

1 Can calcium aluminates activate ternesite hydration?

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7 Abstract

8
9 Aluminium hydroxide (AH_3) has recently been shown to be able to activate hydration in
10 ternesite, a phase found in some calcium sulfoaluminate (CSA) cements.

11 This study explored the capacity of a number of calcium aluminates (C_3A , $C_{12}A_7$, CA
12 and $C_4A_3\bar{S}$) to activate ternesite hydraulic reactivity. After laboratory synthesis, the
13 aluminates were blended with ternesite at a ratio of 1:2 and their hydration was
14 monitored with isothermal conduction calorimetry for 7 days at 25 °C. The resulting
15 pastes were analysed with XRD, FTIR and DTA. The presence of ternesite in the pastes
16 altered the aluminate heat flow curves, shortening the induction period and bringing the
17 reaction peak forward, an indication of hastened hydration. Ternesite also altered the
18 reaction products, which included calcium monosulfoaluminate hydrate and strätlingite.

19 20 Keywords:

21
22 Ternesite, Calcium Aluminates, Hydration Products, Hydration Heat

23 24 25 1. Introduction

26

27 The mass production of portland cement (OPC) entails the consumption of vast amounts
28 of raw materials and accounts for 4 % to 6 % of worldwide anthropogenic CO₂
29 emissions (0.8-0.9 t CO₂/t clinker) [1].

30 The cement industry is currently seeking to develop less energy-intensive, lower
31 Greenhouse gases (GHG) emission cements. Calcium sulfoaluminate (CSA) cements,
32 one of the most promising alternatives, are characterised by lower energy costs (at
33 1250 °C, the clinkering temperature is 200 °C lower than in OPC clinker) and GHG.
34 These products emit from 25 % to 40 % less CO₂ during manufacture than OPC,
35 depending on the composition [2]. Whilst not presently used for structural purposes due
36 to the lack of the necessary regulation, they are commercialised as components for
37 special mortars and concretes (such as repair or self-levelling concrete, sealing mortar
38 or shotcrete).

39 The composition of the clinker in these cements may differ, although all lie in the CaO-
40 Al₂O₃-SiO₂-Fe₂O₃-CaSO₄ system and have lower calcium and silicon and higher
41 aluminate and sulfate phase contents than portland cements. The minerals present in the
42 clinker include C₄A₃ \bar{S} , C₂S, C₁₂A₇, CA, Cs and C₄AF*, although the aluminates
43 predominate [3,4].

44 Ternesite, C₅S₂ \bar{S} , which also lies within this system, is a phase compatible with
45 ye'elemite (C₄A₃ \bar{S}) and C₂S. It is found primarily in the crust covering the areas of
46 portland cement kilns where the temperature is no higher than 1250 °C and as a
47 minority component in some CSA cements [5, 6, 10]. A calcium sulfosilicate, ternesite
48 is formed in the reaction between anhydrite and belite. Bullerjahn et al. [6, 7] proposed
49 a two-stage method for ternesite formation, in which clinker is synthesised at a
50 temperature of 1250 °C and then cooled by gradually ramping the temperature down to

51 800 °C. Further to recent thermodynamic predictions and empirical observation, if
52 atmosphere (partial pressure of SO₂ and O₂) and temperature are controlled, belite- and
53 ternesite-high sulfoaluminate clinkers can be readily produced in a single-stage process
54 at temperatures of over 1200 °C [8].

55 Although ternesite has traditionally been regarded as hydraulically inactive and of no
56 technological interest [9], recent studies have shown that it is activated by amorphous
57 AH₃. It hydrates, then, in their presence, yielding ettringite and C-S-H respectively the
58 main hydration products in CSA and portland cements and the phases to which they
59 owe their mechanical strength and durability. Strätlingite may also form with or instead
60 of C-S-H [11].

61 Solubility of amorphous aluminum hydroxide in water at 25°C is very low ([Al] ≈
62 1,607x10⁻⁷ M/l and pH 6,789; database used: concrete_3T_V07_02 and calculated with
63 Phreeqc), it depends on the pH of medium and it rises both when pH increases or
64 decreases from neutrality, being Al(OH)₄⁻ the only ionic specie present at pH higher
65 than ≈ 8,5 [11]. Solubility of ternesite is not known, but it seems that its dissolution
66 produces hydrolysis and a pH near to 12 [12].

67

68 The solution in contact with ternesite and AH₃, contains Al(OH)₄⁻, Ca²⁺, SO₄²⁻, SiO₃H⁻,
69 OH⁻ ions that will react when the solubility products of ettringite or strätlingite or
70 calcium monosulfoaluminate hydrate or or C-S-H etc, are achieved; the precipitation of
71 the said phases will consume ions in solution stimulating then new ternesite
72 solubilisations.

73 Given the low water solubility of AH₃, other, more soluble aluminates might be thought
74 to act as activators and stimulate ternesite hydration more efficiently.

75

76 Such aluminates and ternesite might be the main phases in a future CSA cement with a
77 composition lying within the $\text{CaO-Al}_2\text{O}_3\text{-SiO}_2\text{-Fe}_2\text{O}_3\text{-CaSO}_4$ system.

78 The purpose of this study is to explore the capacity of a series of calcium aluminates
79 (C_3A , C_{12}A_7 , $\text{C}_4\text{A}_3\bar{\text{S}}$ and CA) to activate ternesite hydration, establishing the reactions
80 involved and identifying the reaction products.

81

82

83 **2. Experimental**

84

85 Ternesite ($\text{C}_5\text{S}_2\bar{\text{S}}$), ye'elemite ($\text{C}_4\text{A}_3\bar{\text{S}}$), C_3A , C_{12}A_7 and CA were synthesised from
86 stoichiometric blends of laboratory-grade CaCO_3 , SiO_2 , Al_2O_3 and gypsum
87 ($\text{CaSO}_4 \cdot 2\text{H}_2\text{O}$). The samples were weighed, ground in a ceramic mortar, homogenised
88 in an ethanol medium and dried at 100 °C. A pressure of 300 kp/cm^2 was applied to the
89 powder obtained to prepare 2 cm diameter cylindrical pellets weighing approximately
90 4 g. The pellets were calcined at 1200 °C ($\text{C}_5\text{S}_2\bar{\text{S}}$), 1250 °C ($\text{C}_4\text{A}_3\bar{\text{S}}$), 1350 °C (C_{12}A_7
91 and CA) or 1400 °C (C_3A) for 2 h. They were subsequently ground, homogenised, dried
92 and new pellets were prepared and calcined at the same temperatures, ternesite and
93 ye'elemite for 6 h and the other calcium aluminates for 5 h. This process was repeated
94 through conclusion of the synthetic reactions.

95 Particle size distribution in the synthesised phases was determined and recorded on a
96 Malvern Mastersizer S particle size analyser with 632.8 nm He-Ne laser optics after
97 subjecting samples suspended in ethanol with 5 drops of dispersant (DOLAPIX CE64
98 1/100) to ultrasonic dispersion for 5 min. The findings are given in Figure 1 and
99 Table 1.

100 Phase purity was determined by applying Rietveld refinement to the X-ray diffraction
101 (XRD) patterns (Table 2).

102 Aluminates and ternesite blend hydration (aluminate:ternesite=1:2 by mass) were
103 monitored for 7 d at 25 °C on a Thermometric TAM Air isothermal conduction
104 calorimeter. Three g of sample were stirred manually with 2.1 g of water (liquid/solid
105 ratio = 0.7) for 3 min prior to insertion in the calorimeter. Upon finalisation of the
106 calorimetric tests, the samples were submerged in acetone to detain hydration and
107 vacuum drying was used to remove water. The products were characterised with X-ray
108 diffraction (XRD), IR spectroscopy (FTIR) and differential thermal analysis-
109 thermogravimetry (DTA-TG).

110 The mineralogy of the synthetic phases and hydrated samples was determined on a
111 Bruker D8 Advance X-ray diffractometer, fitted with a high voltage, 3 kW generator
112 and a 1.54 Å CuK α anode X-ray tube operating at 40 kV and 50 mA. This instrument
113 was coupled to a Lynxeye detector with a 3 mm anti-scatter slit and a 0.5 % Ni K-beta
114 filter, with no monochromator.

115 XRD patterns of synthesized aluminates and ternesite were recorded at 5°-70° 2 θ
116 angles, with a step size of 0.01973°, and a 2 s step time. Rietveld quantitative analysis
117 [13] was conducted on the XRD findings using GSAS software [14].

118 XRD patterns were recorded for the ternesite-bearing hydrated aluminates, likewise at
119 5°-70° 2 θ angles with a step size of 0.01973°, while varying the step time to 6 s-10 s.
120 Corundum (Al₂O₃) was added to these samples at a rate of 25 % to quantify the
121 amorphous phase [15].

122 XRD scans were taken of the aluminate hydrated samples 5°-60° 2 θ at a step size of
123 0.019746° and a 0.5 s step time.

124 $C_{12}A_7$ +ternesite hydration was also XRD-monitored for 20 h: 5° - 30° 2θ angles, step
125 size 0.019746° and step time 0.5 s. Scans were recorded every 10 min during the first
126 10 h, and every 30 min from the 11th through the 20th hours. The paste was introduced
127 in an air-tight sample holder to prevent carbonation and drying.
128 For FTIR analysis of the 7 d pastes, vacuum pressure of 10 atm was applied for 3 min to
129 pellets prepared with approximately 1 mg of sample in 300 mg of KBr. FTIR scans
130 were performed at frequencies of 4000 cm^{-1} - 400 cm^{-1} on a Thermo Scientific Nicolet
131 600 FTIR spectrometer with a spectral resolution of 4 cm^{-1} . The CA and $C_{12}A_7$ pastes
132 were also scanned after heating to 188°C at $10^\circ\text{C}/\text{min}$ in an N_2 atmosphere.
133 Thermogravimetric (TG) and differential thermal (DTA) analyses were conducted on a
134 TA Instruments Q600 TGA-DCS-DTA analyser. The samples were heated from
135 ambient temperature to 1000°C in a nitrogen atmosphere ($100\text{ mL}/\text{min}$) at a rate of
136 $10^\circ\text{C}/\text{min}$.

137

138

139 **3. Results**

140

141 *3.1. Isothermal conduction calorimetry*

142

143 The heat flow curve for ternesite alone contained only the first exothermal peak, which
144 after 1 h had declined to $31\ \mu\text{W}/\text{g}$, given its scarce reactivity with water [6], the total
145 heat of this reaction was very low (Figure 2 and Table 3). During hydration at 25°C , the
146 heat flow curves for both the synthesised aluminates (C_3A , $C_{12}A_7$, CA, $C_4A_3\bar{S}$) and their
147 blends with ternesite (Figure 3 and Table 3) showed an early exothermal reaction after
148 mixing followed by a decline in the heat production rate and, in the $C_{12}A_7$, CA and

149 $C_4A_3\bar{S}$ samples, a second exothermal reaction that appeared earlier in the presence of
150 ternesite.

151 The intensity of the initial exothermal signal in the C_3A heat flow curve tended
152 downward for the first 7.5 h (6.6 mW/g), steadied until 10 h, continued downward until
153 $t=24$ h (0.75 mW/g) and then remained flat through the rest of the test. Gismera-Diez et
154 al. [16] reported the same pattern. In the presence of ternesite the decline was steeper
155 and uninterrupted through 13.5 h (0.08 mW/g), after which no thermal signals were
156 recorded.

157 The heat flow curves for pastes $C_{12}A_7$ and $C_{12}A_7$ +ternesite contained two exothermal
158 peaks following the very intense initial peak and an intermediate induction period. The
159 first appeared after 1 h in both whereas the second was recorded 2.5 h earlier in the
160 sample with than in the one without ternesite. A new, low intensity but long-lasting (9 h
161 to 20 h) exothermal signal appeared in the sample containing ternesite. Raab et al. [17]
162 regarded C_2AH_8 as the first product of $C_{12}A_7$ hydration, giving rise to the first visible
163 peak on calorimetric graphs. The second peak would denote a further hydration
164 reaction.

165 Induction began very quickly on the CA and CA+ternesite paste flow curves, after
166 approximately 2 h (the second peak observed in the $C_{12}A_7$ pastes was not found here),
167 and ended much earlier in the sample with ternesite (2.5 h and 15 h), whilst the highest
168 reaction peak for the blended sample was recorded at 4.5 h, compared to 21 h for the
169 sample with CA only. In addition to the aforementioned signals, the curve for paste CA
170 showed a slight rise in heat flow between 11 h and 13 h (1 mW/g), while the slope on
171 the CA+ternesite curve changed twice on the descending leg of the second heat flow
172 peak, at around 11 h (1.8 mW/g) and 17 h (0.8 mW/g).

173

174 Similar behaviour was observed for the paste $C_4A_3\bar{S}$ heat flow curves. The presence of
175 ternesite barely altered the beginning of the induction period (1.5 h) but changed the end
176 time: the second signal peaked at 4 h, 3 h earlier than in the unblended sulfoaluminate
177 paste. The $C_4A_3\bar{S}$ curve exhibited a slight rise in heat flow between 4 h (0.39 mW/g)
178 and 6 h (2 W/g), in addition to the aforementioned signals. This pattern was also
179 observed by Sánchez-Herrero et al. [18], who reported a first peak at around 4 h and a
180 second and more intense peak at 6.5-7 h. The graph for the ternesite-bearing paste
181 contained a low intensity exothermal signal with a peak at 1 h after the very intense
182 initial dissolution peak, as well as a change in the slope on the descending leg of the
183 second heat flow peak at around 9.5 h.

184 In the absence of ternesite, the total heat released by the 7 d, 25 °C samples was as
185 follows: $C_3A > CA \approx C_{12}A_7 > C_4A_3\bar{S}$. In the presence of the said mineral, the order was
186 $CA > C_{12}A_7 > C_3A > C_4A_3\bar{S}$ (Figure 4 and Table 4). Although the samples containing
187 ternesite exhibited much lower heat of hydration than the unblended aluminate pastes,
188 the total heat per gramm of aluminate increases between 19 and 91%.

189
190

191 3.2. XRD study

192

193 CA, $C_{12}A_7$ and C_3A hydration yields C_3AH_6 , the sole stable calcium aluminate hydrate
194 in the CaO- Al_2O_3 - H_2O system, as well as a number of metastable phases (CAH_{10} ,
195 C_2AH_8 , $C_4A \cdot xH$ ($x=7, 11, 13$ or 19), $C_2A \cdot xH$ ($x= 4, 5, 7.5$ or 8)) depending on the
196 temperature and composition of the aluminate hydrated.

197 The diffractograms for the 7 d (25 °C) C_3A and $C_3A/ C_5S_2\bar{S}$ (1:2 by mass) blended
198 pastes are reproduced in Figure 5. As expected, the product of C_3A hydration was
199 katoite (C_3AH_6) [14, 16, 19]. Calcium hemicarboaluminate hydrate, ternesite, katoite,

200 larnite, $C_3A \cdot C\bar{S} \cdot 12H$ (AFm) and portlandite traces, were identified on XRD pattern of
201 C_3A +ternesite paste. The larnite was sourced from the original synthetic ternesite
202 (6% C_2S). The presence of portlandite traces denoted some minor larnite hydration
203 during the test.

204
205 The diffractogram for 7 d (25 °C) $C_{12}A_7$ paste contained reflections attributed to C_2AH_8 ,
206 CAH_{10} , gibbsite (very low intensity), and katoite, as well as the anhydrous phase. Two
207 unidentified, low intensity lines at $d=3.48 \text{ \AA}$ and $d=2.61 \text{ \AA}$ were also visible on the
208 pattern. The presence of ternesite altered $C_{12}A_7$ hydration products. Instead of the
209 reflections observed on the pattern for $C_{12}A_7$ paste, the respective diffractogram
210 contained lines generated by strätlingite, larnite, calcium monosulfoaluminate hydrate
211 and ternesite (Figure 6).

212
213 The diffractogram for the 7 d (25 °C) CA paste exhibited reflections attributed to CAH_{10}
214 and C_2AH_8 , gibbsite and katoite (low intensity), as well as to anhydrous CA. These
215 findings were consistent with observations reported by Lothenbach et al. [21, 22]. As
216 the most intense diffraction lines for the metastable phases C_2AH_8 and C_4AH_{19} concur
217 at $d=10.7 \text{ \AA}$ and 5.36 \AA , they cannot be readily differentiated with XRD, even though
218 the relative intensities vary somewhat. These phases are generated by dissolved ions,
219 whereas the metastable hydrates in the $C_2A \cdot xH$ and $C_4A \cdot xH$ series are obtained by
220 dehydration of the ions carrying the most water.

221 The shift observed here in the main reflections for C_2AH_8 ($d=10.7 \text{ \AA}$ and 5.36 \AA) to
222 slightly lower values ($d=10.4 \text{ \AA}$ and 5.2 \AA) on the diffractograms for pastes $C_{12}A_7$ and
223 CA may denote slight sample dehydration due to their exposure to acetone and vacuum
224 drying to detain hydration and remove water [23].

225 As in $C_{12}A_7$, the presence of ternesite altered CA hydration products. The diffractogram
226 for the blended paste contained none of the reflections observed on the pattern for CA
227 paste, although lines indicative of strätlingite, calcium monosulfoaluminate hydrate,
228 larnite and ternesite were identified (Figure 7).

229
230 The $C_4A_3\bar{S}$ hydration products identified by XRD were AFm and gibbsite (Figure 8), a
231 finding consistent with prior reports [24]. The hemicarboaluminate identified was
232 deemed to be due to sample weathering. Part of the $C_4A_3\bar{S}$ was still anhydrous after
233 7 days of hydration at 25 °C. The presence of ternesite appeared to stimulate $C_4A_3\bar{S}$
234 dissolution, inasmuch as no reflections for this phase were observed on the
235 diffractogram. Here AFm was the reaction product, while no reflections from crystalline
236 aluminium hydroxide was observed.

237 The mineralogical analysis of the hydrated ternesite-containing aluminates based on
238 quantitative XRD is given in Table 5.

239

240 3.3. *In-situ* XRD

241

242 The reactions taking place over time in paste $C_{12}A_7$ +ternesite generated a heat flow
243 curve with a number of signals. These reactions were identified by continuous XRD
244 scanning during the first 20 h of hydration (Figure 9). The intensity of the reflections
245 attributed to C_2AH_8 rose over time, whilst the intensity of the lines for $C_{12}A_7$ and
246 ternesite declined.

247 The decline in intensity of the main diffraction lines for $C_{12}A_7$ and ternesite concurred
248 in time with the first exothermal peak on the calorimetric curve, an indication that these
249 phases dissolved very early into hydration. C_2AH_8 was identified as the first hydrate,
250 generated at 10 min, along with ϕ - AH_3 (Equation 1). After 1 h (small exothermal peak

251 on the calorimetric curve), the intensity of the diffraction lines for $C_{12}A_7$ and ternesite
 252 declined again, while that of the C_2AH_8 reflections continued to rise. The two reagents
 253 declined throughout the induction period. A new phase, $CaAl_2Si_7O_{18} \cdot 3.5H_2O$, appeared
 254 after 4.5 h through the end of the test (20 h) as a product of the reaction between
 255 ternesite and $\varphi-AH_3$. Phase $\varphi-AH_3$, which was identified from the outset, began to
 256 gradually decline after 2 h and disappeared entirely after 5 h, as strätlingite and
 257 $CaAl_2Si_7O_{18} \cdot 3.5H_2O$ formed.

258



259

260 The strätlingite reflections (Equation 2) were identified after 5 h, as C_2AH_8 continued to
 261 rise and ternesite and $C_{12}A_7$ to decline as a result. The absence of reflections for new
 262 crystalline sulfates may be explained either by their low crystallinity (with the probable
 263 formation of AFm nuclei) or the fact that the sulfate ion resulting from ternesite
 264 decomposition remains in solution after strätlingite crystallisation.

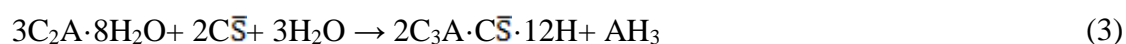
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266

267 Calcium monosulfoaluminate hydrate (Equation 3) appeared at around 9 h-10 h,
 268 however, concurring with the low intensity, long-lasting exothermal shoulder on the
 269 calorimetric curve.

270



271

272 The same sample was scanned after storage for 7 d in a moist environment at 25 °C. The
273 diffractogram showed higher intensity strätlingite reflections and AFm phases, whilst no
274 lines for the intermediate phase, $\text{CaAl}_2\text{Si}_7\text{O}_{18}\cdot 3.5\text{H}_2\text{O}$, were observed. The intensity of
275 the ternesite and C_2AH_8 reflections was low, although the latter was still visible. These
276 findings differed in some respects from the 7 d (25 °C) calorimetric results, where large
277 amounts of unreacted ternesite were observed and no C_2AH_8 was identified.
278 The explanation for the differences between the XRD findings for the sample hydrated
279 for 7 d in the calorimeter and the one hydrated for 20 h in the diffractometer and
280 subsequently stored at 25 °C for 6 d more, might be that in the latter, the incidence of
281 the X-rays on the sample may have induced a temperature substantially higher than 25
282 °C. That may have given rise to different phases than obtained at a constant 25 °C and
283 would also explain why the ternesite was largely consumed in 20 h in the diffractometer
284 whereas a large fraction of the mineral had not reacted after 7 d at 25 °C in the
285 calorimeter.

286

287 3.4. FTIR study

288 The FTIR analyses (Table 6) confirmed the XRD results. The infrared spectrum for
289 paste C_3A exhibited the absorption lines characteristic of katoite, some (in the 790 cm^{-1} -
290 820 cm^{-1} and 520 cm^{-1} - 530 cm^{-1} ranges) overlapping with the signals for the remaining
291 anhydrous C_3A [16]. Slight carboaluminate contamination was likewise observed.
292 The most intense bands on the FTIR for paste $\text{C}_4\text{A}_3\bar{\text{S}}$, in the O-H stretching zone
293 (4000 cm^{-1} - 3000 cm^{-1}) were generated by overlapping AFm and gibbsite signals [18].
294 On the rest of the spectrum the bands for these two hydrates were observed to overlap

295 with those of anhydrous $C_4A_3\bar{S}$ and, to a lesser extent, calcium hemicarboaluminate
296 hydrate (Hc).

297 The spectra for pastes CA and $C_{12}A_7$ exhibited similar bands in the 4000 cm^{-1} -
298 2500 cm^{-1} range, although they were sharper in the former, where they were attributed
299 to gibbsite overlapping with katoite, CAH_{10} and C_2AH_8 . Overlapped bands generated by
300 CAH_{10} , C_2AH_8 (more intense in $C_{12}A_7$) and gibbsite (much more intense in CA)
301 appeared in the 1300 cm^{-1} - 400 cm^{-1} zone. Some of the bands on these spectra were
302 identified by Fernández-Carrasco et al. [25]. In order to identify more clearly the phases
303 present, these samples were thermally treated at 188°C , temperature at which CAH_{10}
304 phase is decomposed and C_2AH_8 partially lose the water.

305 After the two pastes were heated to 188°C , the FTIR spectra changed radically: the
306 CAH_{10} bands disappeared and the C_2AH_8 bands were altered due to a partial water loss.
307 The gibbsite bands intensified on the paste CA spectrum and nearly disappeared in
308 $C_{12}A_7$, where the signals for katoite and partially dehydrated C_2AH_8 intensified. These
309 findings infer that the two pastes contained the same hydrates, albeit in different
310 proportions. CA had more gibbsite and less katoite and C_2AH_8 than $C_{12}A_7$.

311 The bands attributable to ternesite stood out on the spectra for the four aluminate pastes
312 containing the mineral. They appeared in all the samples at 1153 cm^{-1} , 1121 cm^{-1} ,
313 947 cm^{-1} , 878 cm^{-1} , 837 cm^{-1} , 660 cm^{-1} , 630 cm^{-1} , 602 cm^{-1} , 541 cm^{-1} and 517 cm^{-1} [26],
314 i.e., in the 1300 cm^{-1} - 400 cm^{-1} range, where no other anhydrous phase absorptions were
315 observed. All four spectra exhibited a narrow, intense absorption signal at 424 cm^{-1} and
316 other wider, overlapping signals at around 520 cm^{-1} and 540 cm^{-1} , characteristic of a
317 number of calcium aluminate hydrates. The ternesite spectrum (multiplied by a factor of
318 0.4-0.8) was subtracted from the original paste spectra to better identify the hydrates.

319 The spectra for paste C_3A +ternesite (Figure 10) exhibited bands in the O-H stretching
320 vibrations zone mainly generated by the presence of Hc overlapped with bands from
321 AFm and portlandite while the O-H stretching band from katoite was not clearly
322 distinguished. After subtracting ternesite spectra, the most prominent bands in the 1300-
323 400 cm^{-1} range were attributed to AFm (at $\approx 1110, 784, 539, 424 \text{ cm}^{-1}$) and Hc.

324 The spectrum for paste $C_{12}A_7$ +ternesite (Figure 11) exhibited O-H stretching vibrations
325 generated by water and hydrates, attributed to AFm (3670 cm^{-1}), as well as a wide band
326 with two peaks, at 3615 cm^{-1} and 3530 cm^{-1} (possibly indicative of a weathering-
327 induced hemicarboaluminate). The bands remaining after the ternesite spectrum was
328 subtracted, all in the 1300 cm^{-1} - 400 cm^{-1} range, were attributable to C_2ASH_8
329 ($1013 \text{ cm}^{-1}, 964 \text{ cm}^{-1}, 710 \text{ cm}^{-1}$) or AFm ($780 \text{ cm}^{-1}, 572 \text{ cm}^{-1}, 539 \text{ cm}^{-1}$). A band
330 overlapping generously with ternesite attributable to AFm was observed in the sulfate ν_3
331 zone.

332 The spectra for paste CA+ternesite (Figure 12) were very similar to the ones for
333 $C_{12}A_7$ +ternesite, with O-H stretching vibrations from water and hydrates attributed to
334 AFm (3670 cm^{-1}), although an intense band appeared at 3622 cm^{-1} , along with a wide
335 band at 3559 cm^{-1} - 3487 cm^{-1} , which overlapped with another at 3550 cm^{-1} , which could
336 be assigned to AH_3 . The bands observed in the 1300 cm^{-1} - 400 cm^{-1} range on the post-
337 subtraction spectrum at around $1013 \text{ cm}^{-1}, 964 \text{ cm}^{-1}$ and 710 cm^{-1} , attributable to
338 C_2ASH_8 , were much more intense than on the $C_{12}A_7$ spectrum. Less intense bands
339 generated by AFm were found at $780 \text{ cm}^{-1}, 572 \text{ cm}^{-1}$ and 539 cm^{-1} .

340 The spectra for paste $C_4A_3\bar{S}$ +ternesite (Figure 13) also contained O-H stretching
341 vibrations from water and hydrates that concurred with the signals for gibbsite, although

342 their relative intensities differed. The most prominent band in the 1300 cm^{-1} - 400 cm^{-1}
343 range on the spectrum resulting from subtracting the ternesite bands, at 1020 cm^{-1} , was
344 characteristic of gibbsite, which had not been identified by XRD. Signals attributed to
345 AFm were observed at 780 cm^{-1} , 582 cm^{-1} and 539 cm^{-1} .

346

347 3.5. DTA-TG study

348

349 The water content in 7 d aluminate pastes was: $\text{CA} > \text{C}_{12}\text{A}_7 > \text{C}_4\text{A}_3\bar{\text{S}} > \text{C}_3\text{A}$. The order for
350 the pastes containing ternesite (normalised per g of dry aluminate), which held less
351 water, was $\text{CA} > \text{C}_{12}\text{A}_7 > \text{C}_3\text{A} > \text{C}_4\text{A}_3\bar{\text{S}}$ (Figure 14).

352

353 Assigning thermal signals to calcium aluminate hydrate dehydration/dehydroxylation in
354 poly-phase samples is a complex task, for each phase holds various forms of bound
355 water, which may be gradually released with rising heat. As a result, each phase
356 generates several endothermal signals in the same temperature ranges, which explains
357 the lack of uniformity in the literature cited in Table 7.

358

359 The most prominent, i.e., most intense and sharpest, of the endothermal dehydration and
360 dehydroxylation peaks for the hydrates in the pastes studied was attributed to katoite, at
361 $300\text{ }^\circ\text{C}$ on the DTA curve for paste C_3A . This peak was also observed, vaguely and
362 overlapping with gibbsite, on the curves for pastes CA and C_{12}A_7 .

363 Further to the TG findings, hydrated paste C_3A contained 79 wt% katoite and 21 wt%
364 anhydrous C_3A . In other words, the hydration rate in C_3A was 0.73.

365 The DTA curves for pastes CA and C_{12}A_7 (Figure 15) exhibited several thermal signals
366 at temperatures of under $200\text{ }^\circ\text{C}$ and in the $200\text{ }^\circ\text{C}$ - $600\text{ }^\circ\text{C}$ range, all associated with
367 weight loss on the TG/DTG curves (Figure 14). In the first range of temperatures the

368 intense endothermal peaks at around 80 °C-88 °C were attributed to amorphous AH₃
369 [33]. They overlapped with a peak at around 100 °C-110 °C, generated by water loss
370 from CAH₁₀ and the water physically adsorbed on the C₂AH₈ interlayer. Another
371 endothermal signal at around 145 °C-160 °C could be attributed to further partial water
372 loss from C₂AH₈ [35]. In the second temperature range, the DTA curves for both pastes
373 exhibited endothermal signals of varying intensity at around 260 °C, 280 °C and 320 °C,
374 attributed to water loss from C₂AH₈, AH₃ and C₃AH₆. Two much less intense signals at
375 421 °C and 524 °C were also attributed to those phases.

376
377 The DTA curves for sample C₄A₃S (Figure 16) contained very intense <200 °C signals:
378 at 120 and 180 °C for AFm and 120 °C overlapped with hemicarboaluminate, the latter
379 due to sample weathering. Three peaks were observed at 200 °C-300 °C, the most
380 intense at 264 °C and the others at 224 °C and 286, which might be due to AFm, Hc and
381 gibbsite overlapping. The peak at 286 °C was associated with gibbsite water loss.

382
383 C₃A+ternesite curve (Figure 17) contained an intense broad (120-180°C) endothermal
384 signal, peaking at around 154 °C and other broad one at 240 °C, denoting water loss
385 from Hc and AFm. The signal at 285 °C was generated by Katoite and the final one at
386 412°C by portlandite, as per Torrén-Martín et al. [39].

387 Curves for CA, C₁₂A₇ and C₄A₃S with ternesite pastes exhibited a very intense
388 endothermal signal at around 170 °C-180 °C generated by water loss from AFm phases.

389
390 The curves for samples CA+ternesite and C₁₂A₇+ternesite (Figure 18) had endothermal
391 signals of varying intensity at around 125 °C-128 °C, 170 °C and 220 °C, attributable to
392 strätlingite, according to Kuzel [36] and Matschei [40], and as regards the signal at
393 170 °C, also to AFm. The conclusion that could be drawn from a comparison of the

394 intensity of these signals is that CA+ternesite had a higher proportion of strätlingite and
395 paste C₁₂A₇+ternesite a higher proportion of AFm. That finding is consistent with the
396 FTIR and XRD results. These two pastes also exhibited a much lower intensity
397 endothermal signal at 285 °C that might be associated with AH₃ water loss, although the
398 XRD analyses failed to identify that phase.

399
400 In addition to the aforementioned AFm and hemicarboaluminate signals at 170 °C,
401 210 °C and 110 °C, the C₄A₃ \bar{S} +ternesite sample exhibited others attributable to gibbsite
402 overlapped with AFm, which generated two peaks, at 250 °C and 290 °C (Figure 19).

403

404 **4. Discussion**

405

406 The calorimetric curves for the hydrated aluminate pastes contained a first dissolution
407 peak, followed by an induction period and an intense reaction peak in pastes CA, C₁₂A₇
408 and C₄A₃ \bar{S} [17, 18, 21] The presence of ternesite in the pastes altered the calorimetric
409 curves in all cases, shortening the induction period and bringing the reaction peak
410 forward.

411 The pastes containing C₁₂A₇ had a much more active induction period than any of the
412 others, with a heat flow a full order or magnitude higher. C₂AH₈ and ϕ -AH₃ formed
413 from the onset of hydration in pastes with ternesite, which reacted with ϕ -AH₃ in the
414 induction period, giving rise to an intermediate phase, CaAl₂Si₇O₁₈·3.5H₂O. In the main
415 exothermal peak concurred the consumption of all the ϕ -AH₃ and the first strätlingite
416 crystallization, while C₂AH₈ precipitation as well as reagents dissolution (ternesite and
417 C₁₂A₇) continued. The main reaction peak was followed by calcium
418 monosulfoaluminate hydrate precipitation. C₁₂A₇ can therefore be regarded to
419 powerfully activate ternesite, which dissolved from the outset, giving rise to calcium

420 silicoaluminate hydrate precipitation even prior to calcium monosulfoaluminate
421 precipitation.

422 Whilst the 7 d aluminate+ternesite (1:2) pastes released less total heat than the
423 analogous unblended aluminate pastes, however when total heat was expressed per unit
424 of mass of aluminate, the four samples containing ternesite exhibited higher values than
425 their unblended aluminate counterparts. The inference is that ternesite participate in the
426 hydration reactions and not only hastened but activated aluminate hydration. In the
427 presence of ternesite, the amount of heat per aluminate mass unit rose by 91 % in CA,
428 87 % in C₁₂A₇, 19 % in C₃A and 36 % in C₄A₃ $\bar{5}$ (Table 4).

429 Ternesite is activated when hydrated in the presence of C₃A, C₁₂A₇ and CA, accounting
430 for 43 wt% to 22 wt% of the respective pastes, according to XRD data. On the grounds
431 of the quantitative XRD findings in conjunction with the total mass loss at 1000 °C
432 determined with TG, 16 wt% of the ternesite reacted in C₃A, 50 wt% in CA and 53 wt%
433 in C₁₂A₇. Those results denote scant (2 %) ternesite reaction in the ternesite-ye'elemite
434 blended paste.

435 C₃A hydration, which was incomplete after 7 d (hydration degree = 0, 73), yielded
436 katoite (Equation 4). Further to the XRD findings, C₃A hydration in the presence of
437 ternesite produces a small amounts of katoite (5.36 wt%) calcium monosulfoaluminate
438 hydrate (1.76 wt%) and calcium hemicarboaluminate hydrate (6.91 wt%) as well as
439 larnite (1.71 wt%), ternesite (43.11 wt%), C₃A (0.72 wt%) as the main crystalline
440 phases and amorphous phases (40.42 wt%).



443 Calculus derivated from DRX an TG data reveal a scant reaction of ternesite (16,3 wt%)
444 while C₃A reacted almost completely (97 wt%). These data fit very well with the
445 increment of the total heat released per aluminate mass unit in this paste (Table 4).

446

447 Taken into account the loss of mass recorded by TG, it can be concluded that C₃A plus
448 ternesite paste contains 28,7 wt % of AFm+Hc; 11,5 wt% of C₃AH₆ and 1,6 wt% de
449 portlandita; these values are very much more higher than those determined by XRD,
450 which would be an indication of the poor crystallinity of the hydrates obtained.

451 Pastes C₁₂A₇ and CA gave rise to the same hydration products, CAH₁₀, C₂AH₈ and AH₃,
452 although in different proportions, in addition to the respective anhydrous aluminates. In
453 the presence of ternesite the anhydrous aluminates reacted fully, generating AFm and
454 C₂ASH₈ in both cases. No other calcium aluminate hydrate appeared as a crystalline
455 phase.

456 Combining TG and XRD data the amount of ternesite consumed in samples C₁₂A₇ +
457 ternesite and CA + ternesite, is respectively 53% and 50%; assuming that all the sulfate
458 of the dissolved ternesite forms AFm and all the remaining aluminum forms strätlingite,
459 the proportions of these phases was calculated which are shown in Table 8. Both
460 hydrates would be in much higher proportion than that shown by XRD, however the
461 results are consistent with the mass loss observed by TG, which would account for the
462 poor crystallinity of AFm and strätlingite in said pastes.

463 The hydration products for paste C₄A₃ \bar{S} included AFm (33 %) and AH₃ (43 %) (small
464 signal of hemicarboaluminate was also observed) (Eq.5).

465



466

467 The XRD and TG findings showed that in the presence of ternesite, around 3 wt % of
468 the ye'elemite remained unreacted. AFm appeared and AH₃ was identified by FTIR and

469 DTA, although not on the X-ray diffractograms, an indication of its amorphous nature.
470 These findings do not infer that ye'elemite acted as a ternesite activator, for no
471 strätlingite was identified. As noted earlier, further to calculations based on the ternesite
472 content (50.5 wt%), the total TG-determined mass loss (17.6 wt%) and the
473 ye'elemite:ternesite ratio, the sample contained 97.8 % of the ternesite initially added.
474 That was very likely due to the common sulfate ion effect, which delayed ternesite
475 dissolution. Under such circumstances, ternesite would affect the crystallinity of the
476 AH_3 formed (gibbsite in the ye'elemite paste and amorphous aluminium hydroxide in
477 ye'elemite+ ternesite paste) and the ye'elemite hydration rate (shortening the induction
478 period), but the amount of ternesite consumed along the reaction time was very low.
479 The calorimetric data nonetheless revealed a 36 % rise in the total heat per unit of
480 aluminate released, perhaps due to the filler effect.
481 The crystalline AFm of ternesite plus ye'elemite paste content was also very low,
482 inasmuch as 96 wt % of the ye'elemite reacted. The hydrate content determined with
483 XRD and the calculated using the degree of reaction for ternesite and ye'elemite is
484 given in Table 8. The table also lists the mass loss in the four pastes at 1000 °C
485 calculated from those data and from the TG findings.
486 As XRD identified crystalline AH_3 in the ternesite-free paste but not in the paste
487 bearing ye'elemite + TER, the presence of ternesite may be deduced to have affected
488 both AFm and AH_3 crystallinity.
489
490 The XRD-determined crystalline AFm content in the pastes was much lower than would
491 be expected on the grounds of the reaction degree of ternesite and aluminates. The
492 calculated loss of evaporable matter in the hydrates was much higher than calculated
493 with the XRD data and very close to the TG findings. A substantial portion of the AFm

494 may consequently be assumed to be scanty crystalline and included in the amorphous
495 phase determined in XRD patterns.

496

497

498 **5. Conclusions**

499

500 This study explored the possible activation of ternesite hydration by a series of
501 aluminates by recording heat of hydration and analysing the 7 d hydration products with
502 XRD, FTIR and DTA. The ultimate aim was to pave the way for the future development
503 of a new eco-clinker, in which one of the main minerals would be ternesite blended with
504 aluminates that would activate its hydration. The major conclusions are listed below.

505

506 - Ternesite was activated in all the blends, although with varying intensity depending on
507 the aluminate added. Aluminate efficacy was, in descending order, $C_{12}A_7 \approx CA$
508 $> C_3A >>>> C_4A_3\bar{S}$.

509

510 - The presence of ternesite in the pastes altered the aluminate heat flow curves,
511 shortening the induction period and bringing the reaction peak forward, an indication of
512 hastened hydration. The calorimetric findings revealed rises of 19 % to 91% in total
513 heat release per unit of aluminate mass, attesting to the activation of aluminate
514 hydration in the presence of ternesite.

515

516 - Hydration of C_3A produces Katoite as the only hydration product but in presence of
517 ternesite the katoite amount is hardly reduced while AFm and Hemicarboaluminate
518 were found.

519 - $C_{12}A_7$ and CA were analysed jointly, for they yielded the same hydration products.

520 These aluminates activated ternesite hydration and their pastes contained the same

521 phases (AFm and stratlingite), albeit in different proportions..

522

523 - Due to the common sulfate ion effect, much less ternesite (2 wt%) was consumed in

524 paste $C_4A_3\bar{S}$ than in any of the other samples. However ternesite alter the cristallinity of

525 the obtained hydrates; scantly crystalline AFm and amorphous AH_3 were the main

526 hydration products in this paste. As no strätlingite was obtained, ternesite could not be

527 regarded as having been clearly activated by ye'elemite, despite some calorimetric

528 evidence to that effect.

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530

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537

538 **References**

539 [1] O. Edenhofer, et al. IPCC, 2014: Climate Change 2014: Mitigation of Climate

540 Change. Contribution of Working Group III to the Fifth Assessment Report of the

541 Intergovernmental Panel on Climate Change (2014)

542 [2] Marta García Maté. Processing and Characterisation of calcium sulphoaluminate

543 (CSA) eco-cement with taylored performances. (2014) PhD University of Málaga

- 544 [3] M. Carmen Martín-Sedeño, Antonio J.M. Cuberos, Ángeles G. De la Torrea, Gema
545 Álvarez-Pinazo, Luis M. Ordóñez; Milen Gateshkić, Miguel A.G. Aranda. Aluminum-
546 rich belite sulfoaluminate cements: Clinkering and early age hydration. *Cement and*
547 *Concrete Research* Volume 40, Issue 3, March (2010) 359–369
- 548 [4] K. Morsli, A.G. de la Torre, M. Zahir, M.A.G. Aranda. Mineralogical phase analysis
549 of alkali and sulfate bearing belite rich laboratory clinkers. *Cem. Concr. Res.*, 37 (2007)
550 639–646
- 551 [5] M.Marroccoli, F.Montagnaro,M.L. Pace, A. Telesca, G.L. Valenti, Synthesis of
552 calcium sulfoaluminate cements from blends of coal combustion ashes with flue gas
553 desulfurization gypsum, PTSE, Ischia, Italy, 2010.
- 554 [6] Bullerjahn, F., Zajac, M. & Ben Haha, M. CSA raw mix design: effect on clinker
555 formation and reactivity. *Materials and Structures*, Volume 48, (2015) 3895-3911.
- 556 [7] Bullerjahn, F., Schmitt, D. & Ben Haha, M. Effect of raw mix design and of
557 clinkering process on the formation and mineralogical composition of (ternesite) belite
558 calcium sulphoaluminate ferrite clinker. *Cement and Concrete Research*, Volume 59,
559 (2014) 87-95.
- 560 [8] T. Hanein, I. Galan, F. P. Glasser, S.Skalamprinos, A. Elhoweris, M.S. Imbabi, M.N.
561 Bannerman, Stability of ternesite and the production at scale of ternesite-based clinkers
562 *Cement and Concrete Research* 98 (2017) 91–100
- 563 [9] Sherman, N., Beretka, J., Santoro, L. & Valenti, G. L. Long-term behaviour of
564 hydraulic binders based on calcium sulfoaluminate and calcium sulfosilicate. *Cement*
565 *and Concrete Research (CCR)*, Volume 25, (1995)113-126.
- 566 [10] .B. Pliego-Cuervo, F.P. Glasser, The role of sulphates in cement clinkering:
567 subsolidus phase relations in the system $\text{CaO}-\text{Al}_2\text{O}_3-\text{SiO}_2-\text{SO}_3$. *Cem. Concr. Res.*, 9
568 (1979) 51–56

569 [11] Mohsen Ben Haha, Frank Bullerjahn, Maciej Zajac “On the reactivity of
570 ternesite” XIV ICCO, Pekin 2015

571 [12] Peter K. Holt, Geoffrey W. Barton, Mary Wark, Cynthia A. Mitchell. A
572 quantitative comparison between chemical dosing and Electrocoagulation. *Colloids and*
573 *Surfaces A: Physicochem. Eng. Aspects* 211 (2002) 233-248.

574 [13] H.M. Rietveld, A profile refinement method for nuclear and magnetic structures, *J.*
575 *Appl. Crystallogr.* 2 (1969) 65–71.

576 [14] A.C. Larson, R.B. Von Dreele, General Structure Analysis System (GSAS)
577 program, Rep. No. LA-UR-86748, Los Alamos National Laboratory, Los Alamos, CA,
578 1994.

579 [15] M. A. G. Aranda, A. G. De la Torre. Rietveld Quantitative Phase Analysis of OPC
580 Clinkers, Cements and Hydration Products. *Reviews in Mineralogy & Geochemistry*
581 *Vol. 74 pp. 169-209 (2012)*

582 [16] S. Gismara-Diez, B. Manchobas-Pantoja, P.M. Carmona-Quiroga, M.T. Blanco-
583 Varela, Effect of BaCO₃ on C3A hydration Cement and Concrete Research, 73 (2015)
584 70-78.

585 [17] B. Raab, H. Poellmann, Heat flow calorimetry and SEM investigations to
586 characterize the hydration at different temperatures of different 12CaO·Al₂O₃ (C12A7)
587 samples synthesized by solid state reaction, polymer precursor process and glycine
588 nitrate process *Thermochimica Acta* 513 (2011) 106–111

589 [18] M.J. Sánchez-Herrero, A. Fernández-Jiménez, A. Palomo, C₄A₃S̄ hydration in
590 different alkaline media, *Cement and Concrete Research* 46 (2013) 41-49

591 [19] E. Breval, C3A Hydration, *Cement and Concrete Research*, 6 (1976)129-138

592 [20] J.I. Bhatta, A review of the application of thermal analysis to cement-admixture,
593 *Systems Thermochimica Acta* 189 (1991) 313-350

594 [21] S.R. Klaus, J. Neubauer, F. Goetz-Neunhoeffler, Hydration kinetics of CA₂ and
595 CA—Investigations performed on synthetic calcium aluminate cement, Cement and
596 Concrete Research 43 (2013) 62–69

597 [22] B. Lothenbach, L. Pelletier-Chaignat, F. Winnefeld, Stability in the system CaO–
598 Al₂O₃–H₂O, Cement and Concrete Research 42 (2012) 1621-1634

599 [23] T.R. Jensen, A. Nbrlund Christensen, J.C. Hanson, Hydrothermal
600 transformation of the calcium aluminum oxide hydrates CaAl₂O₄·10H₂O and
601 Ca₂Al₂O₅·8H₂O to Ca₃Al₂(OH)₁₂ investigated by in situ synchrotron X-ray powder
602 diffraction, Cement and Concrete Research 35 (2005) 2300-2309

603 [24] F.P. Glasser, L. Zhang, High-performance cement matrices based on calcium
604 sulfoaluminate–belite compositions, Cement and Concrete Research 31 (2001) 1881–
605 1886

606 [25] L. Fernández-Carrasco, D. Torrens-Martín, L.M. Morales, S. Martínez- Ramírez,
607 Infrared Spectroscopy in the Analysis of Building and Construction Materials, Infrared
608 Spectroscopy, Materials Science, Engineering and Technology (2012) 369-382

609 [26] N.V. Chukanov, Infrared spectra of mineral species (2014) Ed Springer

610 [27] S.M. Bushnell-Watson, J.H. Sharp, The application of thermal analysis to the
611 hydration and conversión Reactions of calcium aluminate Cements, Materiales de
612 construcción 42 (1998) 13-32

613 [28] C.M.George, Industrial alumina cement, Structure and Performance of Cement
614 (1983) 415

615 [29] L. Fernandez Carrasco, Procesos de hidratación y carbonatación del cemento de
616 aluminatos de calcio; influencia de los álcalis. Alteraciones microestructurales y
617 relación con sus propiedades mecánicas. Thesis Instituto de la Construcción Eduardo
618 Torroja, Madrid, Spain, 2000

619 [30] N. Ukrainczyk, T. Matusinovic, S. Kurajica, B. Zimmermann, J. Sipusic,
620 Dehydration of a layered double hydroxide—C₂AH₈N, *Thermodynamic Acta* 464
621 (2007) 7-15

622 [31] V.S. Ramachandran, Thermal analysis of cement components hydrated in the
623 presence of calcium carbonate, *Thermodynamic Acta* 127 (1988) 385-394

624 [32] P.A. Barnes, D.J. Baxter, A critical analysis of the application of derivative
625 thermogravimetry to the determination of the degree of conversion of high alumina
626 cement, *Thermochimica Acta* 24 (1978) 427-431

627 [33] J. Szczerba, D. Madej, E. Sniezek, R. Prorok, The application of DTA and TG
628 methods to investigate the non-crystalline hydration products of CaAl₂O₄ and
629 Ca₇ZrAl₆O₁₈ compounds, *Thermochimica Acta* 567 (2013) 40– 45

630 [34] C. Gosselin, Microstructural Development of Calcium Aluminate Cement Based
631 Systems with and without Supplementary Cementitious Materials, Thesis École
632 Polytechnique Fédérale de Lausanne, Lausanne, Switzerland (2009)

633 [35] V.S. Ramachandran, R.M. Paroli, J.J. Beaudoin, A.H. Delgado, Handbook of
634 Thermal Analysis of Construction Materials, ed. W.A. Publishing. 2003, Norwich, New
635 York, U.S.A

636 [36] H.J. Kuzel, Crystallographic data and thermal decomposition of synthetic gehlenite
637 hydrate 2CaO·Al₂O₃·SiO₂·8H₂O, *Neues Jahrbuch fuer Mineralogie Monatshefte* 148
638 (1976) 319-325

639 [37] T. Danner, H. Justnes, M. Geiker, R.A. Lauten, Early hydration of C₃A–gypsum
640 pastes with Ca- and Na-lignosulfonate, *Cement and Concrete Research* 79 (2016) 333-
641 343

642 [38] F. Sáez del Bosque I. Modificaciones nanoestructurales en pastas de cemento.
643 Thesis Instituto de Ciencias de la Construcción Eduardo Torroja (CSIC) and
644 Universidad Autonoma de Madrid, Madrid, Spain (2012).
645 [39] D. Torréns-Martín, L. Fernandez-Carrasco, M.T. Blanco-Varela, Thermal analysis
646 of blended cements, Journal of Thermal Analysis calorimetric 121 (3) (2015)
647 [40] T. Matschei. Thermodynamic properties of Portland cement hydrates in the system
648 CaO–Al₂O₃–SiO₂–CaSO₄–CaCO₃–H₂O, Cement and Concrete Research 37 (2007)
649 1379-1410
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Table 1. Particle size distribution parameters determined by laser diffraction

$\mu\text{m}/\%$	C_3A	CA	C_{12}A_7	$\text{C}_4\text{A}_3\bar{\text{S}}$	$\text{C}_5\text{S}_2\bar{\text{S}}$
d_{90}	100	100	76.5	78	89.5
d_{50}	93.5	99.3	66.4	46	85
d_{10}	31	57	34	14	43.5
$D_{4.3}$	21.17	11.52	59	59.14	35.13
Span	2.41	3.17	8.25	1.99	8.29
RRSB	0.74	0.97	2.36	0.62	2.48

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Table 2. Quantitative mineralogical analysis of the phases synthesised

wt%	$\text{C}_5\text{S}_2\bar{\text{S}}$	C_2S	C_3A	C_{12}A_7	CaO	CA	$\text{C}_4\text{A}_3\bar{\text{S}}$	$\text{C}\bar{\text{S}}$	Rwp
$\text{C}_5\text{S}_2\bar{\text{S}}$	94	6							10
C_3A			100						16.52
C_{12}A_7				99.82	0.18				16.99
CA				2.52		97.48			14.12
$\text{C}_4\text{A}_3\bar{\text{S}}$				1.30			97.98	0.70	15.63
ICSD card	85123	81096	1841	241243	52783	260	80361	16382	

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Table 3. Data for the heat flow curves shown in Figure 3

	Initial peak 1	Peak 2 maximum		Min induction period		Peak 3 maximum		Maximum exothermal signal	
	Heat flow (mW/g)	t (h)	Heat fl. (mW/g)	t (h)	Heat fl. (mW/g)	t(h)	Heat fl. (mW/g)	t(h)	Heat fl. (mW/g)
C_3A	180	-	-	-	-	-	-	-	-
$\text{C}_3\text{A}+\text{ter}$	120	-	-	-	-	-	-	-	-
C_{12}A_7	159	1	20	5.5	9.0	8.5	13	-	-
$\text{C}_{12}\text{A}_7+\text{ter}$	134	1	8.9	3	6.3	5	10	13.5	1.8
CA	99	-	-	2.5	0.39	21	27	12	1
$\text{CA}+\text{ter}$	121	-	-	2	0.39	4.5	24	17	0.8
$\text{C}_4\text{A}_3\bar{\text{S}}$	71	-	-	2.8	0.33	7	114	-	-
$\text{C}_4\text{A}_3\bar{\text{S}}+\text{ter}$	23	1	1.3	2	1	4	28	9.5	1.1
Ternesite	117	-	-	-	-	-	-	-	-

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Table 4. Experimental total heat (J/g aluminate) in 7 day pastes hydration (25°C)

	Experimental total heat	ΔQ
Ternesite	43 J/g	
CA	563 J/g	
CA+Ternesite	1076 J/g	91%
$C_{12}A_7$	557 J/g	
$C_{12}A_7$ +Ternesite	1043 J/g	87%
$C_4A_3\bar{S}$	464 J/g	
$C_4A_3\bar{S}$ +Ternesite	630 J/g	36%
C_3A	744 J/g	
C_3A +Ternesite	885 J/g	19%

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Table 5. Quantitative mineralogical analysis of blended aluminate + ternesite pastes

wt%	$C_5S_2\bar{S}$	C_2S	C_3A	$C_4A_3\bar{S}$	C_2ASH_8	AFm	C_3AH_6	Hc	% amorphous	Rwp
$C_3A + C_5S_2\bar{S}$	43,11	1.71	0.72			1.76	5.36	6.91	40.42	11.31
$C_{12}A_7 + C_5S_2\bar{S}$	22.62	2.70			24.84	11.61			38.23	11.20
$C_4A_3\bar{S} + C_5S_2\bar{S}$	50.51	1.66		0.99		10.63		1.11	35.1	8.80
CA+ $C_5S_2\bar{S}$	23.29	2.21			35.95	5.84			32.71	11.62
ICSD card	85123	81096	1841	80361	69413	100138	202316	263124		

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Table 6. Absorption bands on FTIR spectra for 7 d (25 °C) pastes. s=Strong, vs=very strong, w=weak, vw=very weak

Paste C ₃ A	Paste C ₄ A ₃ S	Paste CA	Paste C ₁₂ A ₇	Attributed to
		3682	3682	
	3676 vw			AFm/Hc
3660 s				C ₃ AH ₆
	3620 w	3620	3620	AH ₃
3543 vw				C ₃ AH ₆
	3526 w	3526	3526	AH ₃
3523 vw				
	3467 w	3467	3467	AH ₃ /C ₂ AH ₈
	1167 w			AFm
	1115-1108 s			AFm
	1024 s	1024	1024	AH ₃
	968 w	968		AH ₃
899 w				C ₃ A
865 w		865	865	C ₃ A/CAH ₁₀ /C ₂ AH ₈
809 s				C ₃ AH ₆ /C ₃ A
	786 s			AFm
		777	777	
		710	710	C ₂ AH ₈
		636		
	617 w			
		578	578	CAH ₁₀
521 s	528 vs	528	528	CAH ₁₀ / C ₃ A C ₂ AH ₈ /AFm
	422 s	422	422	CAH ₁₀ / C ₂ AH ₈ /AFm

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Table 7. Temperatures and temperature ranges on DTA curves for phases obtained during hydration (°C)

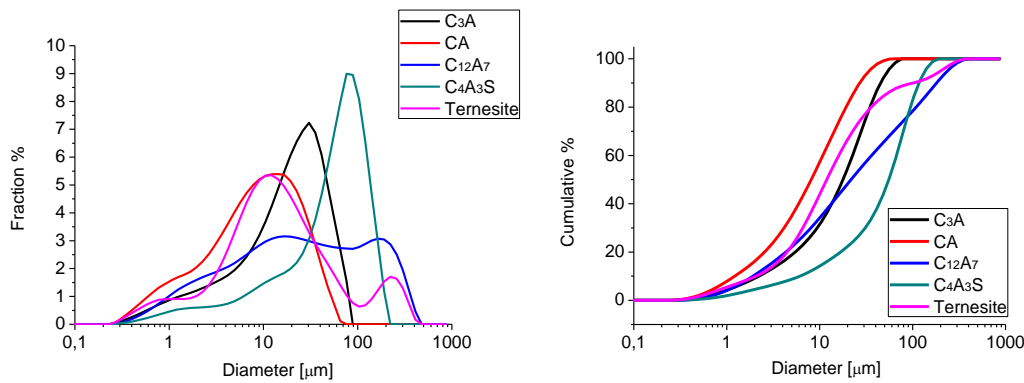
Reference	Year	CAH ₁₀	C ₂ AH ₈	C ₃ AH ₆	AH ₃	C ₂ ASH ₈	AFm	Hc y Mc
[27]	1992	100-160	140-200	290-350	230, 300	190-230		
[28]	1983	~140	~170	~300	~280			
[29]	2000	110-160	170, 270	289,450	181,3			86,155,220
[30]	2007		88,110,175,295					
[31]	1988		145-150,265-280	300				160
[32]	1978	~125		~330	~290			
[33]	2013	165	285	285-451	~260-290			
[34]	2009	~100-170	~110-290	330	~300-470	120-185		
[35]	2003		240-285	290-300	260-270		180-200	
[36]	1976					120,165, 220,430		
[37]	2016						190-200,270-290	
[38]	2012			273,400			120-170	150
This work	2017		92,150,311	303,425	239,300,515		120,180,295	

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680 Table 8. Amount of hydrates determined by XRD and under hypothesis of all sulphate
 681 in solution giving AFm and all aluminium in excess forming strätlingite. Their
 682 corresponding lost of water. TG lost of mass
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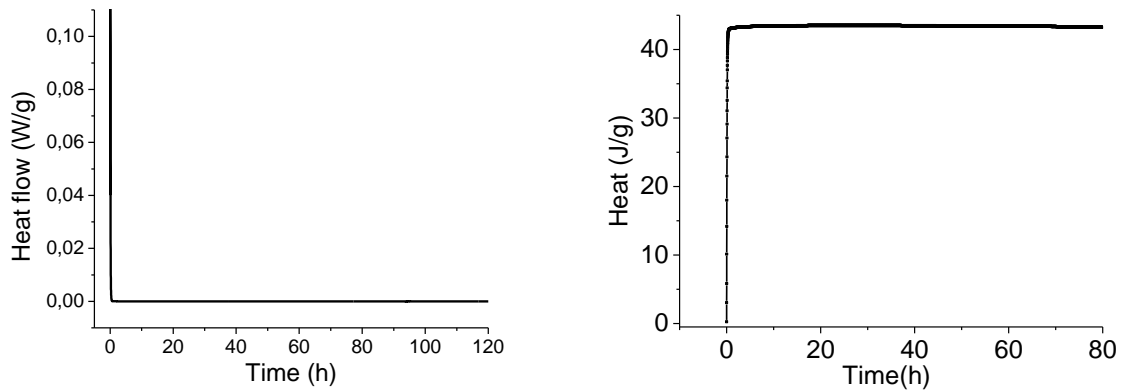
wt(%)	AFm	C ₂ ASH ₈	AH ₃	Lost of mass (water)	P (1000°C)
	DRX/hip	DRX/hip	DRX/hip	DRX/hip	TG
C ₁₂ A ₇ + C ₅ S ₂ \bar{S}	11.6 / 34	24.8 / 17.1	---	11.5 / 17.7	22
CA+ C ₅ S ₂ \bar{S}	5.8 / 30.4	36 / 40.9	---	12.8 / 21.1	25.3
C ₄ A ₃ \bar{S} +C ₅ S ₂ \bar{S}	10.6 / 29.3	---	- / 14	4.1 / 15.3	17.6

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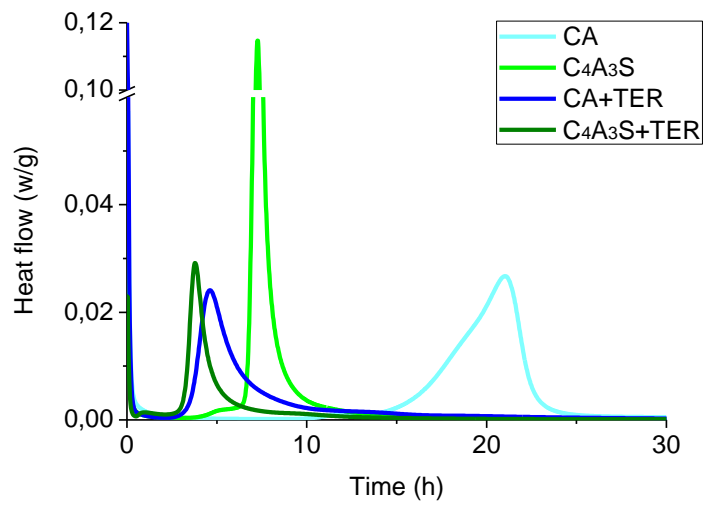
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688 Figure 1. Particle size distribution (PSD) and the derivative of the PSD of samples C₃A,
689 CA, C₁₂A₇, C₄A₃ \bar{S} , and ternesite.

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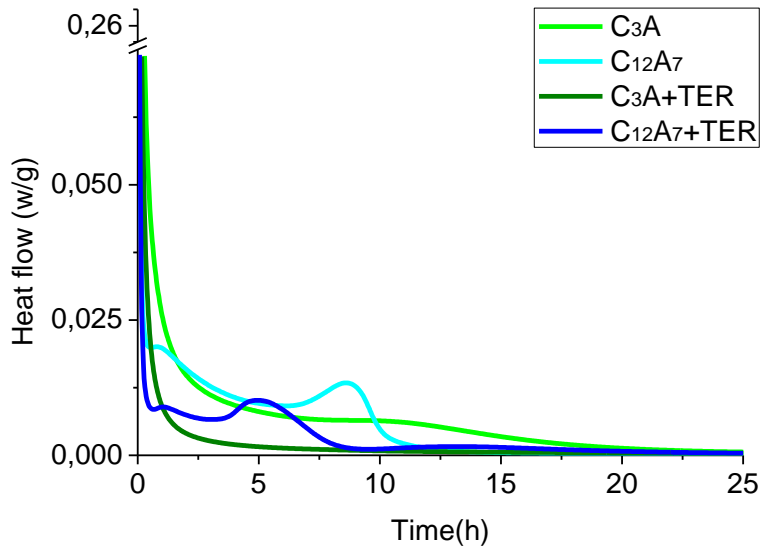


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Figure 2. Heat flow and heat curves for 7 day (25°C) ternesite hydration



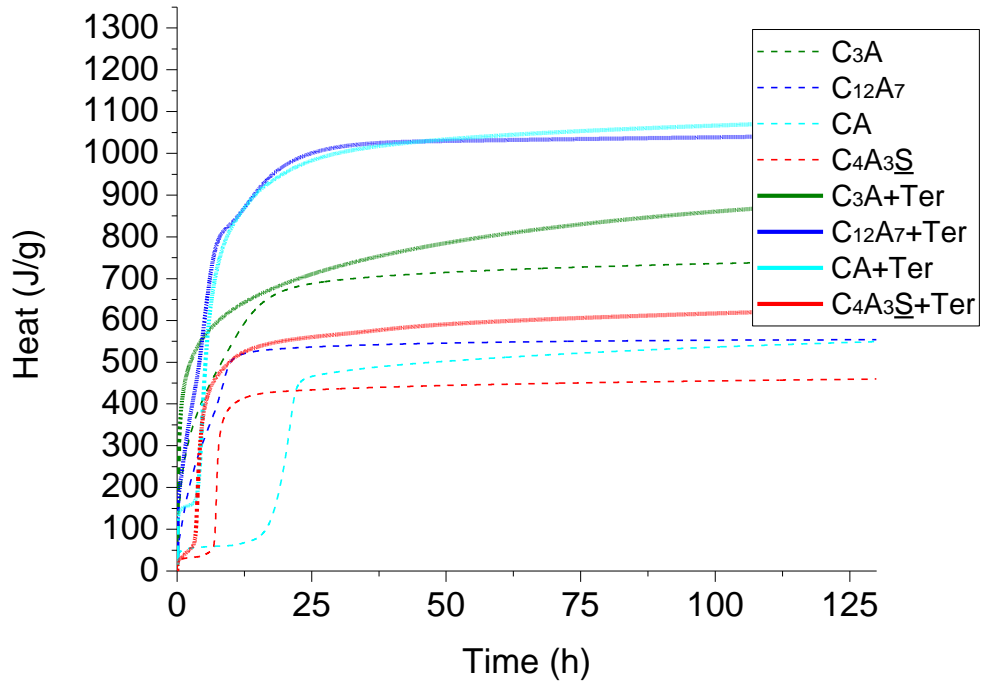
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697 b)
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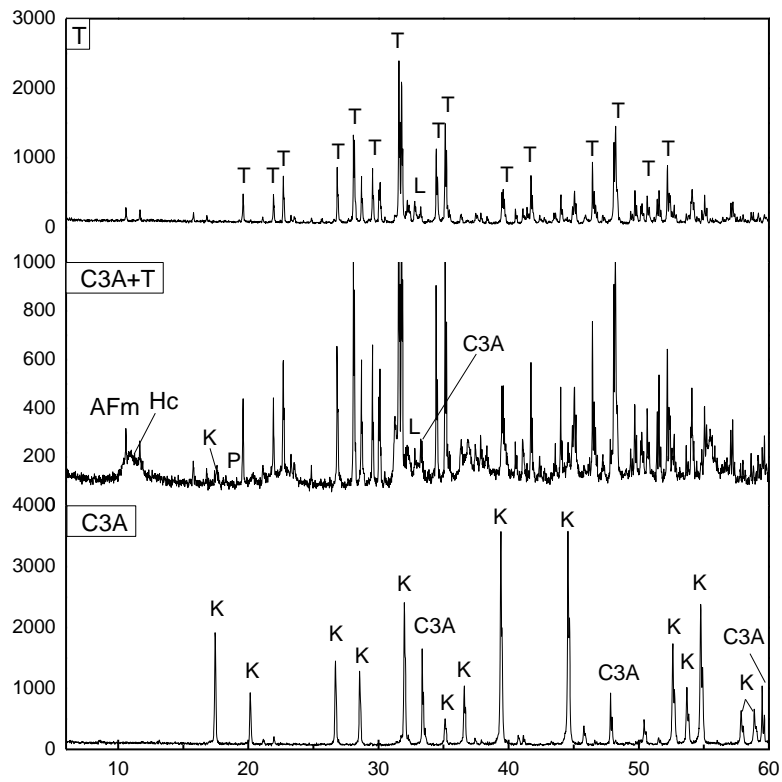
Figure 3. Heat flow curves for synthesised aluminates and their ternesite blends pastes at 25 °C

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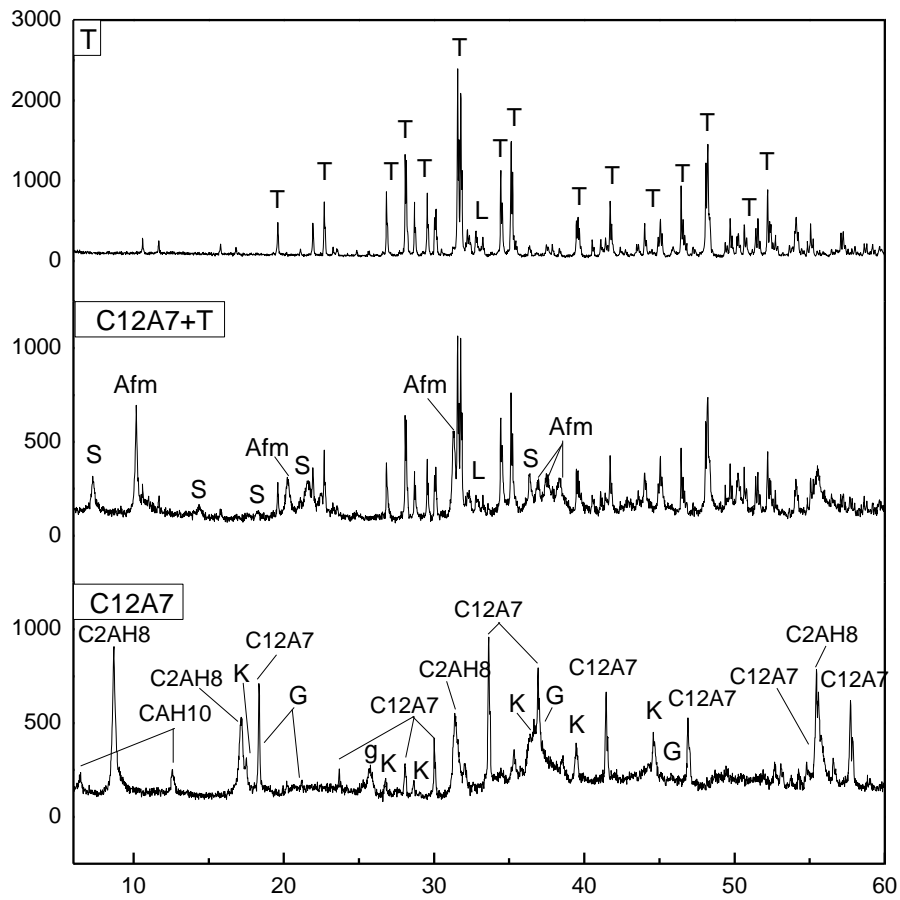


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Figure 4. Heat of hydration curves for 7 d (25 °C) hydrated aluminates and their ternesite blends (per unit of mass of aluminate)

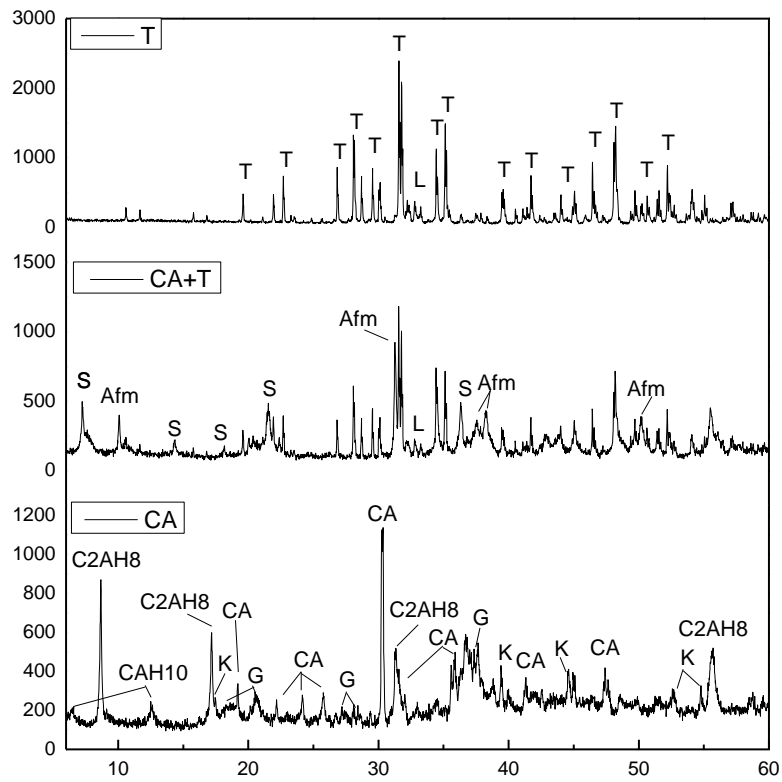


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 712 Figure 5. Diffractograms for 7 d (25 °C) C₃A and C₃A/ C₅S₂S̄ (1:2) pastes: K = katoite;
 713 AFm= Calcium sulfoaluminate.L= larnite; T= ternesite; Hc = calcium
 714 hemicarboaluminate; P= portlandite
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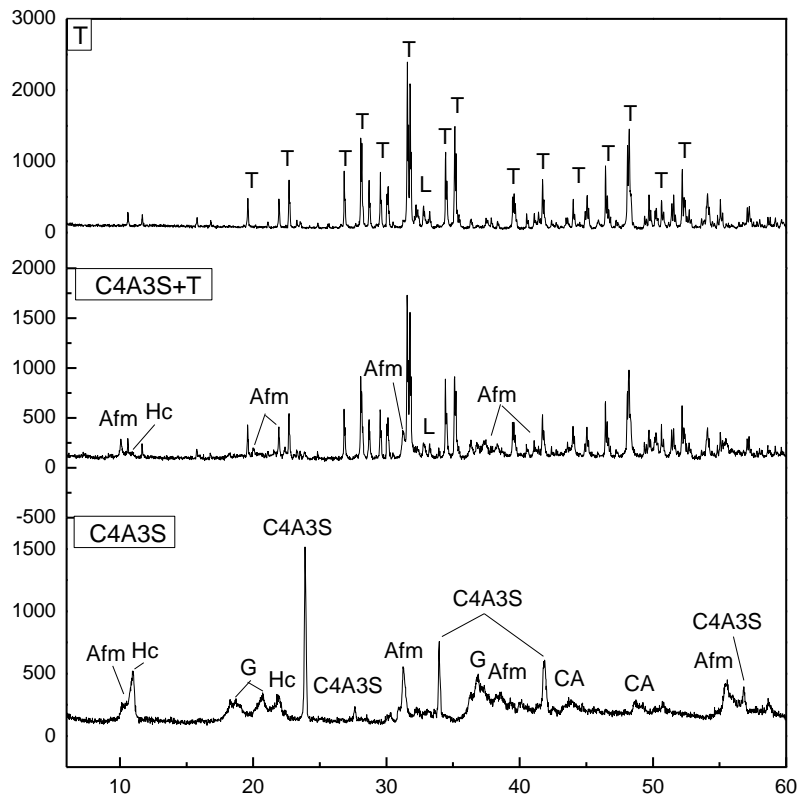
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Figure 6. Diffractograms for 7 d (25 °C) $C_{12}A_7$ and $C_{12}A_7/C_5S_2\bar{S}$ (1:2) pastes: K = katoite; G= gibbsite; L= larnite; S= strätlingite; T= ternesite; Afm = calcium monosulfoaluminate hydrate; g= grossite



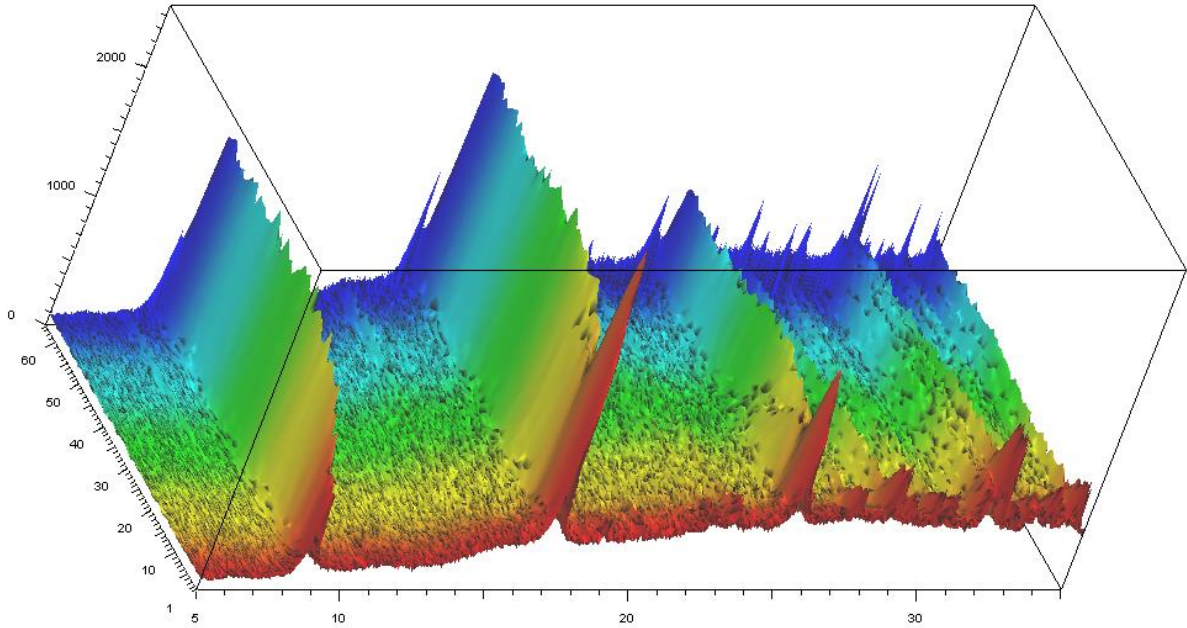
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Figure 7. Diffractograms for 7 d (25 °C) CA and CA/C₅S₂ (1:2) pastes: K = katoite; G= gibbsite; L= larnite; T= tenesite; Afm = calcium monosulfoaluminate hydrate; S= strätlingite



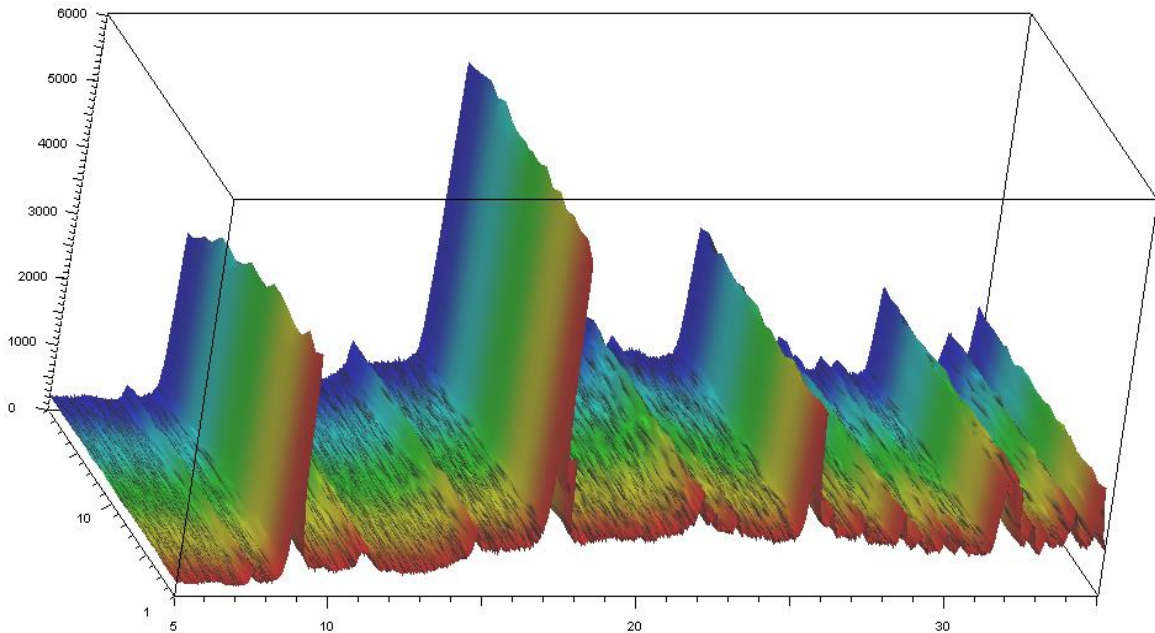
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Figure 8. Diffractograms for 7 d (25 °C) $C_4A_3\bar{S}$ and $C_4A_3\bar{S}/C_5S_2\bar{S}$ (1:2) pastes: = gibbsite; L= larnite; T= ternesite; Afm = calcium monosulfoaluminate hydrate; Hc= calcium monocarboaluminate hydrate



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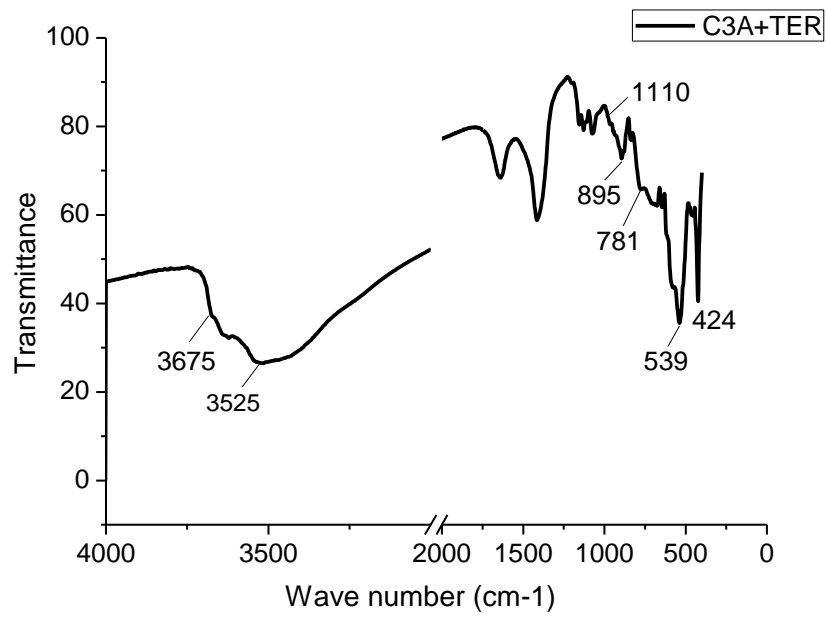
a) 1-64



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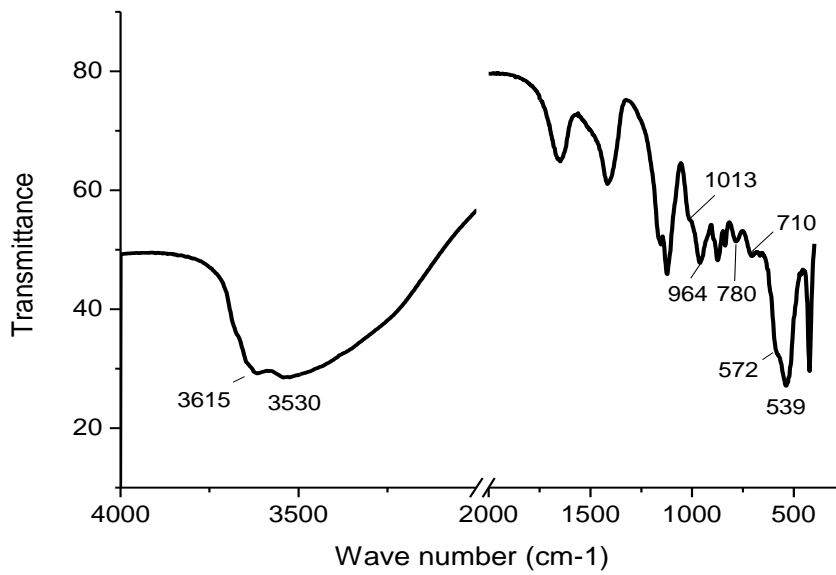
b) 65-84

Figure 9. Three-dimensional diffractograms for 0 h-20 h $C_{12}A_7$ +ternesite pastes: a) samples 1-64 (0-12 h); b) samples 65-84 (12-20 h)



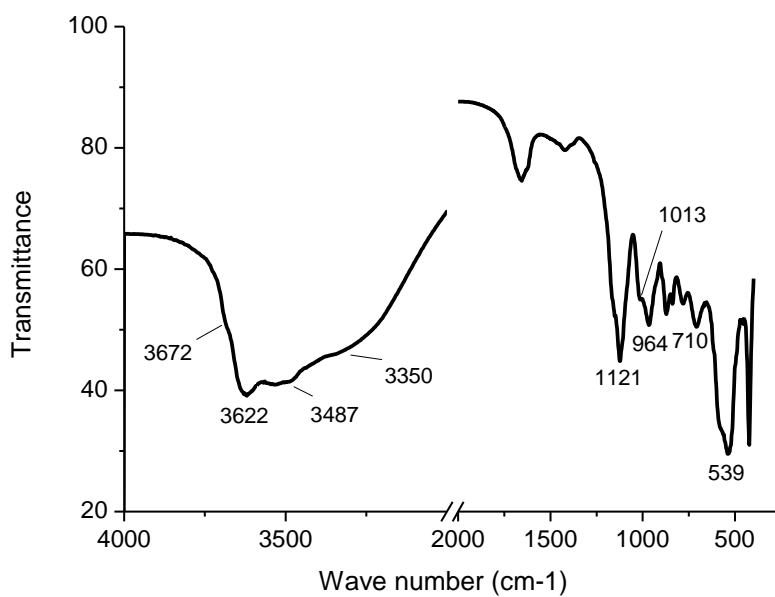
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Figure 10. FTIR spectra for 7 d (25°C) C_3A +ternesite paste after subtracting the ternesite spectrum



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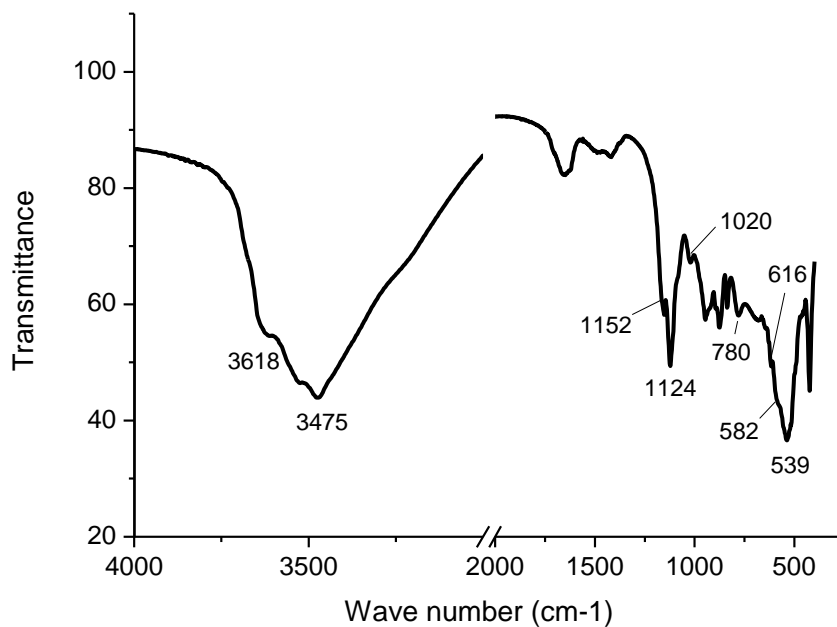
Figure 11. FTIR spectra for 7 d (25°C) $C_{12}A_7$ +ternesite paste after subtracting the ternesite spectrum



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756 Figure 12. FTIR spectra for 7 d (25°C) CA+ternesite paste after subtracting the ternesite
757 spectrum

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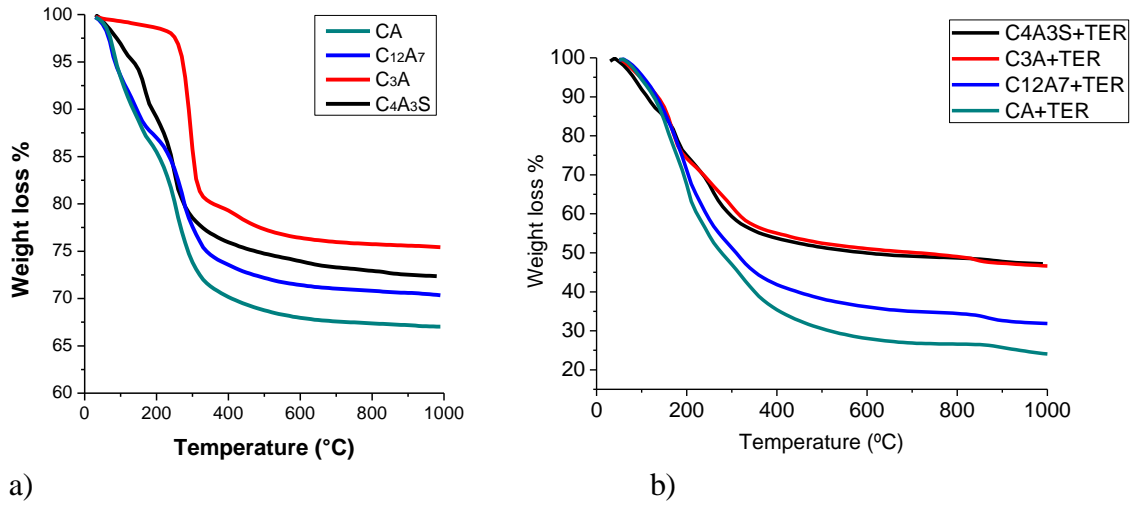


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760 Figure 13. FTIR spectra for 7 d (25°C) C₄A₃S+ternesite paste after subtracting the
761 ternesite spectrum

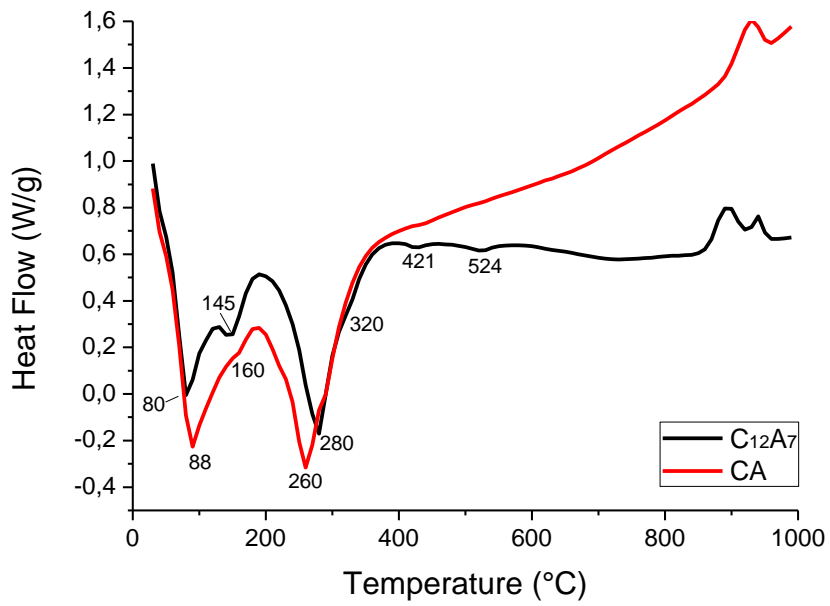
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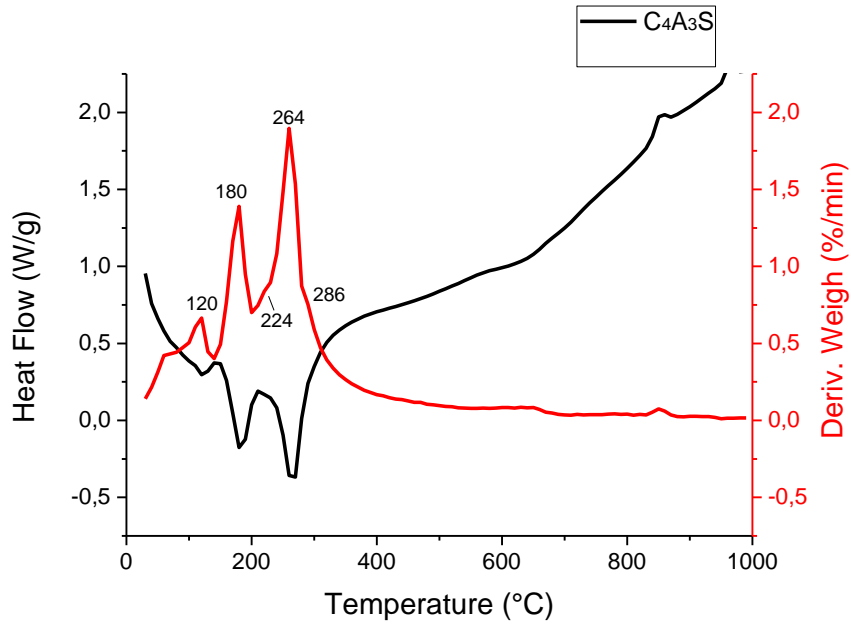
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Figure 14. a) Weight loss in aluminate pastes; b) weight loss in aluminate+ternesite pastes (per g of aluminate)



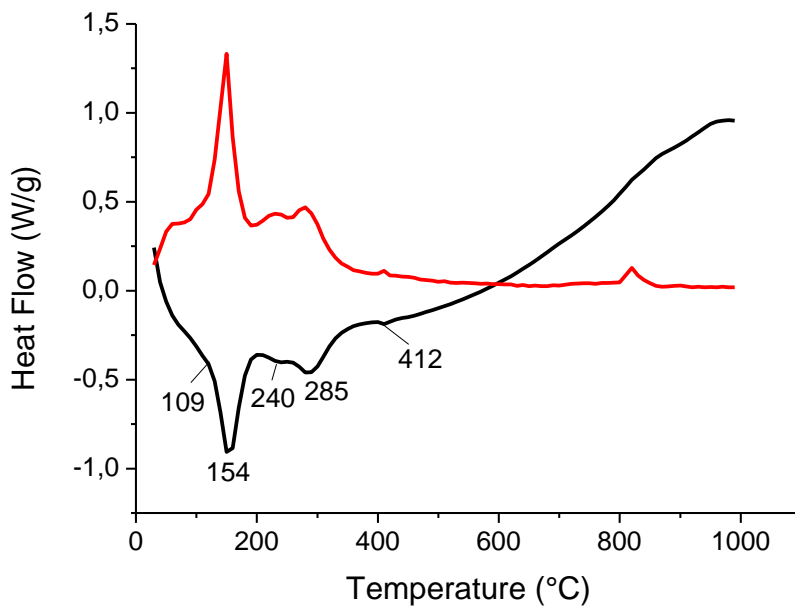
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Figure 15. DTA curves for pastes CA and C₁₂A₇



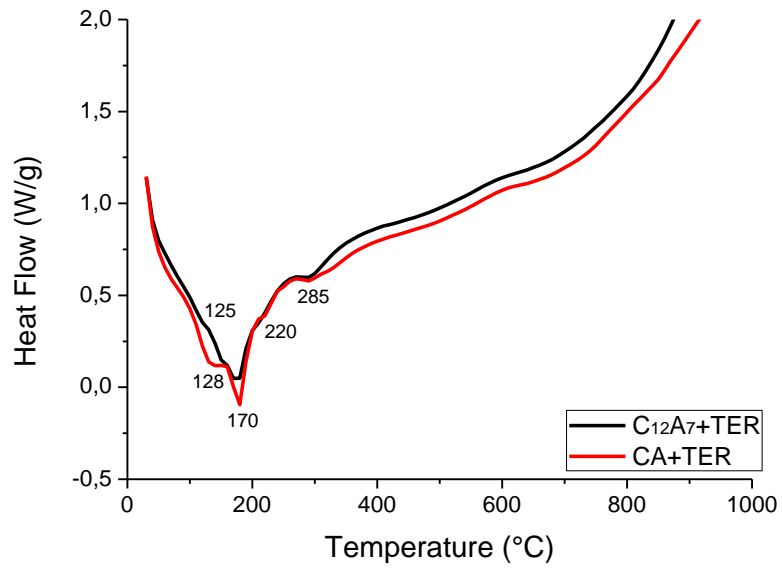
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Figure 16.DTA curves for hydrated paste C_4A_3S



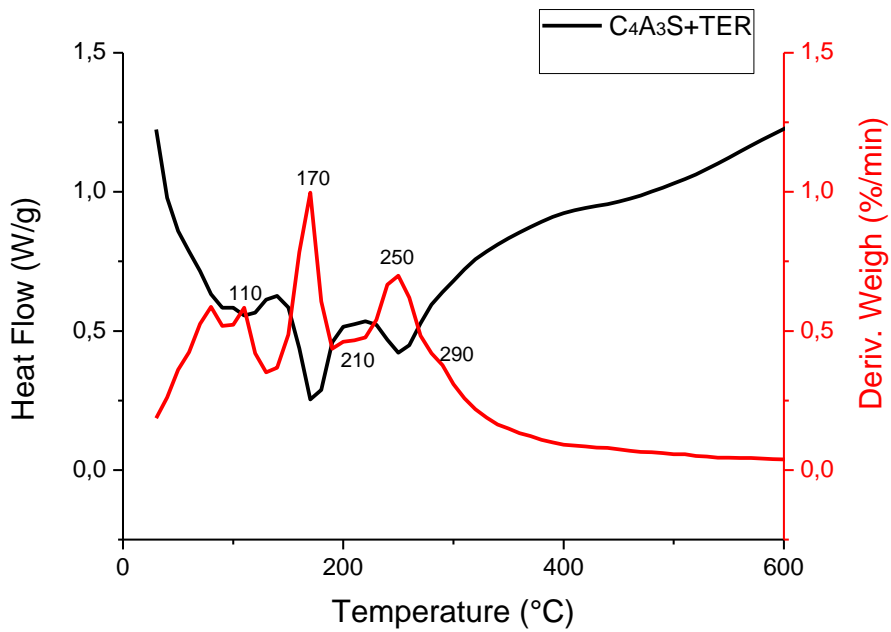
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Figure 17.DTA curves for paste C_3A +ternesite



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Figure 18.DTA curves for pastes CA+ternesite and $C_{12}A_7$ +ternesite



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Figure 19.DTA curves for paste $C_4A_3\bar{S}$ +ternesite