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Manuscript Draft

Manuscript Number:

Title: High surface - highly N-doped carbons from hydrothermally-treated tannin

Article Type: Original Research Paper

Keywords: Hydrothermal carbonisation; Aminated tannin; Nitrogen-doped carbon materials; Carbon gels.

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Abstract: Nitrogen-doped carbon materials (NCMs) were obtained by amination of tannin by hydrothermal carbonisation (HTC). NCMs presented different morphologies depending on the amination method: (i) powders by HTC of tannin in a concentrated aqueous ammonia solution; (ii) gels without using any crosslinker by HTC in distilled water of tannin first dissolved in concentrated ammonia and then evaporated. Whatever the method, (i) or (ii), HTC was carried out at different temperatures: 180, 190, 200, 210 or 220°C, with further pyrolysis at 900°C under nitrogen. The morphology and chemical composition of the resultant materials were discussed. We compared our results with those reported in the open literature and we concluded that HTC of tannin in a concentrated aqueous ammonia solution allowed obtaining outstanding materials because they have both surface areas, higher than 500 m² g⁻¹, and high N content, around 8 wt.%.

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2 High surface - highly N-doped carbons from
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26 Abstract

27 Nitrogen-doped carbon materials (NCMs) were obtained by amination of tannin by
28 hydrothermal carbonisation (HTC). NCMs presented different morphologies depending on the
29 amination method: (i) powders by HTC of tannin in a concentrated aqueous ammonia
30 solution; (ii) gels without using any crosslinker by HTC in distilled water of tannin first
31 dissolved in concentrated ammonia and then evaporated. Whatever the method, (i) or (ii),
32 HTC was carried out at different temperatures: 180, 190, 200, 210 or 220°C, with further
33 pyrolysis at 900°C under nitrogen. The morphology and chemical composition of the resultant
34 materials were discussed. We compared our results with those reported in the open literature
35 and we concluded that HTC of tannin in a concentrated aqueous ammonia solution allowed
36 obtaining outstanding materials because they have both surface areas, higher than 500 m² g⁻¹,
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40 Carbon gels

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46 1. Introduction

47 N-doped carbon materials (NCMs) have gained an increasing interest in the last decade,
48 mainly because doping carbonaceous materials with nitrogen is an effective way of modifying
49 their surface chemistry and improving their electrochemical properties (Lee et al., 2011; Nieto-
50 Marquez et al., 2009). It has been suggested that NCMs provide pseudocapacitance through
51 pseudofaradaic reactions and also enhance the electronic conductivity of carbon structures.
52 For these reasons, such kind of materials can be used in electric double-layer capacitors
53 (Frackowiak et al., 2006; Hulicova et al., 2006; Hulicova-Jurcakova et al., 2009) and also in
54 fuel cells (Qiu et al., 2011; Steele and Heinzl, 2001). NCMs can also act as efficient
55 adsorbents of SO₂ from flue gases, or of phenol from aqueous effluents (Bimer et al., 1998; Li
56 et al., 2001; Przepiórski, 2006).

57 NCMs can be obtained from biomaterials naturally containing high levels of nitrogen or
58 from chemically modified carbons (Bimer et al., 1998). A recent work has shown that NCMs
59 can successfully be prepared from an abundant biomaterial, tannin, submitted to hydrothermal
60 carbonization (HTC) in a concentrated ammonia solution at 180°C, followed by pyrolysis at
61 900°C (Braghioli et al., 2012). The structure of the main compound contained in condensed
62 Mimosa tannin extract (*Acacia mearnsii*, de Wild) is presented in Fig. 1. Mimosa tannin is a
63 suitable product for preparing carbonaceous materials, as it is based on phenolic molecules
64 and contains aromatic structures such as resorcinol, pyrogallol and catechol. Such tannin is
65 quite reactive and can be extracted from several species of *Acacia* trees (Pizzi, 2008; Pizzi and
66 Mittal, 2003). The corresponding condensed polyflavonoid species comprise flavan-3-ols
67 having approximately 70% of repeating units of robinetinidin (resorcinol A-ring; pyrogallol
68 B-ring), which is shown in Fig. 1, and 25% of fisetinidin (resorcinol A-ring; catechol B-
69 ring) (Bate-Smith and Swain, 1962; Hemingway, 1989; Pizzi, 1983; Pizzi, 1994). Moreover, it
70 has been also proved that tannin, in the presence of a crosslinker such as formaldehyde,

71 hexamethylenetetramine, glyoxal, or 1-, 3-, or 5-trioxane, is a very suitable material for
72 preparing high-quality and low-cost gels and related carbon materials. Mimosa tannin is thus
73 able to replace resorcinol, which is the standard reagent used to prepare the same kind of
74 materials (Amaral-Labat et al., 2012; Amaral-Labat et al., 2013b; Braghiroli et al., 2013;
75 Grishechko et al., 2013; Reddy et al., 2003).

76 Amination is an easy way of incorporating nitrogen into phenolic compounds. Until
77 recently, amination of Mimosa tannin was thought to be regioselective, only affecting the
78 position 4' at which the hydroxyl group can be converted into amine (see asterisk in Fig. 1)
79 (Hashida et al., 2009). However, we have recently demonstrated that reactions of flavonoid
80 tannins in concentrated ammonia solution at room temperature are more complex, leading to
81 multiamination of a higher proportion of phenolic hydroxyl groups in both A and B rings, as
82 suggested in Fig. 1, also possibly opening the heterocycle as well as producing
83 oligomerisation and crosslinking between flavonoid units through $-N=$ bridges (Braghiroli et
84 al., 2013). Surface areas as high as $500 \text{ m}^2 \text{ g}^{-1}$, and nitrogen content as high as 6.3 wt. %, were
85 obtained, depending on the amination method (Braghiroli et al., 2012). However, both high
86 surface area and high nitrogen content could not be obtained simultaneously in the same
87 material, the one having the highest N content being the one having the lowest surface area,
88 and vice versa. The need of meeting the desirable target of having both high N content and
89 high surface area motivated the present study.

90 Therein, monolithic NCMs gels and NCMs powders were prepared by HTC of tannin in
91 the absence of crosslinker. We show here that, in hot pressurised water at a temperature within
92 the range 180 – 220°C, HTC is a powerful way of preparing N-doped carbonaceous materials
93 from flavonoid tannin, presenting nitrogen contents up to 13.7 and 8.1 wt.% before and after
94 pyrolysis at 900°C, respectively.

95

96 2. Experimental

97 2.1 Materials

98 2.1.1 *Tannin*

99 Commercial Mimosa tannin, extracted industrially in Tanzania, was kindly supplied by the
100 company SilvaChimica (Italy). The procedure, based on leaching Mimosa barks with a warm
101 aqueous solution, has been detailed elsewhere ([Amaral-Labat et al., 2013a](#)). The resultant
102 tannin solution was then concentrated and spray-dried, leading to a light-brown powder
103 containing 80 – 82% of actual phenolic flavonoid materials, 4 – 6% of water, 1% of amino
104 and imino acids, the remainder being monomeric and oligomeric carbohydrates, in general
105 broken pieces of hemicelluloses.

106 2.1.2 *Amination and HTC*

107 Aminated tannin samples were prepared by dissolving 2.0 g of raw tannin powder in 16
108 mL of a 28-30 wt.% of ammonia in water. Two different procedures were carried out. In the
109 first one, leading to samples called H-AT (hydrothermal-aminated tannin), the tannin solution
110 in ammonia was directly submitted to HTC. In the second one, the same mixture was
111 evaporated at room temperature, and then the resultant powder was submitted to HTC in
112 distilled water, leading to samples called H-EAT (hydrothermal-evaporated aminated tannin).
113 A control sample called H-T (hydrothermal-tannin) was also prepared for comparison
114 purposes, using the same amount of tannin: 2.0 g in 16 mL of water.

115 Both methods have been already described in detail elsewhere ([Braghiroli et al., 2012](#)).
116 Each tannin solution was put in an open glass vial, itself placed into a Teflon-lined autoclave
117 for HTC. The autoclave, always filled at one third of its available inner volume whatever the
118 sample, was closed and placed into a pre-heated ventilated oven for 24 h at different
119 temperatures, between 180 and 220°C. After HTC, the autoclave was left to cool down at room

120 temperature. The resultant dark powders or monoliths, depending on the synthesis method,
121 were recovered, washed with distilled water and then placed inside of a vacuum oven at 80°C
122 for drying during 12h.

123 2.1.3 Pyrolysis

124 Pyrolysis was carried out under nitrogen flow in a tubular furnace at a heating rate of 1°C
125 min⁻¹ up to 900°C, and the final temperature was maintained for 3 hours. Pyrolysed H-T, H-
126 EAT and H-AT samples were labelled CH-T, CH-EAT and CH-AT, respectively.

127 2.2 Characterisation

128 2.2.1 NMR and MALDI-ToF studies

129 Solid-state CP-MAS (cross-polarisation/magic angle spinning) ¹³C NMR spectra for H-T,
130 H-AT and H-EAT were recorded on a Bruker MSL 300 FT-spectrometer at a frequency of
131 75.47 MHz. Chemical shifts were calculated with respect to tetramethylsilane (TMS). The
132 rotor was spun at 4 kHz on a double-bearing 7 mm Bruker probe. The spectra were acquired
133 with 5 s recycle delays, a 90° pulse of 5 μs and a contact time of 1 ms. The number of
134 transients was 3000, and the decoupling field was 59.5 kHz.

135 MALDI-ToF (Matrix-Assisted Laser Desorption Ionisation – Time of Flight) spectra were
136 recorded on a Shimadzu Axima Precision instrument. 5 mg of each sample were dissolved in 1
137 mL of an aqueous solution acetone:water (1:1), and then mixed with a matrix, 2,5-dihydroxy
138 benzoic acid/tetrahydrofuran. For the enhancement of ion formation, NaCl was added to the
139 matrix. The resulting solution was placed on the MALDI target. After evaporation of the
140 solvent, the MALDI target was introduced into the spectrometer. The irradiation source was a
141 pulsed nitrogen laser with a wavelength of 337 nm. The length of one laser pulse was 3 ns.
142 The measurements were carried out using the following conditions: polarity-positive, flight

143 path-linear, mass-high (20 kV acceleration voltage), 100 – 150 pulses per spectrum. The
144 delayed extraction technique was used applying delay times of 200 – 800 ns.

145 2.2.2 *Elemental analysis*

146 Bulk elemental analyses were carried out in a ThermoFlash 1112 apparatus to determine
147 carbon, hydrogen, nitrogen and sulphur contents. Oxygen content was obtained by difference.
148 Surface elemental analyses were performed by X-ray photoelectron spectroscopy (XPS), using
149 an ESCAPlus OMICROM system equipped with a hemispherical electron energy analyzer.
150 The spectrometer was operated at 10 kV and 15 mA, using a non-monochromatized Mg K α
151 X-ray source ($h\nu = 1253.6$ eV) under vacuum ($< 5 \times 10^{-9}$ Torr). Analyzer pass energy of 50
152 eV was used for survey scans and 20 eV for detailed scans. The C1s peak around 284.5 eV
153 was used for binding energy correction. A survey scan (1 sweep/200 ms dwell) was acquired
154 between 1100 and 0 eV. Current region sweeps for N1s, O1s and C1s were obtained. The
155 CASA data processing software allowed smoothing, Shirley-type background subtraction,
156 peak fitting and quantification.

157 2.2.3 *Morphology and texture*

158 Samples' morphology was observed with a FEI Quanta 400 scanning electron microscope
159 (SEM). Pore texture parameters were derived from nitrogen or krypton adsorption isotherms
160 at -196°C using a Micromeritics ASAP 2020 automatic apparatus, whether they were
161 determined before or after pyrolysis at 900°C , respectively. The choice of the probe molecule
162 was dictated by the surface area, which was very low for non-pyrolysed samples, and for
163 which Kr adsorption was thus much more accurate. Prior to any adsorption experiment, non-
164 pyrolysed and pyrolysed samples were degassed for 48 h under vacuum at 60°C or at 250°C ,
165 respectively. Kr and N $_2$ adsorption studies were carried out, and the corresponding isotherms
166 were treated for obtaining the texture parameters. From N $_2$ adsorption isotherms, the following

167 quantities were obtained: surface area (S_{BET}), from the BET calculation method (Brunauer et
168 al., 1938) in the 0.01 to 0.05 P/P₀ interval, micropore volume (V_{μ}) (Dubinin, 1989); total pore
169 volume ($V_{0.97}$) determined as the N₂ adsorbed at P/P₀ = 0.97; and mesopore volume (V_m),
170 calculated as the difference $V_{0.97} - V_{\mu}$ (Gregg and Sing, 1991). From Kr adsorption isotherms,
171 the surface area (S_{BET}) was obtained in the 0.05 to 0.2 P/P₀ interval.

172

173 3. Results and discussion

174 3.1 Chemical structure of hydrothermally treated tannin

175 Solid-state ¹³C NMR spectra of hydrothermal carbons H-T, H-EAT and H-AT, obtained
176 after HTC at 180°C, are shown in Fig. S1 of Supplementary Material. The detailed
177 interpretation of these spectra is also given in Supplementary Material, but the main
178 conclusions are the following. H-T has a similar structure as that of flavonoid tannin, but
179 presents a much higher level of polymerisation, mainly through the opening of the
180 heterocyclic C2-ring and subsequent partial condensation between C2 (open) to C6 or C8, and
181 partly through some autocondensation C4 to C6 and C8. H-EAT shares the main features of
182 H-T in terms of autocondensation and heterocyclic C2-ring opening, but is also characterized
183 by the amination of C3' and C4' sites, the preservation of some heterocyclic structures, and a
184 lower level of polymerization. The presence of -NH- bridges is also observed, possibly
185 explaining why H-EAT is a gel whereas no crosslinker was initially present in the system.
186 Finally, H-AT is a highly polymerized and crosslinked material, in which the typical tannin
187 structure is almost completely lost.

188 MALDI-ToF spectra of the same materials and their detailed interpretation are given in
189 Fig. S2 of Supplementary Material. H-T, although polymerised as proved by NMR spectra,
190 exhibits the classical features of normal tannin, as the expected flavonoid units are detected.

191 H-EAT, as a gel, is totally different, and the long three-dimensional chains from which it is
192 made cannot be detected with this technique. The few peaks of the spectrum were attributed to
193 flavonoid structures doped by nitrogen, such as those shown in Fig. 2(a-b), accounting for the
194 observed peaks at 554 and 555 Da. As the $-NH_2$ groups can have substituted any one of the $-$
195 OHs of the two flavonoid units from which these species are based on, their distribution is
196 likely to be variable. A small peak at 843-844 Da was interpreted as shown in Fig. 2(c), i.e. by
197 a 554 or 555 Da compound above to which has been added a $290-1 = 289$ Da unit, either a
198 robinetinidin or a catechin. Finally, H-AT presents many species in which $-OH$ groups have
199 been substituted by $-NH_2$ groups. More details are given in Supplementary Material.

200

201 3.2 Elemental composition of all materials

202 The elemental analyses of all materials prepared by HTC at different temperatures: 180,
203 190, 200, 210 and 220°C are given in Table S3 of Supplementary Material, before and after
204 pyrolysis at 900°C. The carbon content slightly increased with HTC temperature, whatever the
205 preparation method. This finding was observed at the same small extent after pyrolysis at
206 900°C. The carbon content of natural tannin mimosa extract used in this work was 53.8 wt.%,
207 and only increased by 5-11% after HTC. Whereas such kind of material is unanimously called
208 “hydrothermal carbon” in the literature (Baccile et al., 2009), the resultant microspheres or
209 gels are thus not really carbon materials until a pyrolysis is carried out at much higher
210 temperature. Indeed, and as shown in Table S3, the carbon content reached around 90 wt.%
211 after pyrolysis.

212 Fig. 3 shows nitrogen content for materials before and after pyrolysis. Tannin contains
213 naturally nitrogen due to the presence of amino and imino acids. H-T materials had nitrogen
214 content lower than 1%, which still decreased after pyrolysis. As expected, pyrolysis at 900°C
215 produced a significant decrease of the nitrogen content. Nitrogen is indeed a volatile element,
216 which evolves as more easily as the heat-treatment temperature and/or the porosity of the

217 material in which it is contained is high. The challenge was thus to prepare materials having
218 both a high porosity and a high nitrogen content. H-EAT materials, having a nitrogen content
219 around 4 wt.%, led to materials, CH-EAT, whose N content was close to 2 wt.% after
220 pyrolysis, irrespective to the HTC temperature as a first approximation. In contrast, increasing
221 HTC temperature produced a nitrogen enrichment of H-AT materials, with an outstanding N
222 content ranging from 12 to 14 wt.%. After pyrolysis, a roughly constant but still amazing 8
223 wt.%, on average, was obtained in CH-AT materials.

224 The percentage of nitrogen after amination of tannin would have been 4.8 wt.% if only the
225 site 4' of the B-ring (see again [Fig. 1](#)) was aminated, which situation happens at room
226 conditions according to [Hashida et al. \(2009\)](#) and [Morisada et al. \(2011\)](#). This statement is
227 consistent with the results for H-EAT. However, as recently shown [Braghiroli et al. \(2013\)](#) and
228 again suggested by the present NMR and MALDI-ToF spectra, tannin can be aminated on
229 both A and B rings, and can also bear $-N=$ bridges between flavonoid units as seen in [Fig. 1](#),
230 thereby justifying higher nitrogen contents. However, for the sample having the highest N
231 content, H-AT, the NMR spectrum was hardly interpreted and the MALDI-ToF studies did
232 not help much, as $-NH_2$ and $-OH$ have very close molecular weights, and cannot be separated
233 in oligomers prone to be partly deprotonated. The molecular structure of N-doped tannin after
234 direct HTC in concentrated ammonia is still unknown, but should explain so high nitrogen
235 contents ranging from 11 to 13 wt.% before pyrolysis. Our previous characterizations of this
236 kind of materials by XPS showed that H-EAT had a higher content of N-6 than H-AT, the
237 remainder being neutral amines. After pyrolysis at 900°C, the amount of N-6 decreased, and
238 N-5, N-X and maybe some N-Q were formed, but the corresponding moieties were the same
239 in both CH-EAT and CH-AT ([Braghiroli et al., 2012](#)).

240 Nitrogen-doped carbon materials were also prepared by HTC of chitosan and subsequent
241 pyrolysis at 750°C, having carbon and nitrogen contents of 79.2 wt.% and around 9.0

242 wt.%, respectively (Zhao et al., 2010a). The same method applied to glucosamine led to C and
243 N contents of 81.6 and 6.6 wt.%, respectively (Zhao et al., 2010a). Such N contents are similar
244 or below those measured in CH-AT, despite the significantly lower temperature of pyrolysis,
245 750°C, in favour of a much lower loss of nitrogen during the heat treatment. Zhang et al.
246 (2012) studied the amination of sucrose by HTC in ammonia at 160, 180 and 200°C, followed
247 by pyrolysis at 900°C. The resultant materials can thus be compared to H-AT and CH-AT and
248 their C and N contents were around 64 and 20 wt.% before pyrolysis, respectively, and 94.8
249 and 4.4 wt.% after pyrolysis, respectively. Reaching around 8 wt.% of nitrogen in CH-AT
250 even after heat-treatment at 900°C suggests that nitrogen was well stabilised in the
251 carbonaceous structure.

252 3.3 XPS analysis

253 The chemical analysis of the surface of the aminated tannin (EAT), dried hydrogels (H-
254 AT, H-EAT) and carbon gels (CH-AT, CH-EAT) was investigated by XPS. XPS spectra of N-
255 doped materials, with emphasis on the energy range corresponding to N1s are given in Fig. S3
256 of Supplementary Information part, with the corresponding curve fits allowing the
257 deconvolution of the peaks. The corresponding data are given in Tables 1 and 2. Caution
258 should be taken regarding the assignment of XPS peaks to chemical groups since the low
259 abundance of N increases the uncertainty of the multi-component curve-fitting peak and, non-
260 conductive samples (EAT, H-AT and H-EAT) could lead to local charging effect with shifted
261 binding energy to higher values. Moreover, the assignment is complicated by the location of N
262 within the polyaromatic network and the degree of condensation of the graphene layer
263 (Lahaye et al., 1999). For instance, N in pyridinic groups has its core level binding energy that
264 lie from 398.3 eV for conducting samples (e.g. N-doped graphite) up to 399.3 eV for isolated
265 pyridine group within the non-conducting poly(2-vinylpyridine) polymers or pyridinic N
266 located in a border of a polyaromatic cycle. A common way to assign peaks and especially N

267 in pyridinic group is to determine the energy difference $\Delta|N1s - C1s|$ between the binding
268 energy of N1s and reference C1s taken at the maximum of the C1s envelope for C-C or C=C
269 bonds. For different reference compounds containing pyridinic N, Δ values range from 113.7
270 to 114.3 eV (Lahaye et al., 1999). Assignment of other peaks could then be done by the
271 relative shift to that N-pyridinic reference.

272 For tannin modified with ammonia at room temperature (EAT), XPS composition reveals
273 a high fraction of O (28 at.%), a minor amount of N (3 at.%) and traces of K (0.65 at.%). The
274 N1s peak exhibits two components at 399.9 and 402.1 eV. Based on the previous discussion,
275 $\Delta|N1s - C1s|$ was calculated with the binding energy of 399.9 eV and 284.7 eV for N1s and
276 C1s respectively. It leads to a value of 115.2 eV, outside the range of 113.7 to 114.3 eV
277 expected for N pyridine species (Lahaye et al., 1999). It suggests that the main peak at 399.9
278 eV could not be attributed to pyridinic N species but rather to functional groups having
279 slightly higher binding energy, around + 1.1 eV with respect to the averaged N pyridinic
280 binding energy (398.8 ± 0.5 eV). This feature is in agreement with our previous study of EAT
281 using Maldi-Tof and solid NMR spectroscopies where no pyridinic groups were detected
282 (Braghiroli et al., 2013). Moreover, it is well known in the literature that conversion of
283 carbonaceous materials containing no pyridinic groups (e.g cellulose) to pyridinic nucleiis
284 supposed to occur at temperature higher than 250°C in presence of gaseous NH₃ (Cagniant et
285 al., 2002). Therefore, the possible N-containing chemical groups are attributed mainly to
286 neutral amine and/or amide since both functions could have their binding energy in the range
287 around 399.9 eV. Regarding imine groups, only traces could be present if we consider their
288 classical binding energy around 398.5 eV. They could more significantly be present by
289 considering their assignment at 399.0 ± 0.1 eV as performed by Cagniant et al. (2002) by
290 coupling XPS and IR spectroscopies for ammonia-treated cellulose at low temperature
291 (250°C). In that case, these data appear in agreement with the previous characterization of

292 EAT (Braghioli et al., 2013) where multiamination of phenolic hydroxyl groups with
293 formation of bridging -C=N- group and oxidation of phenolic-OH groups to quinone were
294 observed. The latter point could explain the high oxygen content determined by XPS at the
295 surface of the sample. The peak at 402.1 eV was attributed to quaternary nitrogen such as
296 protonated amides or amines.

297 When tannin and EAT were submitted to hydrothermal treatment in concentrated aqueous
298 ammonia (H-AT) or in water (H-EAT) respectively, the XPS compositions evolve as shown in
299 Table 2, with a decrease of the oxygen content in both cases and a significant nitrogen
300 enrichment, around 3 at% for H-EAT up to 8-9 at% for H-AT compared to 0.6 at% for natural
301 tannin and 1 at% after HTC of tannin. Presence of Si (0.7 at % for H-EAT) could also be
302 detected when silica vials were used in the autoclave during HTC.

303 Based on the literature, the presence of pyridinic species is now possible since it is well
304 known that HT process promotes the carbonisation of tannin with polycondensation of aromatic
305 units and formation of N-containing cycles such as pyridinic species (Baccile et al., 2011;
306 Wohlgemuth et al. 2012a; Wohlgemuth et al., 2012b). For instance, Cagniant et al. (2002),
307 have proposed the formation of pyridinic group during the carbonization of cellulose at 250°C
308 in ammonia atmosphere (Cagniant et al., 2002). The N1s XPS spectra shown an asymmetry of
309 the main peak associated with a shift to lower binding energy compared to EAT sample
310 (399.3-eV instead of 399.9 eV). The $|\Delta|N1s - C1s|$ values were around 114.8 eV, outside the
311 expected range for pyridinic species, suggesting that the N- pyridinic species are not the main
312 contribution to this peak. Indeed, its presence could not be excluded since the N1s binding
313 energy domains of pyridinic groups are scattered in a large range (1eV). The latter overlap
314 those of neutral amines and therefore, the asymmetric N1s peaks were assigned to these two
315 species without any possible deconvolution of individual contributions as mistakenly done in
316 our previous study (Braghioli et al., 2012).

317 After carbonization at 900°C, no amines were present, as expected, and they were mainly
318 replaced by pyridinic N-6 and to a lower other species such as pyrrolic (N-5) or pyridonic (not
319 distinguishable by XPS measurements), quaternary pyridinic nitrogen (N-Q) (Lahaye et al.,
320 1999) or pyridine oxide groups. Despite the high purity of the inert gases used for
321 carbonization, the presence of oxide forms (pyridone, pyridine oxide) could be ascribed to the
322 reactivity of carbonized materials with air as previously observed in the works of Pels et al.
323 (1995) and Lahaye et al. (1999).

324 3.4 Materials' morphology and porous texture

325 Figure 4 shows materials morphology for samples prepared at 190 and 210°C. CH-T
326 materials are spherical particles, more or less connected with each other, whose average
327 diameter and agglomeration state increased with HTC temperature (Fig. 4(a-b)). The same
328 was observed for the material directly aminated at HTC conditions, CH-AT (Fig. 4(c-d)), but
329 the polydispersity was much higher at 210°C. Indeed, whereas the particles were rather
330 monodisperse at 190°C, a mixture of small and very big particles was found after HTC at
331 210°C. Fig. 4(d) even shows big, basketball-like, spheres. CH-EAT (Fig. 4(e-f)) is a carbon
332 gel within which the spherical nodules were so small that they were hardly observed at the
333 magnifications used. CH-EAT is in fact a carbon xerogel as the material was dried in vacuum
334 at 80°C before pyrolysis at 900°C. Fig. 4(f) also shows that bigger particles with a higher
335 polydispersity were obtained with increasing temperature. Li et al. (2011) submitted glucose
336 to HTC and also found that the most spherical particles were formed at lower temperatures of
337 180-190°C, whereas more polydisperse spheres with broken shapes appeared at higher
338 temperatures such as 210°C. On the contrary, Titirici et al. (2013) found that higher
339 temperatures led to larger but more homogeneous particles when glucose was submitted to
340 HTC at temperatures from 120 to 280°C.

341 Although comparisons are made all along the present paper with materials derived from
342 HTC of polysaccharides, for which the literature is the most abundant, tannin treated in similar
343 conditions behaves differently in some aspects. For example, amino-functionalised materials
344 prepared by HTC of glucose in ammonia at pH 11 at 150°C were obtained in the form of
345 spheres having a very narrow distribution of diameters centred on around 2-3 μm (Wang et al.,
346 2012). In the case of tannin, the best, most spherical particles were obtained at 180 – 190°C,
347 lower temperatures within the range 130 – 160°C leading to much more irregular and
348 polydisperse spheres (Braghioli et al., 2014).

349 Fig. 5 shows the specific surface areas of all materials prepared at different HTC
350 temperatures, measured before and after pyrolysis at 900°C. Before pyrolysis, the gel H-EAT
351 was the material having the highest surface area, around 100 $\text{m}^2 \text{g}^{-1}$. This is a logical result,
352 considering that such material comprises the smallest particles. In contrast, H-T was the one
353 having the lowest surface area, less than 1 $\text{m}^2 \text{g}^{-1}$, due to his rather monodisperse, solid and
354 big, spherical particles. As expected, H-AT had values of surface area between those of the
355 former two materials, slightly lower than 20 $\text{m}^2 \text{g}^{-1}$, since its morphology is in-between a
356 powder and a gel. All the materials had surface areas that decreased slightly when the HTC
357 temperature increased, due to both growth and agglomeration of the particles. The effect was
358 the most significant in the case of H-EAT, whose very small nodules were more affected by
359 particle growth and Ostwald ripening.

360 After pyrolysis at 900°C, the materials CH-T, CH-AT and CH-EAT presented developed
361 porosity and significantly higher surface areas, as shown in Fig. 5. Depending on the
362 preparation method, different effects of the HTC temperature were observed. CH-T presented
363 the highest surfaces areas, around 670 $\text{m}^2 \text{g}^{-1}$, which did not change much when the HTC
364 temperature increased. Such values are higher than most results reported so far for
365 carbohydrates submitted to HTC and subsequently pyrolysed at high temperature. Glucose,

366 chitosan and glucosamine indeed presented low porosity not only after HTC, but also after
367 pyrolysis, so that the resultant measured surface areas were around 30-50 m² g⁻¹ in general
368 (Zhao et al., 2010a). In the case of CH-AT, the materials obtained by HTC at 185-190°C and
369 subsequent pyrolysis had the highest value of surface area, up to 550 m² g⁻¹. Higher HTC
370 temperatures led to lower surface areas probably due to the formation of bigger and less
371 porous carbon spheres, as suggested by Fig. 4(d). The surface areas of the carbon gel CH-EAT
372 presented a maximum at 180°C, the lowest HTC temperature. On average, the surface area of
373 CH-EAT was around 400 m² g⁻¹.

374 Fig. 6 shows the nitrogen adsorption-desorption isotherms at -196°C for the materials
375 shown in Fig. 4. Nitrogen adsorption isotherms of CH-T materials were type I, according to
376 the IUPAC classification, with a sharp increase of the nitrogen amount adsorbed at P/P₀ lower
377 than 0.05, a narrow knee and a plateau up to P/P₀ equal to 0.99. These isotherms are
378 characteristic of purely microporous solids. Nitrogen adsorption isotherms of CH-AT
379 materials were similar to those of CH-T but presented a lower volume adsorbed at low P/P₀ <
380 0.05, indicating a lower microporosity than CH-T materials, and a sudden increase of the
381 nitrogen uptake P/P₀ < 0.95, indicating capillary condensation in wide mesopores. Nitrogen
382 adsorption isotherms of CH-EAT materials were combinations of types I and IV, for which
383 adsorption of nitrogen takes place in both micro and mesopores. The existence of a well-
384 developed mesoporosity is also indicated by the slope of the adsorption isotherms, in the P/P₀
385 range from 0.05 to 0.7, and by the large hysteresis cycle. Textural parameters of all the carbon
386 materials prepared are given in Table S4.

387 The carbon xerogels, CH-EAT materials, were those materials presenting the highest
388 proportion of mesopores, between 66 and 76% of the total pore volume. Whereas these
389 materials were xerogels, i.e. were not submitted to supercritical drying after synthesis, a
390 significant amount of mesoporosity was thus maintained. N-doped carbon materials were also

391 prepared by [White RJ et al. \(2011\)](#) from the HTC of glucose and ovalbumin. A gel was
392 obtained, which was dried by exchange with supercritical CO₂, and then pyrolysed. The
393 resultant carbon aerogel also presented a high mesopore volume of 0.51 cm³ g⁻¹, and its
394 surface area was 310 m² g⁻¹. These characteristics are thus similar to those of our CH-
395 EAT_200°C sample, but in our case a simple subcritical drying was carried out. Therefore,
396 our process is expected to be much more convenient and cheaper for preparing N-doped
397 porous carbon materials.

398 3.4 Comparison with other N-doped carbon materials

399 [Fig. 7\(a\)](#) gathers the data of surface area and nitrogen content of all the tannin-derived
400 materials prepared by HTC in the present work. The same kind of data is also given on the
401 same plot for other carbonaceous materials produced by HTC of various precursors, such as
402 chitosan, glucose and glucosamine ([Zhao et al., 2010a](#)), glucose and albumin ([Baccile et al.,](#)
403 [2010](#)), glucose and ovalbumin ([White et al., 2011](#)) and D-glucosamine ([Zhao et al., 2010b](#)). H-
404 T presented both low surface area and very low nitrogen content, but their aminated
405 counterparts were among the materials having either the highest N content (H-AT) or the
406 highest surface area (H-EAT). But none presented both high characteristics at the same time.
407 The H-EAT gel was comparable to the material obtained by HTC of a mixture of glucose and
408 albumin ([Baccile et al., 2010](#)), having values of S_{BET} and N content around 100 m² g⁻¹ and 4-5
409 wt.%, respectively. [Fig. 7\(b\)](#) shows results for the same precursors as in [Fig. 7\(a\)](#) and a few
410 others more, such as cystine and T-cystine ([Wohlgemuth et al., 2012b](#)) and prawn shells
411 ([White et al., 2009](#)) after HTC and posterior pyrolysis at different temperatures: 550
412 ([Wohlgemuth et al., 2012b](#)), 750 ([Baccile et al., 2010](#); [White et al., 2009](#); [White et al.,](#)
413 [2011](#); [Zhao et al., 2010a](#)), 900 ([Wohlgemuth et al., 2012b](#)) and 950°C ([Baccile et al., 2010](#);
414 [White et al., 2011](#)). After KOH activation at 600°C of D-glucosamine submitted to HTC
415 ([Zhao et al., 2010b](#)), S_{BET} as high as 600 m²g⁻¹ was obtained but N content was only 2.3

416 wt.%. Even higher S_{BET} , $730 \text{ m}^2\text{g}^{-1}$, was obtained after carbonisation at 900°C of T-cystine
417 submitted to HTC but N content was lower than 4 wt.% (Wohlgemuth et al., 2012b). Pyrolysis
418 obviously led to a significant decrease of nitrogen content, and especially at the highest
419 temperatures. However, it is important to point out that despite H-AT was heat-treated at
420 900°C , the resultant carbon CH-AT was among the materials presenting the highest N content,
421 around 8 wt.%, with a surface area as high as $500 \text{ m}^2\text{g}^{-1}$.

422 In this study, we obtained carbon materials with simultaneously high nitrogen content and
423 high surface areas. Specially tannin directly submitted to HTC in concentrated ammonia at
424 190°C and pyrolysed at 900°C presented S_{BET} and nitrogen content of $500 \text{ m}^2 \text{ g}^{-1}$ and 8 wt.%,
425 respectively. The incorporation of nitrogen in the carbonaceous structure is indeed expected to
426 increase its electrical conductivity and provide pseudofaradaic reactions, thus further
427 increasing the capacitance and the performances of the porous electrode. Therefore, this
428 material is foreseen to be tested in several applications, in particular as electrode of electric
429 double-layer capacitor. This will be done in the near future.

430

431 **4. Conclusion**

432 Hydrothermal carbonisation (HTC) has been used to produce carbonaceous materials from
433 Mimosa tannin in water or in 28 – 30 wt.% ammonia solution, and at temperatures ranging
434 from 180 to 220°C . N-doped materials were obtained either by tannin amination in ammonia
435 in room conditions followed by HTC in distilled water (H-EAT), or by direct HTC in
436 ammonia (H-AT). The former method led to gels despite no crosslinker was present, and the
437 latter led to microspheres. All were based on autocondensed and partly dehydrated tannin,
438 mainly through heterocycle opening, the level of polymerisation depending on the preparation
439 method. The N-doped materials presented rather low surface areas, but H-AT had outstanding
440 nitrogen contents as high as almost 14 wt.%. Most of the nitrogen was in the form of amines,

441 with some possible pyridinic N. MALDI-ToF and NMR studies proved that nitrogen was
442 incorporated at several sites of the flavonoid units, unlike previous studies indicated.

443 Pyrolysis at 900°C systematically decreased the N content, as expected given the volatility
444 of this element, but simultaneously led to a significant porosity development. As a
445 consequence, the CH-AT material prepared by HTC at 190°C presented high surface area and
446 high N content, 500 m² g⁻¹ and 8 wt.%, respectively. To our deepest knowledge and after
447 having extensively revised the open literature, no other natural precursor submitted to HTC
448 reached so high N content and so high surface area at the same time. Such N-doped carbons,
449 prepared by a cheap and sustainable method, are very promising materials in a broad range of
450 applications such as water purification and electrochemical energy storage and conversion,
451 amongst others. These applications are presently under study.

452 **Acknowledgements**

453 The present research was partly made possible by the support of the Region Lorraine and
454 UHP through the programme 'Jeunes chercheurs'. The French authors from laboratories
455 hosted by ENSTIB also gratefully acknowledge the financial support of the CPER 2007-2013
456 "Structuration du Pôle de Compétitivité Fibres Grand'Est" (Competitiveness Fibre Cluster),
457 through local (Conseil Général des Vosges), regional (Région Lorraine), national (DRRT and
458 FNADT) and European (FEDER) funds.

459

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580

581 **Captions of the figures**

582 **Fig. 1:** The main flavonoid unit contained in Mimosa tannin: prorobinetinidin (above), and the
583 two possible structures that are obtained after amination of Mimosa tannin (Braghioli
584 [et al., 2013](#)).

585 **Fig. 2:** Chemical species suggested from the analysis of MALDI-ToF spectra of hydrothermal
586 aminated carbon gel before pyrolysis, H-EAT: (a) 554 Da; (b) 555 Da; (c) 843 Da (the
587 sodium of weight 23 was implicitly taken into account).

588 **Fig. 3:** Nitrogen content of N-doped materials prepared by HTC at different temperatures,
589 before and after pyrolysis before (H-T \diamond , H-AT \square and H-EAT \triangle), and after pyrolysis
590 at 900°C (CH-T \blacklozenge , CH-AT \blacksquare and CH-EAT \blacktriangle).

591 **Fig. 4:** SEM images of tannin-based materials after HTC and posterior pyrolysis at 900°C: (a)
592 CH-T_190°C and (b) CH-T_210°C; (c) CH-AT_190°C and (d) CH-AT_210°C; (e)
593 CH-EAT_190°C and (f) CH-EAT_210°C.

594 **Fig. 5:** Surface areas, S_{BET} , of all materials obtained after HTC at different temperatures:
595 before (H-T \diamond , H-AT \square and H-EAT \triangle), and after pyrolysis at 900°C (CH-T \blacklozenge , CH-
596 AT \blacksquare and CH-EAT \blacktriangle).

597 **Fig. 6:** Nitrogen adsorption (full symbols) – desorption (empty symbols) isotherms at -196°C. (
598 CH-T_190°C \blacklozenge , CH-T_210°C \blacksquare ; CH-AT_190°C \blacklozenge , CH-AT_210°C \blacksquare , CH-
599 EAT_190°C \blacklozenge , CH-EAT_210°C \blacksquare).

600 **Fig. 7:** Nitrogen content vs. surface areas, S_{BET} , of our materials: (a) before pyrolysis (H-T \diamond ,
601 H-AT \square and H-EAT \triangle); (b) after pyrolysis (CH-T \blacklozenge , CH-AT \blacksquare and CH-EAT \blacktriangle)
602 compared to results reported in literature.

High surface - highly N-doped carbons from hydrothermally-treated tannin

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Table 1: Contributions to the N1s bands in XPS patterns of HTC samples.

	Binding energy (eV)	(at. %)		
	Isolated Pyridinic (N-6) and/ or Neutral amines	N _{XPS}	C _{XPS}	O _{XPS}
H-AT_180°C	399.2	7.9	76.4	15.6
H-AT_190°C	399.1	8.1	78.7	13.2
H-AT_210°C	399.3	9.4	78.6	12.0
H-EAT_180°C	399.1	2.5	76.6	18.3
H-EAT_190°C	399.5	3.2	78.5	18.3
H-EAT_210°C	399.3	3.4	78.2	18.4

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Table 2: Contributions to the N1s bands in XPS patterns of carbonised HTC samples.

	Binding energy (eV) and area of the peak (%)			(at. %)		
	Pyridinic (N-6)	Pyrrolic (N-5) or pyridonic	Quaternary N-Q or oxydised pyrinic N	N _{XPS}	C _{XPS}	O _{XPS}
CH-AT_180°C	398.1 (35.7)	400.6 (55.3)	402.8 (9.0)	4.9	90.8	4.3
CH-AT_190°C	398.1 (38.2)	400.7 (57.7)	402.7 (4.1)	2.8	92.4	4.8
CH-AT_210°C	398.1 (39.2)	400.7 (59.2)	402.7 (1.6)	4.0	90.2	5.8
CH-EAT_180°C	398.1 (33.1)	400.7 (60.1)	402.8 (6.8)	1.6	95.8	2.6
CH-EAT_190°C	398.0 (36.7)	400.6 (60.9)	402.6 (2.4)	2.1	92.1	5.8
CH-EAT_210°C	398.1 (37.1)	400.7 (61.2)	402.7 (1.7)	1.9	94.4	3.7

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Figure 1

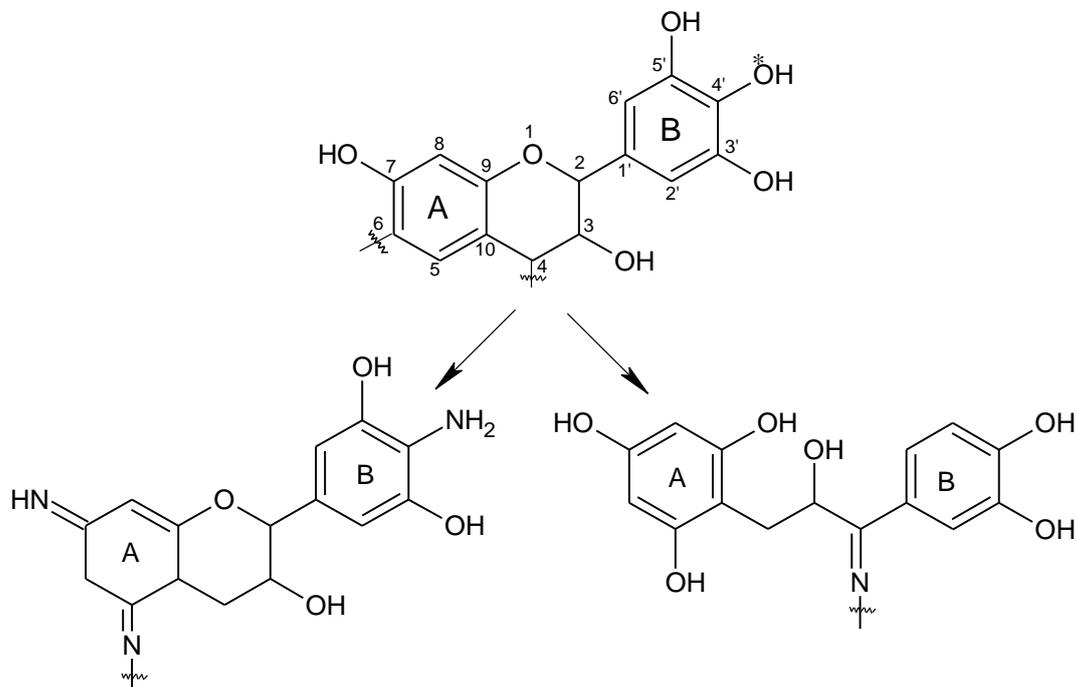


Fig. 1: The main flavonoid unit contained in Mimosa tannin: prorobinetinidin (above), and the two possible structures that are obtained after amination of Mimosa tannin (Braghiroli et al., 2013).

High surface - highly N-doped carbons from hydrothermally-treated tannin

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Figure 2

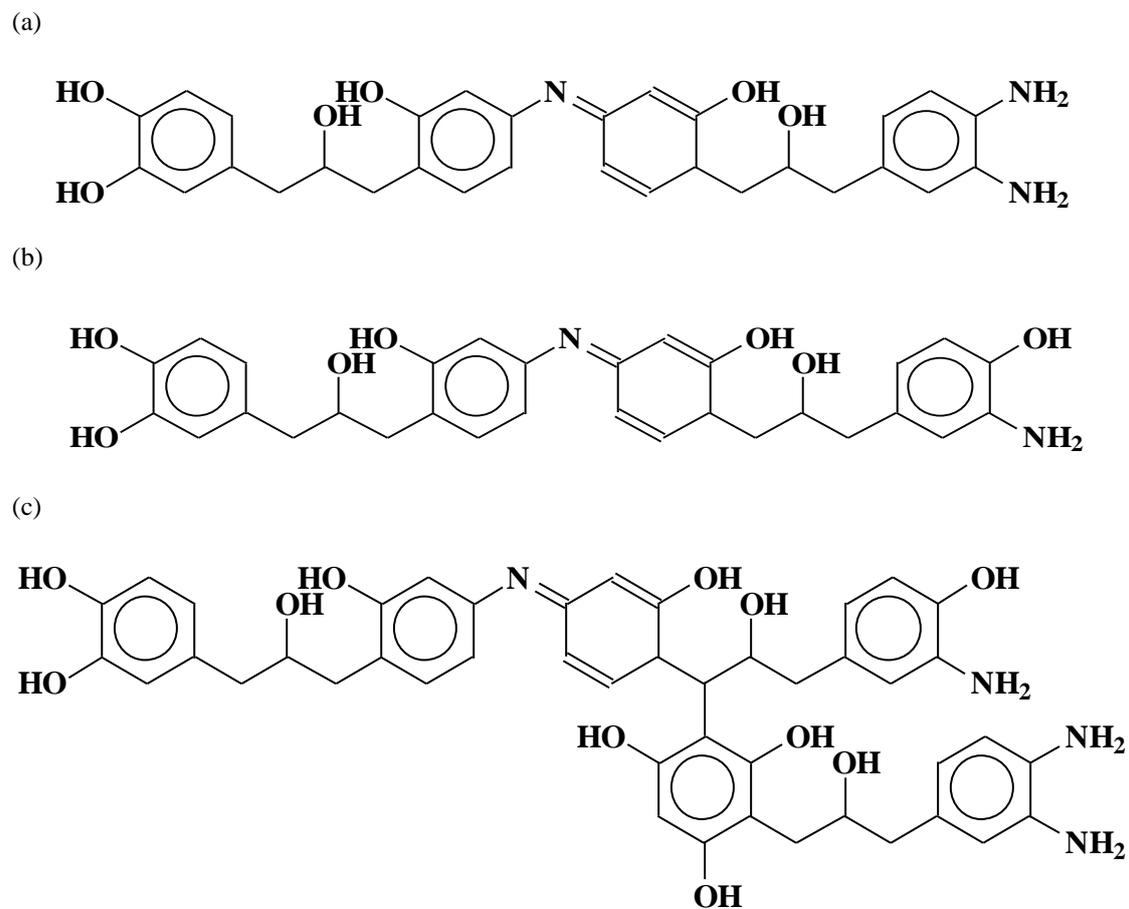


Fig. 2: Chemical species suggested from the analysis of MALDI-ToF spectra of hydrothermal aminated carbon gel before pyrolysis, H-EAT: (a) 554 Da; (b) 555 Da; (c) 843 Da (the sodium of weight 23 was implicitly taken into account).

High surface - highly N-doped carbons from hydrothermally-treated tannin

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Figure 3

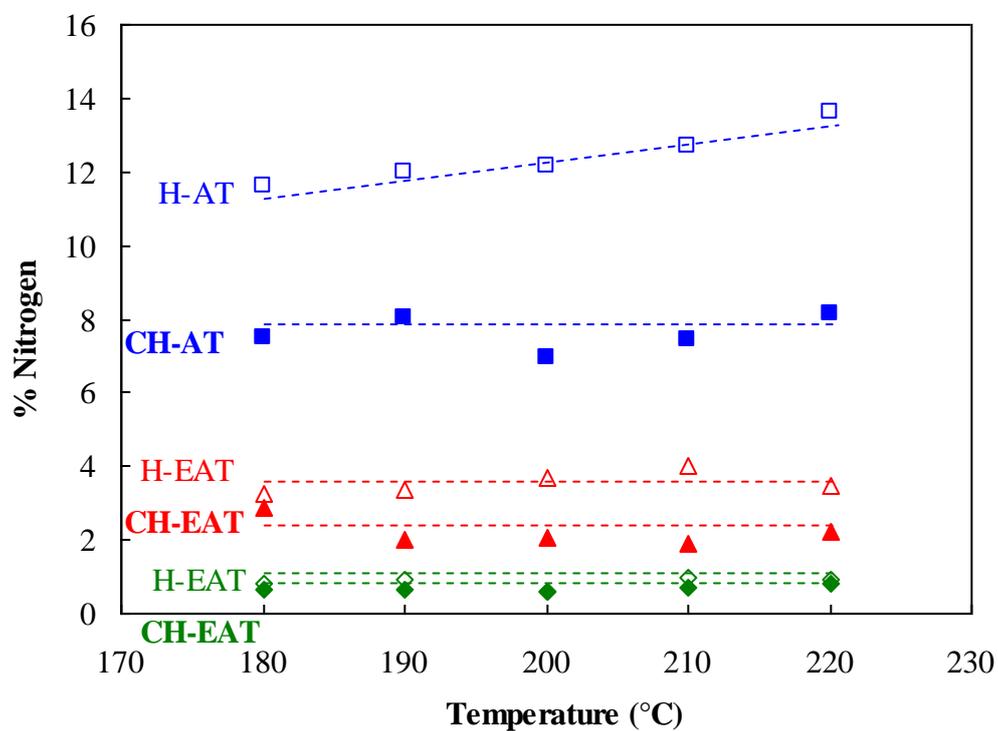


Fig. 3: Nitrogen content of N-doped materials prepared by HTC at different temperatures, before and after pyrolysis before (H-T \diamond , H-AT \square and H-EAT \triangle), and after pyrolysis at 900°C (CH-T \blacklozenge , CH-AT \blacksquare and CH-EAT \blacktriangle).

High surface - highly N-doped carbons from hydrothermally-treated tannin

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Figure 4

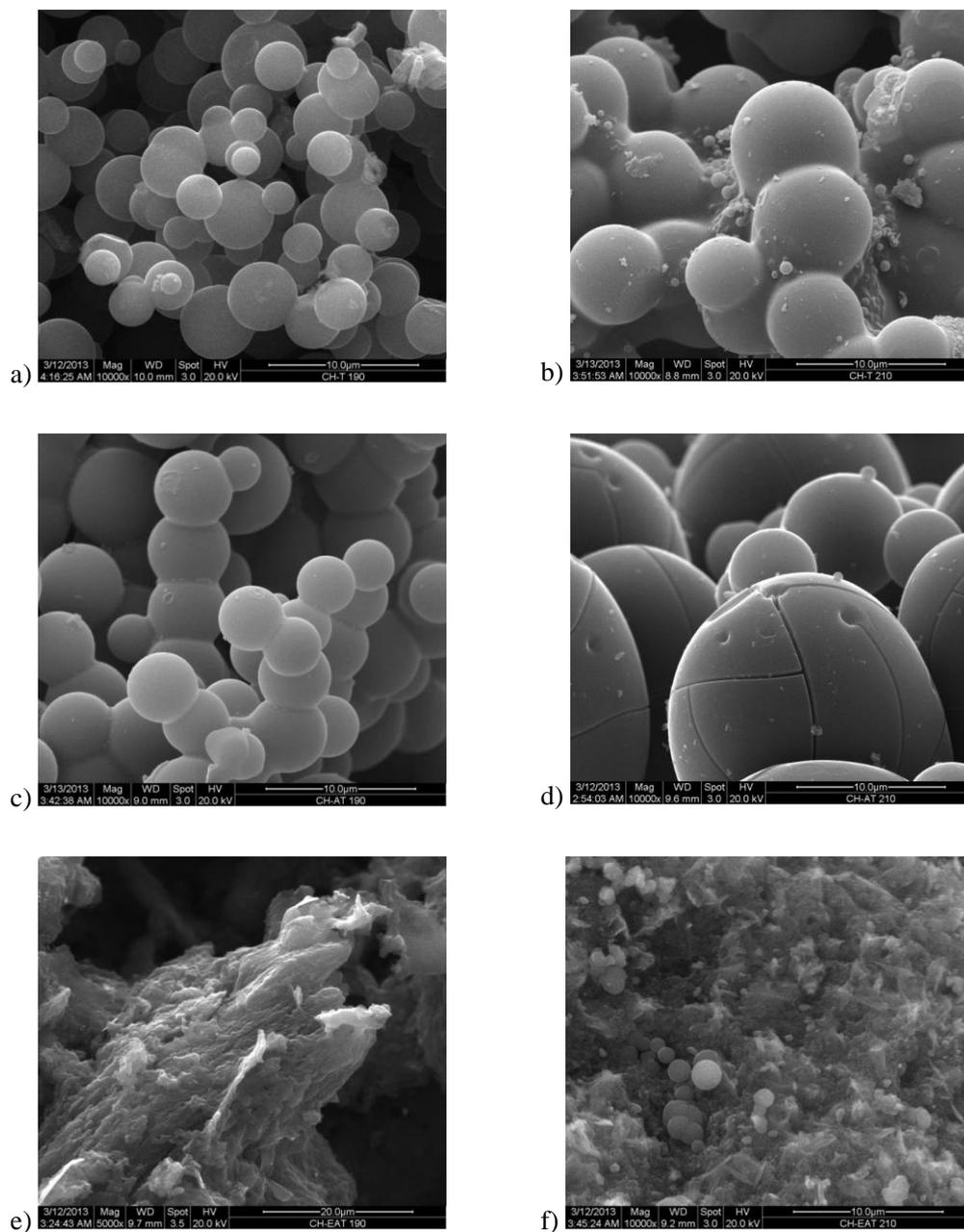


Fig. 4: SEM images of tannin-based materials after HTC and posterior pyrolysis at 900°C: (a) CH-T_190°C and (b) CH-T_210°C; (c) CH-AT_190°C and (d) CH-AT_210°C; (e) CH-EAT_190°C and (f) CH-EAT_210°C.

High surface - highly N-doped carbons from hydrothermally-treated tannin

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Figure 5

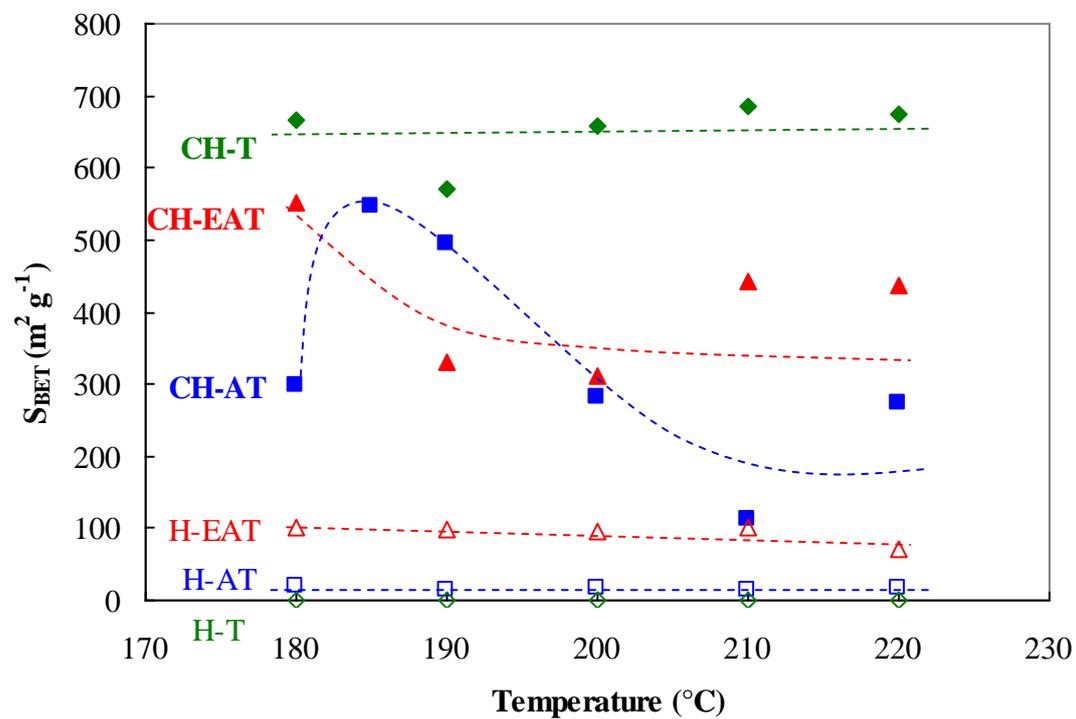


Fig. 5: Surface areas, S_{BET} , of all materials obtained after HTC at different temperatures: before (H-T \diamond , H-AT \square and H-EAT \triangle), and after pyrolysis at 900 $^{\circ}\text{C}$ (CH-T \blacklozenge , CH-AT \blacksquare and CH-EAT \blacktriangle).

High surface - highly N-doped carbons from hydrothermally-treated tannin

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Figure 6

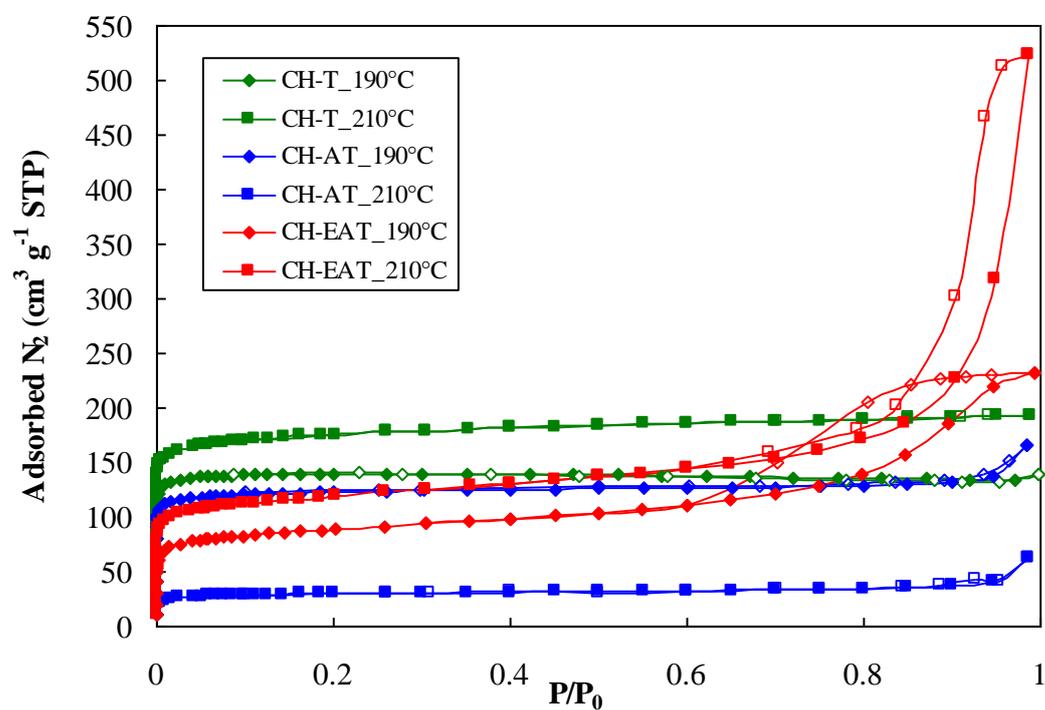


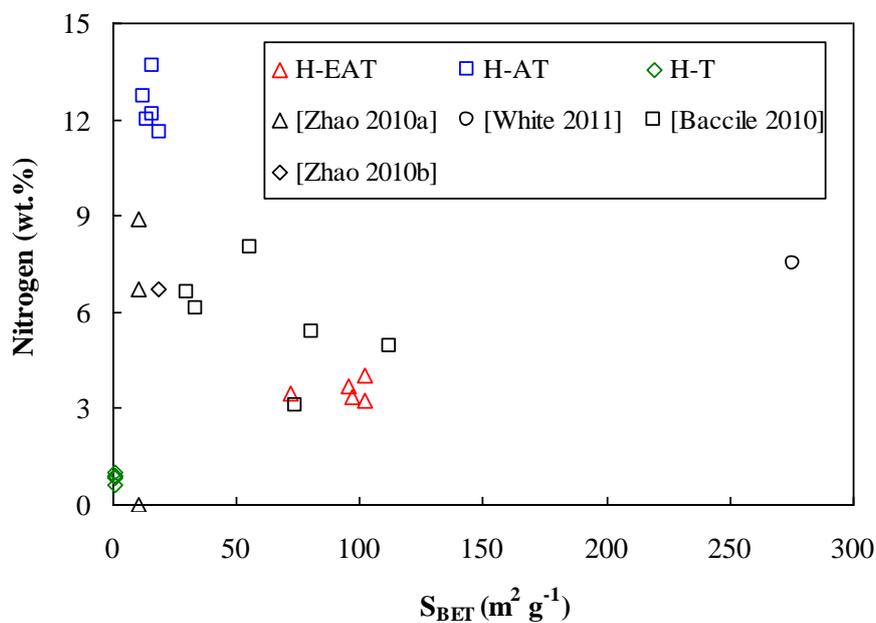
Fig. 6: Nitrogen adsorption (full symbols) – desorption (empty symbols) isotherms at -196°C. (CH-T_190°C ◆, CH-T_210°C ■; CH-AT_190°C ◆, CH-AT_210°C ■, CH-EAT_190°C ◆, CH-EAT_210°C ■).

High surface - highly N-doped carbons from hydrothermally-treated tannin

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Figure 7

a)



b)

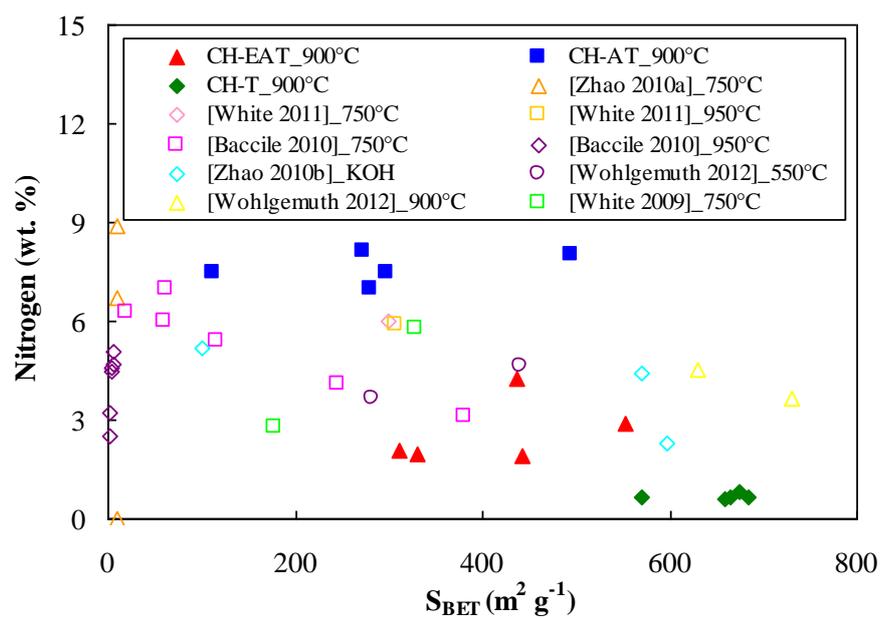


Fig. 7: Surface areas, S_{BET} , and nitrogen content of our materials: (a) before pyrolysis

(H-T \diamond , H-AT \square and H-EAT \triangle); (b) after pyrolysis (CH-T \blacklozenge , CH-AT \blacksquare and CH-

EAT \blacktriangle) compared to results reported in literature.

e-component

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