC (4:1)	8.80	0.0575
TEOS+NC (1:4)	11.13	0.0583
NS+TEOS (1:9)	10.27	0.0934
NS+TEOS (1:1)	11.14	0.0743

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637 Table 5. Water vapour conductivity coefficients ( $\delta$ ) in untreated and consolidant-treated cement  
 638 mortars.

	$\delta$ (Kg/m.s.Pa)	% Reduction
Untreated	1.90E <sup>-12</sup>	-
TEOS	0.87E <sup>-12</sup>	54.2
TEOS+NC (4:1)	0.70E <sup>-12</sup>	63.1
TEOS+NC (1:4)	0.77E <sup>-12</sup>	59.4
NS+TEOS (1:9)	0.84E <sup>-12</sup>	55.9
NS+TEOS (1:1)	0.95E <sup>-12</sup>	50.0

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643 Table 6. Low pressure water absorption in untreated and consolidant-treated cement mortar  
 644 specimens.

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	Water absorption after 60 min (kg/m <sup>2</sup> ·h)	Absorption (cm <sup>3</sup> ) from 5 min to 15 min
Untreated	1.98	0.1
TEOS+NC (4:1)	0.35	0.03
TEOS+NC (1:4)	0.75	0.08
NS +TEOS (1:9)	0.33	0.02
NS+TEOS (1:1)	0.42	0.04

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673 Table 7. Variations in lightness ( $L^*$ ), chromatic coordinates  $a^*$  and  $b^*$ , total colour ( $\Delta E^*$ ) and  
 674 gloss ( $\Delta G$ ).  
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	$L^*$	$a^*$	$b^*$		$G$
Untreated	73.75	0.40	5.65	-	0.17
	$\Delta L^*$	$\Delta a^*$	$\Delta b^*$	$\Delta E^*$	$\Delta G$
TEOS	-4.87	0.10	1.13	5.03 $\pm$ 1.01	0.13 $\pm$ 0.05
TEOS+NC (4:1)	-3.14	-0.34	0.22	3.35 $\pm$ 0.20	0.23 $\pm$ 0.1
TEOS+NC (1:4)	0.82	-0.07	-2.03	2.32 $\pm$ 1.03	0.08 $\pm$ 0
NS+TEOS (1:9)	-10.38	0.20	1.57	10.50 $\pm$ 0.29	2.48 $\pm$ 0.35
NS+TEOS (1:1)	-0.26	-0.04	-1.31	3.07 $\pm$ 0.16	3.93 $\pm$ 0.8