Plasmonic Response of Metallic Nanojunctions Driven by Single Atom Motion: Quantum Transport Revealed in Optics

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Abstract

The correlation between transport properties across sub-nanometric metallic gaps and the optical response of the system is a complex effect which is determined by the fine atomic-scale details of the junction structure. As experimental advances are progressively accessing transport and optical characterization of smaller nanojunctions, a clear connection between the structural, electronic and optical properties in these nanocavities is needed. Using *ab initio* calculations, we present here a study of the simultaneous evolution of the structure and the optical response of a plasmonic junction as the particles forming the cavity, two Na₃₈₀ clusters, approach and retract. Atomic reorganizations are responsible for a large hysteresis of the plasmonic response of the system, that shows a jump-to-contact instability during the approach process and the

formation of an atom-sized neck across the junction during retraction. Our calculations demonstrate that, due to the quantization of the conductance in metal nanocontacts, atomic-scale reconfigurations play a crucial role in determining the optical response of the whole system. We observe abrupt changes in the intensities and spectral positions of the dominating plasmon resonances, and find a one-to-one correspondence between these jumps and those of the quantized transport as the neck cross-section diminishes. These results reveal an important connection between transport and optics at the atomic scale, which is at the frontier of current optoelectronics and can drive new options in optical engineering of signals driven by the motion and manipulation of single atoms.

Keywords

Plasmonics, quantum transport, nanocontacts, optoelectronics, optical response, TDDFT calculations

Understanding and controlling the optical response of nanosystems is a key aspect in the search for nanoarchitectures that combine the current performance of miniaturized silicon-based electronics with the fast control and engineering of optical signal. A solution that combines the advantages of one and another inevitably needs to account for the difference in frequency and size ranges that characterize the electronic and optical regimes. One possibility to close the gap between electronics and photonics is provided by the ability of light to coherently excite collective electronic excitations at the surface of metallic nanostructures,¹ effectively bringing light to the nanometer-scale. These excitations, commonly referred to as surface plasmons,^{2,3} have the ability to localize and enhance light in the proximity of the surface of nanostructures,⁴⁻⁷ thus establishing metal nanoparticles as relevant building blocks in current Nanophotonics.^{8,9} Surface plasmons are central for the development and the high optimization level of many techniques and processes such as field-enhanced spectroscopy,^{5,10-15} thermotherapy,¹⁶ photovoltaics,^{17,18} optical sensing,¹⁹⁻²¹ optical nanoengineering,²² opto-

electronics^{23,24} or near-field microscopy.^{5,25–27}

In this context, one option in the search for a proper electro-optical interface relies in pushing the limits of nanometric plasmonics down to the realm of the atomic-scale. It has recently been shown that the interaction of metal surfaces in sub-nanometric proximity drives new optoelectronic phenomena, where an interplay between the photons, single electron transitions, plasmons, vibrations and motion of atoms present in the junction, determines the complex outcome of the optical response including strong quantum effects and nonlinearities.^{28,29} On the one hand, strong non-local dynamical screening³⁰ and quantum tunnelling^{31–34} have been shown to drastically modify the optical response in a metallic sub-nanometric gap, establishing the limit of localization and enhancement of the optical fields far below the predictions from simple classical approaches.^{6,31,35–39} On the other hand, even if typical surface plasmon excitations localize in the nanometer scale, recent *ab initio* calculations considering realistic nanoparticle structures have shown that the fine atomistic details of the crystallographic facets and vertices of the metal particle, with the presence of single atomic protrusions and edges, introduce further non-resonant light localization.^{6,40} This is analogue to the macroscopic lightning rod effect,^{41,42} but brought down to the atomic scale.

It is thus timely to carry out a deep exploration on how the optical response of plasmonic cavities simultaneously correlates with their structural and transport properties,²⁴ going beyond the macroscopic description⁴³ to the atomic-scale regime where the effects of quantum transport at the interface must be taken into account.^{44–46} Furthermore, when two metallic surfaces are approached and put into contact, the formation of small metal necks or nanojunctions connecting them is a very likely process⁴⁷ that has been theoretically predicted^{47,48} and experimentally observed.^{47,49–51} These structures are at the root of friction phenomena in metal surfaces⁵² and give rise to quantized transport following discontinuous changes in the contact cross-section.^{47,53–55} Thus, the key question that we want to address in this manuscript is whether a slight modification of the geometry of the cavity, involving the movement of a few or even a single atom in such sub-nanometric junction, e.g., due to

migration or repositioning driven by strain accumulation in a metal neck, and its corresponding change in conductance are clearly reflected in the optical response. If this were the case, one could expect to observe a discontinuous change in the plasmonic response of the system accompanying each plastic deformation event. As shown below, our simulations indeed confirm that such expectations are fulfilled. This connection between quantized transport and the evolution of the plasmonic response has also been pointed out in recent report⁵⁶ that considers the optical response of a small Na₂₆₁ nanorod as it is stretched.

The present study is particularly relevant in the light of recent progress in fabrication and processing techniques. As the dimensions of nanoscale architectures are progressively reduced, we are facing a regime where the actual distribution of the atoms in a system matters.⁵⁷ The fact that Optics might follow the atoms is of utmost importance in optical engineering and optoelectronics, targeting optical modulators or electro-active control of optical signals, where instabilities and modifications of the performance can be attributed to atomic-scale features.⁵⁸

In order to address the complex correlation of electronics and optics in sub-nanometric junctions where the atoms in the system are allowed to adapt to the mechanical boundary conditions, we performed atomistic quantum mechanical calculations of the electronic structure, the optical response, and the structural evolution of a plasmonic cavity. In our calculations we employed an efficient implementation⁵⁹ of linear response Time-Dependent Density Functional Theory (TDDFT) in conjunction with the SIESTA Density Functional Theory (DFT) package.⁶⁰ The plasmonic cavity in our simulation is formed by two large sodium clusters, containing 380 atoms each of them, in close proximity. This is a canonical example of a metallic system whose properties can be quite straightforwardly extrapolated, with care, to other metallic systems. Further details of the calculations can be found in the Supporting Information. The use of sodium allows performing larger calculations, in terms of the number of atoms, as compared to other more technologically relevant materials like, e.g., gold. This increases the relevance of our results, since the number of atoms involved in

the structural reorganizations of the neck is indeed a small percentage of those contained in the system.

Our approach consists in tracing the energetics, the geometry and the optical response of the two metallic clusters which gradually get closer while allowing atoms to rearrange. Remarkably, at a given separation distance, the clusters show a jump-to-contact instability⁶¹ leading to a strong modification of the optical response, a result in striking contrast to the smooth evolutions found in the context of previous classical and quantum descriptions based on static geometries.^{6,31,32,38,62} The subsequent process of retraction of the clusters is particularly interesting. In such a situation, consistently with previous studies,^{63–65} a metallic atom-sized contact is formed and the conductance across the gap gets quantized. This allows revealing the strong correlation between the transport and optical properties of the system and how, paradoxically, the motion of a few atoms or even a single atom in a nanometric gap can drive a quantized-like (abrupt, discontinuous) evolution of the optical response.

Atomic rearrangements in the plasmonic junction: nanoparticles approach and retraction

Our model of the plasmonic cavity is formed by two sodium clusters that are progressively approached and retracted from each other. Structural relaxations, using forces obtained from DFT calculations, are performed for each approaching and retraction step. Each cluster forming the dimer has an icosahedral shape and contains 380 sodium atoms (see the Supporting Information). The lateral dimension of each cluster is ~ 25 Å. The initial configuration consists of the two clusters placed at a distance of 16 Å, a distance large enough to avoid direct interaction between the nanoparticles. The cavity is initially symmetric with the two clusters opposing planar facets. Starting from that distance the two clusters are slowly brought together. We monitor the inter-particle distance using the *nominal gap size*, defined

as the distance between the two cluster inner facets if the system would remain unrelaxed. Thus, a nominal size gap of zero value would correspond to the superposition of the atoms forming the two opposing facets in the absence of relaxation. Deformation steps of 0.2 Å were chosen as a compromise between computational convenience and a sufficiently small deformation in each step, so that we always keep the system reasonably close to a local minimum configuration. Importantly, this deformation is distributed uniformly along the whole structure, translating into a rather small ($< 0.5\%$) modification of the bond lengths. The system is then allowed to relax so that the excess of energy created by the deformation can be dissipated and the system evolves into the closest local energy minimum. This procedure is similar to that followed in other *ab initio* DFT studies of the formation and evolution of metal junctions and nanocontacts,^{64,65} and mimics a situation in which the system is slowly deformed at low temperature. The heat transport away from the (nanosized) contact region is supposed to be sufficiently efficient to keep the cavity cold in spite of the work being done in the system. This is a reasonable assumption for instance in STM cavities at cryogenic temperatures. As described in detail in the Supporting Information, in order to control the clusters distance and to mimic the presence of tips or surfaces the clusters are attached to,³³ the atoms belonging to the outer facet of each cluster are kept fixed in their initial positions, i.e, they are not relaxed but move rigidly. Once the distance between clusters corresponds to the typical inter-layer distance in bulk sodium, a process of retraction is started by pulling the clusters apart, again in steps of 0.2 Å following a procedure similar to that described above, until they completely separate.

The total energy of the system during this process of approaching and retracting is shown as a function of the nominal gap size in Fig. 1. The evolution of the clusters and junction geometry is shown in the panels of Fig. 1 for selected separation distances. The latin letters that label each panel relate the geometry of the system to the corresponding nominal gap size and energy, as indicated in the curves of the figure. During the approach (red circles) the total energy remains constant until a nominal gap size of about 7.5 Å is reached. From this

Figure 1: Total energy of a plasmonic cavity formed by two Na_{380} clusters as a function of the nominal distance between them (*nominal distance* is defined as the distance among the facets defining the cavity if the system would not be relaxed). Red circles represent the approaching process while blue circles indicate the retracting process. The inner panel shows the actual gap size as a function of the nominal value during the approach, showing the clear signature of a jump-to-contact event. Images of the geometries around the graph show the rearrangement of the atoms in the clusters and the formation of a nanojunction during retraction. Latin letters indicate the correspondence between some of the total energy points in the graph and the configurations shown in the surrounding panels.

separation, the total energy starts decreasing smoothly. At a nominal gap size of ~ 6.1 Å the two clusters suddenly jump to contact.⁶¹ At this point the actual interface distance abruptly decreases and the energy is substantially reduced. The clusters are now connected and elongated along the intermolecular axis. The inset in Fig. 1 shows the actual gap size indicating the real distance between the inner facets of the clusters. After the jump-to-contact, the value of the actual gap size drops to a value of about 3.2 Å remaining fairly constant and close to 3.0 Å as the two clusters get closer together. The abrupt reduction of

the energy at the jump-to-contact point is mostly due to the reduction of the surface energy of both clusters (two facets disappear). However, this happens at the expense of a large elastic deformation of the clusters. By further approaching the two clusters we reduce the elastic deformation of the system and, correspondingly, the total energy decreases. Eventually the system suffers some reorganizations which are also reflected (although they are somewhat less obvious than the jump-to-contact) in the energy versus distance curve in Fig. 1. For example, the stacking of the atomic layers at the interface, initially imposed by the mirror symmetry of our starting geometry, gets optimized at a nominal gap size of approximately 2.5 Å. Later, the particles start to deform to try to reduce further their surface area by increasing the contact cross-section. The energy finally stabilizes and starts to slowly increase for nominal gap sizes below ~ 1 Å. We stop our approaching process at this point.

Once the two clusters are clumped together at a nominal distance comparable to the interlayer distance in bulk sodium, we start pulling apart them (blue circles in Fig. 1). During the retraction process the whole structure evolves creating and thinning a neck that connects the two clusters until a monatomic chain is formed and, eventually, until a complete separation of the clusters is achieved (point l in Fig. 1). In agreement with previous studies, the evolution of the contact structure takes place via an alternation of elastic and plastic deformation events.^{47,51,55,64–66} The contact is elongated until the accumulated elastic energy is sufficient to produce atomic rearrangements, mainly driven by the atoms in the neck area. During these plastic events the energy of the system decreases abruptly. Thus, there is a one-to-one correspondence between the discontinuities of total energy in Fig. 1 and the changes in the configuration of the metal neck. It is striking to note the dramatic contrast between the distance at which the jump-to-contact takes place and the clusters “touch” for the first time during the approaching process (close to point a in Fig. 1), and the distance at which they finally detach (indicated by a vertical green dashed line). A nominal gap distance of 32.3 Å is needed to separate completely the clusters.

In summary we have seen that the geometry of the system strongly departs from the

idealized situation in which two clusters simply change their relative position. The role of such atomic-scale structural rearrangements on the optical response of plasmonic cavities has been scarcely considered in the literature.⁵⁶ However, as we describe in the following, atomic-scale structural modifications can play a key role in determining the optical response of the cavity.

Optical response of a forming plasmonic cavity: jump-to-contact instability

To explore the connection between the structural evolution and the plasmonic response of the cavity, we calculate the optical absorption of a plasmonic junction as the clusters approach for two situations: first, no relaxation of the clusters is allowed and the only parameter modified is the distance between the particles [Fig. 2 (a)]; second, the relaxation of the atoms is taken into account [Fig. 2 (b)] following the atomic-scale restructuring shown in Fig. 1 (red symbols). The resonant plasmonic modes of the forming cavity, as obtained from the calculated polarizability of the system, are displayed as a function of the inter-particle distance in both situations. The component of the polarizability parallel to the dimer axis is considered here, i.e., induced by an electrical field polarized along the same axis, that we take hereafter as z . In both cases, and depending on the separation, we can identify three distinct resonances.^{6,31–33,38,39,43} A single intense so-called Bonding Dipolar Plasmonic (BDP) resonance around 3 eV dominates the response at large inter-cluster distances when the two clusters interact weakly. The BDP shows an induced charge distribution characterized by a capacitive coupling of charges of opposite sign at both sides of the cavity, as schematically depicted in the right drawing of the top panel in Fig. 2. When both clusters are in contact, so that free charges can efficiently move across the junction, we enter a conductive coupling regime characterized by the so-called Charge Transfer Plasmon (CTP)⁴³ and the associated high-energy Charge Transfer Plasmon (CTP') modes. The conducting link of the CTP

Figure 2: Evolution of the imaginary part of the polarizability of a Na_{380} dimer (external field applied along the dimer axis) as the clusters are approached, plotted as a function of the separation distance and photon energy. Both unrelaxed (a) and relaxed (b) geometries of the cluster dimer are considered. The dark dots indicate the position of the peak maxima in the polarizability for those distances for which the optical response has been computed. The arrow lines indicate the *contact point* for the two cases, i.e., the distance at which the clusters merge into one single larger object. The top panel shows the schematic representation of the induced charge in the modes that dominate the optical response before contact (bonding dipolar plasmon mode, BDP) and after contact (charge transfer plasmon mode, CTP). Panels c) and d) show the imaginary part of the induced density and the corresponding modulus of the electron current flowing through each cross sectional [i.e., (x, y)] plane along the dimer axis (see the Supporting Information for more details). An external electric field of magnitude of 1×10^{-9} atomic units is assumed here with a polarization parallel to the junction main axis. The nominal gap size is $6.1 \text{ \AA}$, corresponding to the jump-to-contact configuration in the relaxed case.

through the junction of the clusters, produces a screening of the charges in the cavity, and thus redistributes the induced charge density to produce a net dipole that extends to the whole dimer structure, as depicted in the top-left scheme in Fig. 2 [see also the panel c) and d)].

The BDP resonance is redshifted as the inter-cluster distance is reduced and the Coulomb interactions among the clusters increase. This shift is due to the strong interaction of the parallel induced dipoles along the dimer axis, which hybridize⁶⁷ lowering the energy of the resulting optically active mode. In this capacitive (weak interaction) regime both unrelaxed and relaxed cases show the same dependence on the inter-particle distance. The BDP mode lives until the clusters are brought to a distance of about 6.1 Å. At this point, for the unrelaxed dimer [see Fig. 2 (a)], the BDP mode is quenched and higher energy modes start gaining intensity. If the clusters are approached further we observe a smooth transition from the capacitive to the conductive coupling regime. For separation distances right below 6 Å the electron tunnelling current at relevant frequencies gradually starts flowing, giving rise to the progressive emergence of the CTP resonance.^{6,29,32,39} This transition region is frequently referred to as the quantum tunnelling regime of plasmonic cavities.³³ At a distance comparable to the sodium interlayer distance, ~ 3.0 Å, the clusters become chemically bonded and a clear contact is established. Under those conditions, a substantial current can be established and the CTP appears fully developed.

In contrast, the situation shown in Fig. 2(b) for the relaxed dimer is strikingly different. The relaxed dimer undergoes a jump-to-contact instability (see red curve in Fig. 1) that dramatically modifies the evolution of the optical spectrum. The transition regime, found between 6 Å and 3 Å for the unrelaxed dimer, has almost completely disappeared in the relaxed case. There are not stable geometries for those intermediate gap sizes and, thus, the resistive tunnelling (transition) regime cannot be clearly identified in the optical response in this case. Although the details of the jump-to-contact process strongly depend on the size and shape of the facets and the effective elastic constants of the systems being brought

into contact, this effect is a quite general behavior which is routinely taken into account in the interpretation of data from scanning probe microscopies. Our results indicate that the effect of the jump-to-contact must be considered when exploring and interpreting the optical response of metallic particles in close proximity, particularly when large atomic-scale reconfigurations can be expected. Importantly, this phenomenon can hinder the appearance of a smooth transition between the capacitive and charge-transfer regimes in the optical response of plasmonic cavities.

In panels (c) and (d) of Fig. 2 we explore the real space distribution of the induced charge for the CTP and CTP' modes right after the clusters get into contact, i.e., right after the jump-to-contact instability. Here we plot the imaginary part of the induced density at the resonant frequencies. The isosurfaces in these plots, and those shown below, correspond to $\sim 10\%$ of the maximum value of the induced density. We also plot the corresponding electron current (graph to the right of each charge density plot) flowing through (x, y) planes (i.e., perpendicular to the dimer axis) as function of z , the coordinate along the dimer axis. Details of the method used to calculate the current can be found in the Supporting Information. The density charge associated with the CTP forms a dipolar pattern over the whole system having a single node placed at the center of the system. Thus, the charge accumulation does not take place in the cavity interfaces, but rather extends to the whole system. Correspondingly, the current associated with the CTP resonance has its maximum at the gap center. On the other hand, the CTP' mode presents two dipolar patterns on each cluster with nodes of the induced density charge in the center of the system as well as in the middle of each cluster. The charge distribution in this case is somewhat similar to what one can expect for the BDP mode. However, the current reveals a key piece of information to rule out this interpretation. In the case of the CTP' resonance the maxima of the current are found both in the center of the system as well as within each cluster. This is indeed confirming that there is charge transfer among both clusters also in this high energy mode. Thus, the observed induced density pattern is better interpreted as the second optically active mode of a metal rod.

Finally, below 2 Å nominal gap size the conductive coupling regime of the junction is fully developed in both the unrelaxed and relaxed cases, with the CTP and the CTP' resonances converging to similar values of energy, around 2.25 eV and 3.3 eV, respectively. This underlines the fact that the details of atom rearrangements at the cluster interface might not be so important in the determination of the optical response once the two clusters are fully chemically bonded.

Optical response of a retracting plasmonic junction: Optics driven by individual atoms

Figure 3: a) Evolution of the resonances in the polarizability of a plasmonic junction as a function of the nominal gap size and energy, as the clusters forming the junction move apart (i.e., move towards larger nominal gap sizes). The dots indicate the positions of the peak maxima in the polarizability for the considered configurations. Panels in b) show the spectral lines of the polarizability at distances before and after each of the jumps highlighted in panel a) by means of Greek letters and colored arrows and dots. The colors of the lines of the spectra correspond to the colors of those arrows and dots, thus indicating whether a given spectrum corresponds to a configuration before or after the jump. c) Imaginary part of the induced charge density distribution around the junction for the three selected distances in a) and b), before (top) and after (bottom) the jumps.

Fig. 3 shows the optical polarizability of the junction during the retraction process.

Surprisingly, as the two clusters are retracted, the CTP and CTP' modes dominate the spectrum for most separation distances, all the way to nominal inter-particle distances of several tens of Angstroms. This is in clear contrast with the results obtained in the previous section (approaching situation), and it is a result of the structural evolution of the junction, characterized by the formation of a thin conducting neck among the clusters [as shown in the panels (c)-(i) of Fig. 1]. As the clusters get separated, the neck gets longer and thinner. As a consequence, the charge transfer modes disperse towards lower energies (due to the overall elongation of the system). Moreover, as the neck cross-section is reduced, the intensity of the CTP' mode increases at the expense of the lower-energy CTP resonance, consistent with calculations of stretched clusters.⁵⁶ As the current flowing across the neck diminishes, the CTP' mode converges towards the BDP mode while the CTP mode tends to disappear.

While the polarizability of the approaching situation in Fig. 2 only shows a clear discontinuity associated with the jump-to-contact instability of the cavity, the retracting situation shows a completely different behavior as a function of the separation distance. During retraction the optical spectrum is characterized by the appearance of many discontinuities both in the spectral position and the intensity of the resonances. A careful inspection of Fig. 1 reveals that these discontinuities happen at exactly the same nominal distances where jumps in the total energy are detected. Some of the most visible jumps are highlighted with arrows of different colors and marked with Greek letters in Fig. 3 (a), and the corresponding polarizability is plotted in detail on the panels of Fig. 3 (b), which extracts the spectral lines from the contour plot in (a). Each panel shows spectra corresponding to distances before and after one of the jumps, identified in the contour plot of Fig. 3 (a) with the corresponding colored dots and arrows. Consecutive curves correspond to configurations in which the inter-particle distance is changed by $0.2 \text{ \AA}$. In each panel there are several, almost indistinguishable, spectra of the same color. This highlights that noticeable changes in the spectrum are indeed linked to the plastic deformation events in the neck, and not to the small rearrangements during elastic deformation. At each jump we observe clear changes in the

intensities, widths and positions of the resonance peaks. The jumps affect primarily the low energy resonance, CTP, although they are also visible in the CTP' mode. They are owing to the atomic reorganization in the neck region and they are specially visible for distances above ~ 20 Å due to the small cross-section of the neck. Remarkably, for such thin necks even single atom movements produce visible changes in the optical response of the system, clearly associated with the quantized nature of the conductance through the junction neck. The jump at 29.3 Å indicates the formation of a well-ordered monatomic neck, i.e., the clusters are connected by a single row of atoms. The formation of such structures has been observed for many metals, for example in the case of Au, for which these monatomic wires have even been visualized by electron microscopy^{47,49–51,63,68}.

The intensity of the CTP resonance suffers an abrupt decrease for a nominal gap size of around 23 Å, becoming broader between 27 and 29.3 Å. Afterwards, simultaneously to the monatomic neck formation, the CTP resonance gets sharper with a consequent regain in intensity. This evolution is due to a combination of several effects, the most important being the quantization of electron transport in the metal neck. Such quantization is a well-known effect due to the small cross-section of the contact, comparable to the electron wavelength.⁶⁹ As a result of the lateral confinement, the electronic energy levels in a thin metal nanowire or neck get quantized and, at a given energy, only a discrete number of bands (or “channels” using the standard terminology in quantum transport) can contribute to the electron transport. Thus, under a small, static bias, if the electron injection from the electrodes (in our case the clusters) is efficient and the neck structure is sufficiently long and ordered, we can expect each channel at the Fermi level to contribute to transport with a quantum of conductance $G_0 = 2e^2/h$,⁶⁹ with h the Planck’s constant. In the presence of defects or strong scattering in the connections to the electrodes, the transmission probability of the channels gets reduced.⁶⁹

With these ideas at hand, we can easily explain the observed behaviors. The abrupt jump in the intensity of the CTP peak around 23 Å (α jump) is caused by the sudden reduction

of the neck's cross-section, as can be clearly seen in Fig. 3 (c) and the inset of Fig. 4. As expected, the reduction of the cross-section reduces the number of conduction channels and, therefore, the electric current flowing through the junction (this is confirmed in Fig. 4, discussed later in detail). The resonance peak also shifts to slightly lower energies. The origin of the intensity jump at $\sim 27 \text{ \AA}$ (β jump) is also similar: a cross-section reduction that translates onto a sudden decrease of the current as can be seen in Figures 3 (c) and 4. After this jump at $27 \text{ \AA}$, the neck develops into a less ordered structure, creating a region of high scattering that hampers the electron transport between the clusters. As a consequence the CTP resonance broadens. Finally, once the relatively defect-free monatomic wire is formed, the transport through the neck becomes completely ballistic, i.e. all the electrons that are injected to the monatomic wire get across the junction, and the peak in the polarizability becomes more defined again.

These quantization effects can also be observed in the shape of the distributions of induced charge density as the neck evolves during retraction. In panels (c) of Fig. 3 the imaginary part of the induced density associated with the CTP mode is plotted for those configurations immediately before and after the α , β and γ jumps (indicated by the colored arrows in the polarizability plot). Although the density change has a quite complex distribution, it is possible to follow the evolution of the patterns towards simpler schemes of charge oscillation after each jump. The induced density presents a complex distribution and nodal structure, with a decreasing number of nodes as the cross-section of the neck gets thinner, a fact that reflects the larger number of open conduction channels for the wider structures. Subtle changes in the structure that have a direct translation in the optical response can also be observed in these density plots. For example, in the case of the γ jump, the three-atoms-long monatomic wire becomes more straight and the connections to the cluster more symmetric. This slightly increases the current flowing through the structure and produces the aforementioned changes in the plasmonic response.

Once the two clusters totally separate, breaking the neck, two split resonances arise

near the CTP' resonance at about 2.6 and 3.2 eV. The initial face-to-face configuration has been substituted by an asymmetric tip-to-tip configuration (see panel *l* in Figure 1). The lower energy resonance recalls a BDP mode, with the largest charge accumulations around the central gap (see the Supporting Information). The higher energy resonance has a more complex charge distribution, corresponding to higher order mode, showing charge accumulations both in the tips inside the cavity and in the facets of the clusters (see the Supporting Information).

Figure 4: Modulus of the current flowing through a plasmonic junction as a function of the separation of the clusters forming the junction. The current is evaluated at a cross-sectional plane passing through the center of the cavity (and cutting the center of the neck when present). Colored arrows indicate the direction of the process (approaching or retracting). The current is computed at the resonance frequency of the different modes of the cavity, as indicated by the labels and described in the text. Black arrows indicate the position at which the spectral jumps in Figure 3 occur. The inset shows the one-to-one correspondence between the jumps in the current for the CTP mode and the cross-section of the metal neck. An external electric field of 1×10^{-9} atomic units is assumed with a polarization parallel to the junction main axis.

To fully account for the connection between high-frequency electron transport and optical response of the plasmonic junction, we have calculated the current through the junction as a function of the nominal gap size. In Figure 4 the results for the modulus of the current passing through a plane cutting the center of the junction are shown. Here we present the

current computed at the frequencies of the main resonances of the polarizability, described in previous sections.

The current during the approach process is shown by red solid circles, corresponding first to the BDP mode, and later to the CTP mode, once the clusters are in contact. The current for the BDP mode is negligible until the jump-to-contact event takes place. Once the clusters are connected the current can flow through the whole system and therefore its value increases dramatically. The current calculated for the CTP resonance increases almost linearly as we decrease the nominal gap size.

The values of the current across the junction at the energies of the CTP and CTP' resonances, while retracting the clusters and the neck is getting thinner, are plotted respectively in blue and yellow. The current related to the CTP resonance decreases monotonously as we elongate the system. As commented above, its evolution is characterized by abrupt jumps whenever the neck suffers a plastic deformation. The current eventually reaches a plateau associated with the formation of a well-defined monatomic neck. Interestingly, once the monatomic neck is formed, a further neck stretching does not affect considerably the current. This can be expected since the conductance of such small necks mostly depends on the cross-section, which is fixed for the monatomic wire. It is interesting to mention that for the low energy CTP mode, it was possible to define the conductance of the monatomic neck as the ratio of the current and the voltage drop across the wire. The modulus of the optical conductance (which is a complex number now) computed in this way at the CTP frequency is $\sim 0.65 G_0$, close to the expected value for a monatomic sodium wire at low frequency, G_0 . In this case it was possible to define the conductance without ambiguity since the potential drop is confined to the gap region, with the electrostatic potential flat inside the clusters (see the Supporting Information). For larger cross-sections of the necks and/or higher energies of the mode, this condition is not fulfilled and the definition of the conductance is quite involved. The current associated with the CTP' mode follows the same trends than that of the CTP, although shows a less pronounced dependence on the overall elongation of the

system. Obviously, once the clusters separate the current becomes negligible.

The arrows in Figure 4 indicate the position of the jumps shown in Figure 3 (a). Except for the last jump at 29.3 Å, the other two jumps observed in the polarizability (Fig. 3) and in the total energy (Fig. 1) show a clearly correlated sudden change in the current. This points towards a remarkable effect of a few atoms (or even a single atom), whose motion influences the overall optical response of the whole system (containing 760 atoms in our case). This observation can be of utmost importance in the control and manipulation of optical signal in subnanometric junctions which are clearly affected by this type of physical processes at the atomic scale.

To establish a more direct connection between the computed current as a function of the gap separation within the junction and the well-known quantization of transport in metal nanocontacts, in the inset panel of Fig. 4 we show the current for the CTP while retracting (left axis of the graph) plotted in relationship with the neck cross-section (right axis of the graph). As can be seen, there is an almost perfect correlation between the changes in the current and the evolution of the neck cross-section (see the Supporting Information for the procedure adopted here to calculate the neck cross-section). Such correlation has been already well-established in the case of low-frequency driving-fields being applied to the necks. It has been observed in the formation of metal nanocontacts in Scanning Tunnelling Microscopy (STM) and break-junctions experiments, and corroborated by many calculations.⁴⁷ Our Fig. 4 goes one step beyond, establishing such correlation at optical frequencies. With this additional piece of information, we can now summarize the results described in this section by unambiguously establishing the following cause-effect relationships: plastic deformation of the neck during elongation $\rightarrow$ cross-section reduction $\rightarrow$ abrupt drop of the current $\rightarrow$ decrease of the intensity of the CTP mode. This process is sometimes accompanied by small shifts of the position of the resonance peak. Finally, as mentioned above, more disordered structures translate into broader and dimmer CTP resonances.

Conclusions

In summary, we have shown how atomic-scale structural reorganizations are crucial to determine the optical properties of plasmonic cavities. Besides the importance of jump-to-contact events, that can almost completely eliminate any signature of the plasmonic tunnelling regime, the effects are particularly dramatic when a metal nanocontact is formed across the cavity. This is due to the strong dependence of the plasmonic response of the system on the quantized current flowing through the connecting neck.

The mechanical response of atom-sized necks is characterized by sudden rearrangements of the atomic structure, which frequently involve just a few atoms in the thinner part of the contact. Since the electron transport through thin metal nanocontacts is quantized, the corresponding changes of the current flowing across the junction are necessarily discontinuous. Our calculations demonstrate that this common observation under small applied DC biases can be extrapolated to the optical frequencies of the plasmon resonances of the cavity, at least for the short ballistic contacts considered here. These jumps in the current translate onto abrupt changes in the plasmonic response of the system. Thus, the discontinuous evolution of the spectral position, width and intensity of the CTP mode observed in our simulations is a direct consequence of the transport quantization in the connecting neck. A similar connection between quantized transport and the evolution of the plasmonic response has also been pointed out recently.⁵⁶ In spite of the smaller scale of the systems treated, discontinuities in the computed DC transport across the junction are related to the intensities of the plasmonic peaks. In our work, the discontinuity of AC quantum transport is revealed in the optical fingerprint of both a snap-to-contact situation in a forming plasmonic junction, as well as during retraction in a nanojunction connected by atom-sized metal contacts.

The correlation is clearly demonstrated, showing that remarkably, Optics follows the atoms. This is absolutely important in the design of subnanometric-scale optical modulators that rely in slight changes of the optical response against tiny configurational modifications. In our case we have analyzed relatively small icosahedral sodium clusters, however, we expect

to find a similar behavior for other materials suitable for electronic applications, such as gold.

The effect of a single atom in the optical properties of a nanoscopic object as the one reported here, which can be probably extended to somewhat larger objects, has important consequences in optical engineering, molecular electronics, and photochemistry, where the optical response can now be tailored by a few atoms.

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Supporting Information Available: Supporting Information contains details on the computational methods both for DFT and TDDFT calculations, the calculation of the currents at optical frequencies, the definition of the nanocontact cross-sections and the initial structure of our clusters. The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.photonics.XXXX .

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Graphical TOC Entry

Supporting Information for “Plasmonic Response of Metallic Nanojunctions Driven by Single Atom Motion: Quantum Transport Revealed in Optics”

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Methods: Structural Relaxations

The structural relaxations of the Na₃₈₀ dimers were performed using standard density functional theory (DFT) as implemented in the SIESTA code.^{1,2} We used the Perdew-Burke-Erzenhof density functional (GGA-PBE)³, norm-conserving pseudopotentials⁴ to effectively describe core electrons, and a double- ζ polarized (DZP) basis set of numerical atomic orbitals generated using an *energy shift* of 100 meV.⁵ The fineness of the real-space grid used to compute the Hartree and exchange-correlation contributions to the energy and Hamiltonian corresponds to a plane-wave cut-off of 130 Ry,² while a grid of 110 Ry has been used to

produce the plots of the induced density at optical frequencies shown both in the main text and in this Supporting Information. Structural relaxations were stopped when forces acting on all atoms were smaller than $0.02 \text{ eV/\AA}$.

Figure 1: Relaxed initial structure of one of the Na_{380} clusters forming the plasmonic cavity.

Figure 1 shows the relaxed initial structure of each of the Na_{380} clusters that form the plasmonic cavity. The cluster presents an icosahedral structure and has been obtained starting from a configuration optimized with empirical potentials⁶ and available at the Cambridge Cluster Database (CCD).⁷ The relaxed structure obtained using GGA-PBE is very similar to the one provided at the CCD site, and it is characterized by the presence of planar facets, sharp edges and single-atom vertices.

Figure 2 shows a snapshot of the relaxation of the plasmonic cavity when the clusters are approached towards each other. As a mechanical constraint, and mimicking the presence of macroscopic scanning probe tips or surfaces where the clusters are attached, the atoms in the outer surfaces of both clusters (highlighted in light-green in Fig. 2) remain fixed and move rigidly during the approach/retraction events. Each new configuration is generated by changing the distance between these fixed facets by $0.2 \text{ \AA}$. The rest of the atoms, blue atoms in Fig. 2, are then relaxed. Steps of $0.2 \text{ \AA}$ are chosen as a compromise between numerical efficiency and a sufficiently small deformation in each step, so that we always keep the system reasonably close to a local minimum configuration. Importantly, this deformation is distributed uniformly along the whole structure, translating into a rather small ($< 0.5 \%$)

Figure 2: The atoms in the outer facets of both clusters (the atoms in one of these facets have been highlighted in light-green) are kept fixed during the relaxation process.

modification of the bond lengths. The system is then allowed to relax so that the excess of energy created by the deformation can be dissipated and the system evolves into the closest local energy minimum. This procedure is similar to that followed in other *ab initio* DFT studies of the formation and evolution of metal junctions and nanocontacts,^{8,9} and mimics a situation in which the system is slowly deformed at low temperature. The heat transport away from the (nanosized) contact region is supposed to be sufficiently efficient to keep the cavity cold in spite of the work being done in the system. This is a reasonable assumption for instance in STM cavities at cryogenic temperatures.

The presence of two unrelaxed facets allows to measure the displacement applied to the system unambiguously, which in turn permits the definition of a *nominal gap size*, corresponding to the distance between the inner facets of the neighboring clusters in case the system were not allowed to relax. This is the magnitude that is used to define the inter-particle distance in the plots in the main text of our paper.

Methods: Optical Response Calculations

The calculations of the optical response of the plasmonic cavity were performed within the framework of time-dependent density functional theory (TDDFT) using the so-called adiabatic local-density approximation (A-TDLDA), i.e., using the LDA^{10,11} exchange-correlation kernel computed for the instantaneous electron density. TDDFT is a rigorous (in principle) extension of DFT to time dependent systems.¹² We use an efficient implementation of linear response TDDFT that utilizes an iterative scheme to compute the optical response in frequency domain and a local basis to expand the products of Kohn-Sham orbitals.^{13–15} The program uses as an input the information from ground-state electronic-structure calculations performed using localized basis sets (e.g., linear combination of atomic orbitals). In particular, the method has been interfaced with SIESTA and, as exemplified here, allows computing the linear optical response of large clusters and molecules with moderate computational resources.^{16,17} We imposed an intrinsic (phenomenological) broadening $\Delta = 300$ meV, which brings the total width of the calculated plasmonic resonances for the single cluster in reasonable agreement with available experimental data.¹⁸ The frequency axis is sampled with a resolution of 0.02 eV.

Before computing the polarizability, for each relaxed structure, a DFT ground-state calculation has to be performed employing the LDA functional. The resulting Kohn-Sham orbitals and energies are used as an input for the subsequent A-TDLDA calculations. While the use of the GGA functional substantially improves the description of the geometries for sodium clusters (in particular, LDA tends to produce too dense structures,¹⁹ which in turns can give rise to a blueshift of plasmon resonances¹⁷), A-TDLDA has proven to provide accurate results for the optical properties of sodium clusters.^{20–25}

Methods: Calculation of Electric Currents at Optical Frequencies

Figure 3: Representation of the integration volume Ω used to define the current passing through the junction with the help of the continuity equation.

Since we are dealing with a finite object we can use the continuity equation and an integration region like the one shown in Fig. 3 to define the current that flows across a plane perpendicular to dimer axis and passing through the center of the junction. From our TDDFT calculation we obtain the induced electron density $\delta\tilde{n}(\mathbf{r},\omega)$ in response to a monochromatic field with frequency ω . We can now integrate the induced density over the volume in Fig. 3

$$\delta\tilde{Q}(\omega) = \int_{\Omega} d\mathbf{r}^3 \delta\tilde{n}(\mathbf{r},\omega), \quad (1)$$

to obtain $\delta\tilde{Q}(\omega)$ the Fourier transform of the time-dependent total induced charge contained in Ω . The continuity equation provides with the trivial relation between the total induced

charge in Ω , $\delta Q(t)$, and the current flowing across the junction $I(t)$ at each instant of time t . In frequency domain this relation is expressed as

$$\tilde{I}(\omega) = -i\omega\delta\tilde{Q}(\omega). \quad (2)$$

Thus, we can easily obtain the modulus of the current (maximal current) flowing through the junction in response to an external electric field of the form $\mathbf{E} = \mathbf{E}_0\cos(\omega t)$:

$$|I_{max}(\omega)| = \omega\sqrt{\left[\delta\tilde{Q}'(\omega)\right]^2 + \left[\delta\tilde{Q}''(\omega)\right]^2}, \quad (3)$$

with $\delta\tilde{Q}'(\omega)$ and $\delta\tilde{Q}''(\omega)$, respectively, the real and imaginary parts of $\delta\tilde{Q}(\omega)$.

In our calculations we compute the current at the frequencies of the plasmonic resonances for each geometry, and $\delta\tilde{Q}(\omega)$ is calculated by integrating the density change $\delta\tilde{n}(\mathbf{r},\omega)$ over a given real space volume. By changing the integration volume we define the (x,y) plane through which the current is calculated. For example, the current as a function of the position of such plane is shown, for a particular configuration and two different resonance frequencies, in the panels c) and d) in Figure 2 of our paper. It is important to stress that such maps do not refer to a given instant of time, but they rather depict the maximum current passing through each (x,y) plane at any instant of time. In order to compute the current flowing through the center of the junction (as shown of Fig.4 of the main text) we need to integrate the induced charge over a volume similar to that depicted in Figure 3.

Methods: Definition of the Neck Cross-Section

The neck formation during the retracting process opens the question of how to measure the neck cross-section. In particular, it is not obvious how to compare the cross-sections of necks with similar structures. Here we decided to use the distribution of the ground-state electron density $\rho(\mathbf{r})$ as a means to measure the neck cross-section for an arbitrary structure. The

Figure 4: The neck section is calculated by analyzing the electron density in a (x, y) plane cutting the center of the neck. The right panel shows the 2D electron density distribution in that plane (blue dashed line). The red solid line represents the isocontour corresponding to a $0.00169 \text{ e}/\text{\AA}^3$ threshold density. The cross-section is defined as the area of the region limited by such isocontour. We can clearly see that the neck cross-section is formed in this case by three Na atoms.

density is computed in an (x, y) plane passing through the middle of the junction and the neck cross-section A_{cs} is obtained as the area in which the electron density is larger than a given threshold value ρ_{th} (see Fig. 4). In other words, we use the integral

$$A_{cs} = \iint dxdy f(\mathbf{r})$$

$$\text{where} \quad f(\mathbf{r}) = 1 \quad \text{if } |\rho(\mathbf{r})| > \rho_{th}, \quad (4)$$

$$\text{and} \quad f(\mathbf{r}) = 0 \quad \text{if } |\rho(\mathbf{r})| < \rho_{th}.$$

The value of ρ_{th} is arbitrary and was chosen here so that the radius of an isolated Na atom is $2.88 \text{ \AA}$, a reasonable value if we compare to Na bulk density (characterized by Wigner radius $r_s \sim 2.12 \text{ \AA}$) and we take into account the spillage of charge towards vacuum in a finite object. In any case, the specific value of the cross-section assigned to a particular neck structure is irrelevant (as far as *reasonable*), the importance of this method is the ability to continuously monitor the cross-section change as the structure evolves.

Split modes after nanocontact break

Figure 5: Imaginary part of the induced charge density distribution for the frequencies of the two plasmon resonances found after the break of the connecting neck and final separation of the clusters.

When the metal contact connecting the two clusters finally breaks and they separate, their final structures differ considerably from the initial ones. The two clusters are not identical anymore and the mirror symmetry in the junction is lost. Furthermore, rather than in the initial facet-to-facet configuration, the clusters now present some sort of tip-to-tip configuration. With the rupture of the connecting neck, the CTP and CTP' modes that dominated the optical response of the system during the whole retraction process disappear and give way to two modes, rather than a single BDP resonance. The new modes are found at slightly different energies. Moreover, the corresponding induced charge density distributions in Fig. 5 show different patterns. The lower energy resonance at 2.55 eV recalls a BDP

mode, with the largest charge accumulations around the central gap. The higher energy resonance at 3.14 eV has a more complex charge distribution, corresponding to higher order mode, showing charge accumulations both in the tips inside the cavity and in the facets of the clusters.

Real-time Dynamics: CTP Mode Across a Monatomic Chain

Figure 6: Real time evolution of the induced density associated with the CTP mode when the cavity is connected by a short monatomic wire. Black arrows indicate the time direction while green arrows indicate the direction and intensity of the external electric field in each instant. At $t = 0$ and $t = \pi/\omega_{res}$ the distribution is completely derived from the real part $\delta n'(\mathbf{r}, \omega_{res})$, at $t = \pi/2\omega_{res}$ it is given by the imaginary part $\delta n''(\mathbf{r}, \omega_{res})$.

The real space distributions of the induced charge density shown in the main text of the paper only correspond to the imaginary part. Therefore, although they can serve to

characterize the resonant mode at a particular frequency, they may not give a complete picture of the time evolution of the screening charge in the system.

The imaginary part of the polarizability (induced density distribution) corresponds to the optical absorption and provides information about the resonance frequencies (spatial distribution) of the different modes. For a single, sharp excitation the maximum of the peak in the imaginary part of the polarizability coincides with a zero in the real part of the polarizability. Therefore, when at resonance the external field and the induced dipole in the system are always out of phase. However, when dealing with a system having multiple resonances and especially at low frequencies, even at resonance the real part can be different from zero and have a non-negligible intensity as compared to the imaginary part. In these cases, at the resonance frequency the real time evolution of the induced charge is a combination of the contributions coming from the imaginary and real part of the density change,

$$\begin{aligned}\delta n(\mathbf{r}, t; \omega_{res}) = & \quad \delta n'(\mathbf{r}, t; \omega_{res}) \cos(\omega_{res} t) \\ & + \delta n''(\mathbf{r}, t; \omega_{res}) \sin(\omega_{res} t),\end{aligned}\tag{5}$$

being the external electric field $\mathbf{E} = \mathbf{E}_0 \cos(\omega t)$. The real part is in phase with the external field and the imaginary part represents the out-of-phase (resonant) component of the total density change. This is the case for the CTP low energy mode seen in the polarizability once the mono-atomic neck is formed, that shows a large real part of the induced charge density even at resonance.

The sequence of images shown in Fig. 6 describes the evolution in time of the CTP mode across the cavity when there is a monatomic wire connecting the clusters. At $t = \pi/2\omega_{res}$ we find the imaginary part of the induced charge, that characterizes the mode resonant at that frequency. As expected, the dipole pattern extends over the whole system with the presence of a node in the center of the junction. Thus, it corresponds to the expected charge transfer among the clusters. In contrast, the real part, found at $t = 0$ and $t = \pi/\omega_{res}$, presents a

pattern formed by dipoles placed on each cluster that resembles a BDP mode.

From a physical point of view what we see here is quite transparent. The monatomic wire across the junction represents a bottleneck for electron conduction as compared to the facile movement of charges within each of the clusters. As a consequence, electrons can easily move across each of the clusters and react fast to the applied external field. However, when they reach the gap in the center of the system they accumulate there, since the movement of charge across that gap is limited by the monatomic chain, which provides just a single channel for electron conduction.

Conductance at Optical Frequencies

The conductance is defined as the ratio between the current flowing across and the bias applied to a particular structure. The conductance is frequently used to characterize the DC transport properties of nanowires. For wires of atom-sized cross-sections the DC conductance can be quantized, i.e., it appears in multiples of the quantum of conductance G_0 ($G_0=2e^2/h$ with h Planck's constant and e the electron charge). In principle, it is also possible to calculate the neck conductance at optical frequencies from the knowledge of the current passing through the neck, Eq. 3, and the total potential drop between the two clusters. In this case, the potential in real space and frequency domain is a complex number calculated from the complex induced charge density distribution. The current also has a real and a imaginary part. Thus, the result in this case will be a complex number reflecting the capacitive component of the impedance of the system.

In order to extract the potential drop across the junction in a meaningful way it is necessary that the potential inside each cluster becomes constant. This condition is met for configurations having a monatomic neck and at low frequencies. However, when the neck becomes wider and the energy of the mode increases is not possible anymore to define a potential drop restricted to the neck, i.e., the mismatch between the transmission through

Figure 7: In panels (a) and (b) we can find the real and imaginary parts of the total electrostatic potential when a external field of 1 a.u. is applied along the dimer axis. Panel (a) corresponds to a nominal distance of 31.3 Å and panel (b) to a distance of 34.1 Å. The profiles are calculated along the line shown in panel (c) and for the CTP resonance frequency.

the neck and the clusters themselves is not sufficiently large that the potential drop localizes mainly along the neck. In particular, the imaginary part of the potential strongly varies inside the clusters. Due to this behavior of the imaginary part of the potential profile it is not possible to define a voltage drop for wide contacts. Thus, we focus on configurations formed by clusters connected by monatomic chains.

In Fig. 7 (a) and (b) we depict the real and imaginary parts of the total electrostatic potential for a nominal gap size of 31.3 Å and 34.1 Å plotted along the path in panel (c). As can be clearly seen, a bias drop across the monatomic chain can be meaningfully defined for these geometries. The computed conductances for these two cases are:

$$\begin{aligned} G(\omega_{CTP}) &= (4.52 + i2.21) \times 10^{-5} \text{ S}, & d = 31.3 \text{ Å} \\ G(\omega_{CTP}) &= (4.13 + i2.35) \times 10^{-5} \text{ S}, & d = 34.1 \text{ Å}. \end{aligned}$$

The corresponding moduli $|G(\omega_{CTP})|$ of the conductance are, respectively, $5.03 \times 10^{-5} \text{ S}$ ($0.65 G_0$) for $d=31.3 \text{ Å}$ and $4.75 \times 10^{-5} \text{ S}$ ($0.61 G_0$) for $d=34.1 \text{ Å}$, where $G_0 = 7.75 \times 10^{-5} \text{ S}$ is the quantum of conductance.

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