

DIELECTRIC BEHAVIOR OF POROUS PMMA: FROM THE MICROMETER TO THE NANOMETER SCALE

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ABSTRACT

In recent years, there has been a significant interest of the scientific community on nanocellular polymeric foams as a possible next generation of materials with a low dielectric constant for microelectronics applications. In this work, the dielectric behavior of microcellular and nanocellular poly(methyl methacrylate) (PMMA) based foams has been characterized, both as a function of frequency and temperature, in order to analyze the effect of reducing the cell size to the nanoscale on the dielectric properties. Experimental results have shown clear differences in the dielectric behavior of the samples with cell sizes in the nanoscale as well as a sharp reduction of the dielectric constant when the porosity increases.

Keywords: nanocellular foams, dielectric properties, cellular polymers, nanocellular polymers.

1. INTRODUCTION

Nanocellular foams have recently attracted significant attention in the microelectronics industry as a means of producing materials with a low dielectric constant (k) [1-6]. As the devices scale to smaller feature sizes, a new generation of low dielectric constant materials is needed to minimize cross talk and maximize signal propagation speed. The low dielectric constant of polymers and air ($k = 1$) make polymeric foams potential candidates as low k systems [7].

Poly(methyl methacrylate) (PMMA) is a thermoplastic material with a medium to high thermal stability and was one of the first polymers used in microelectronic systems [8]. In fact, this material is frequently used as a dielectric thin film [9-11]. In addition, PMMA has been recently used to produce nanocellular foams with different densities and cell sizes. Due to these reasons PMMA was selected for this study.

In microelectronics, the decreasing dimensions of the devices induces severe size restrictions to the cell size, which should ideally be an order of magnitude smaller than the thickness of the

dielectric film [12], i.e. if a polymer foam is used as a dielectric film, as the thickness of the film could be of a few micrometers the cells should have dimensions in the nanoscale. Therefore, according to the literature, these required cell sizes are merely the result of the dimensions of the devices and it is not expected a significant difference between the well-known dielectric behavior of conventional, or microcellular foams, and that of the recently developed nanocellular foams [13]. However, a preliminar work by the authors has showed the emergence of a Maxwell Wagner Sillars (MWS) phenomenon in PMMA-based nanocellular foams at room temperature [14].

Further research to analyze this unexpected behavior as a function of temperature and frequency could provide important information for the application of nanocellular foams in microelectronic devices [14], where thermal stability is critical and one of the main drawbacks of polymer-based materials [15]. Therefore, the aim of the present study was to investigate in detail the dielectric behavior of both micro and nanocellular foams as a function of frequency and temperature showing the main differences between the behaviors of the two type of materials when cell size is reduced to the nano-scale.

2. EXPERIMENTAL SECTION

PMMA was supplied by Arkema Company (France) in the form of pellets. This polymer presents a density (ρ) of 1180 kg/m³ and a glass transition temperature (T_g) around 112 °C.

PMMA pellets were injected into pieces of 50 x 15 mm² with 3 mm in thickness (model DSM Xplore), and used later for foaming.

Foaming experiments were performed at room temperature following the solid state foaming process (details can be found in the Supporting Information), modifying the saturation pressure between 10 and 30 MPa. The saturation pressures used are included in table 1.

Dense skin of foamed samples was removed using a polishing machine (model LaboPOI2-LaboForce3, Struers). Then solid and polished samples were machined using a precision cutting machine (Mod. 1000, IsoMet) to be approximately 10 x 10 mm² with 1 mm in thickness. The cellular structure of foams was characterized by scanning electron microscope (SEM, model Quanta 200FEG, FEI).

The dielectric behavior of cellular polymers was evaluated by means of a direct current (DC) electrical resistivity adapter (Keithley 6105, model) and a broadband dielectric spectrometer (ALPHA, Novocontrol Technologies GmbH).

DC resistivity data were determined according to ASTM D257-99 [16]. Solid and foamed samples were measured four times at + 500 V, -500 V, + 500 V, -500 V. Time of electrification was 60 seconds, and the time of discharge before making a measurement with reversed voltage was 4 minutes. Resistivity (R) was calculated as follows:

$$R = \frac{A \cdot V}{t \cdot I} \quad (1)$$

where A and t are the area and thickness of the sample respectively, V is the applied voltage, and I is the intensity measured.

In the case of broadband dielectric measurements, foamed and solid samples were held in a dielectric cell between two parallel gold-plated electrodes. The thickness of the samples was taken as the distance between the electrodes and was determined using a micrometer gauge. The complex dielectric permittivity ($\epsilon^* = \epsilon' - i\epsilon''$) of the solid and foamed materials was measured over a frequency window of $10^{-2} < F/\text{Hz} < 10^6$ (F is the frequency of the applied electric field) in the temperature range from -20 to 110 °C. The amplitude of the alternating current (ac) electric signal applied to the samples was 1 V. Furthermore, the real part of the complex dielectric permittivity, ϵ' , (or dielectric constant, k) was modeled using the series model [17], the parallel model [17], and the Maxwell Garnett model [18].

3. RESULTS AND DISCUSSION

DC resistivity (R , Ω/cm) at room temperature of PMMA foams with pore sizes between 90 and 3290 nm (Table 1, representative SEM micrographs of the foams can be found in Figure S.1, see Supporting Information) did not follow a simple trend as a function of to the relative density (Figure 1, left). The resistivity of the foams increases by two orders of magnitude when the pore size falls from 1460 nm to 710 nm, despite a simultaneous and slight decrease of the relative density (that should reduce the resistivity).

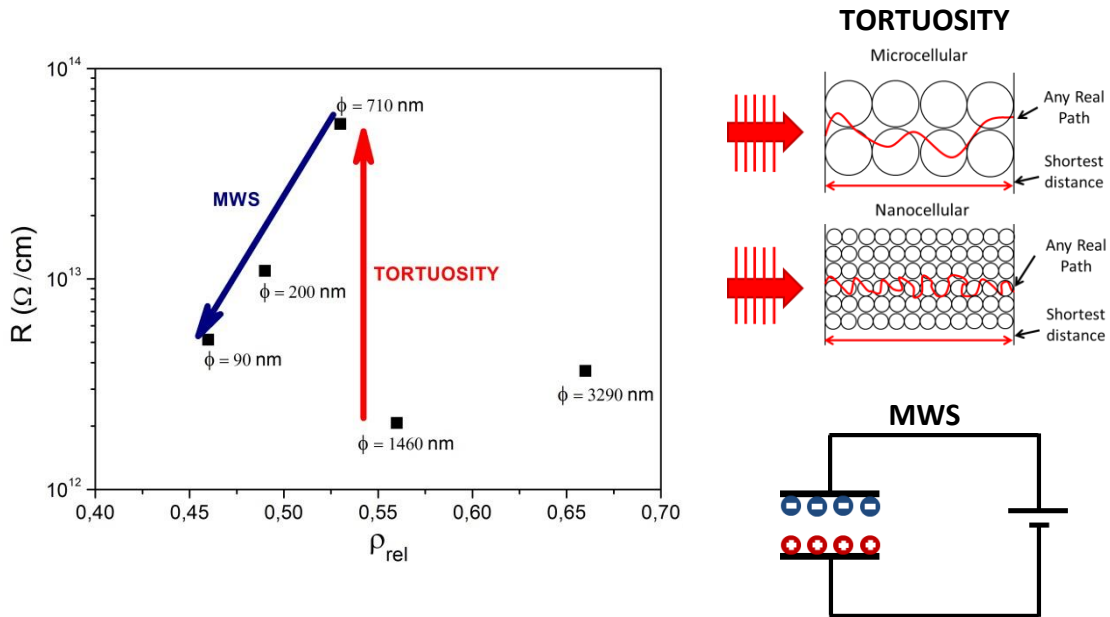


Figure 1. Left) Evolution of the DC electrical resistivity as a function of the relative density. Right) Scheme of the tortuosity (top) and MWS (bottom) phenomenon. The numbers in the figure of DC resistivity vs relative density correspond to the cell size of each foam.

This behavior was previously observed by J. Pinto and coworkers [14], who demonstrated the appearance of an electrical conductivity component and an interfacial polarization phenomena

(or MWS) in nanocellular PMMA based foams at low frequencies. MWS occurs in heterogeneous materials, such as blends or composites, at the interfaces, leading to a separation of charges (Figure 1, right). Furthermore, they found an increase in the electrical resistivity of PMMA based foams when cell size shifts from the micro to the nanoscale due to an increase tortuosity of the solid phase (tortuosity of the solid phase is the ratio between the minimum distance of any real path between two areas of the cellular material and the shortest distance between these two areas (Figure 1, right)).

The appearance of both phenomena can clearly be observed in Figure 1 left. A clear increase of the resistivity is observed when cell size changes from the micro (1460 nm) to the nanometer range (710 nm); this is due to a higher tortuosity of the solid phase in the nanocellular foam (Figure 1, right). In addition, for materials with cells sizes below 700 nm a reduction of the resistivity when the cell size is reduced is detected. The reason for this behavior is a conductive mechanism at low frequencies related to the MWS, which is due to the accumulation of charges in opposing cell walls.

As mentioned before, nanocellular polymeric foams are required in the microelectronics due to the reduced thickness of dielectric materials. Therefore, the dielectric behavior of the fabricated PMMA microcellular and nanocellular foams was studied with the aim of understanding if the change in the internal structure can affect the dielectric behavior.

First, the dielectric constant (k) of solid, microcellular and nanocellular PMMA at a frequency of 1 KHz (Table 1) was measured. As expected the foams presented reduced k values in comparison with the solid materials. For instance the foam with the smallest pore size (90 nm) that has a relative density of 0.46 showed an important reduction of k of around 55 % (from 7.73 to 3.56).

Table 1. Main characteristics (saturation pressure, cell size and relative density) and , experimental and theoretical values of the dielectric constant (k) of the analyzed materials

Saturation Pressure (MPa)	Sample - Cell Size (Φ) (nm)	Relative Density (ρ_{rel})	K at 1 KHz	K_f Series Model [17]	K_f Parallel Model [17]	K_f Maxwell Garnett Model [18]
0 (Solid)	- (Solid)	1	7.73	-	-	-
30	90	0.46	3.56	1.66	4.01	3.53
25	200	0.49	4.25	1.74	4.28	3.73
20	710	0.53	4.50	1.85	4.63	3.99
15	1460	0.56	5.07	1.95	4.89	4.19
10	3290	0.66	5.61	2.35	5.76	4.90

With the aim of identifying the possible influence of the pore size in the dielectric constant, the results were compared with the predictions provided by the most common theoretical models used in previous literature. These models consider the system as an isotropic medium and only take into account the fraction of voids (porosity= $(1-\rho_{rel})$) of the foams. It was found that these models fail to predict the experimental results (see Table 1), being not able to

simultaneously fit the k values measured for the largest and the smallest pore sizes. This result is probably due to one of the models' hypothesis. In all these models the properties of the constituent phases (both solid and gas phase) do not change with cell size. However previous works [14, 19-21], have proved that both the gas within the pores and the polymeric chains in the solid matrix have different properties when the cell size is decreased to the nanometer range. Once these changes are fully understood they should be introduced on more advanced theoretical models able to provide accurate predictions of the dielectric constant for both microcellular and nanocellular foams.

It should be noticed that the values of the dielectric constant of the materials included in this paper are far from the ones currently required by microelectronics industry ($k < 2$ [22]). However, the results presented here show a very promising way of achieving low dielectric constant materials by the production of low density nanocellular foams

Second, , it was analyzed if the previous evidences of a different behavior of microcellular and nanocellular foams could be confirmed in the frequency domain. With this aim the complex plot of the impedance (Z) or Nyquist plot was obtained for the solid, microcellular and nanocellular PMMA (Figure 2). Nyquist curves include the entire range of frequencies measured and are used to obtain information about the equivalent electric circuit. In this study a clear change in the samples behaviour is observed. The solid sample presents a dominating capacitor behaviour over the entire frequency range. Nevertheless, since an ideal capacitor appears as a vertical straight line (infinite resistance), the slope that the solid samples presents indicates a slight resistive contribution most likely arising from the capacitor losses present in any dielectric material. This slope then starts to decrease with the presence of the cellular architecture reaching its minimum value in the nanocellular material. Therefore, the nanocellular PMMA develops a resistive behaviour and can be modelled as a capacitor in parallel with a resistor. Reduction of the capacitive behaviour found on nanoporous PMMA foams could be related to the progressive immobilization of the polymer chains leading to a progressive decrease of free dipoles able to rotate with the reduction of the pore size.

In short, the previous result confirms that the pore size present a clear influence on the dielectric behavior of PMMA foams. Moreover, dielectric measurements at room temperature confirmed previously published experimental results obtained by measuring the dynamic mechanic behavior ($\tan \delta$) of microcellular and nanocellular PMMA [20]. These results showed that the $\tan \delta$ at low and medium frequencies was increased when the cell size is reduced to the nanometer range (Figure S.2, see Supporting Information).

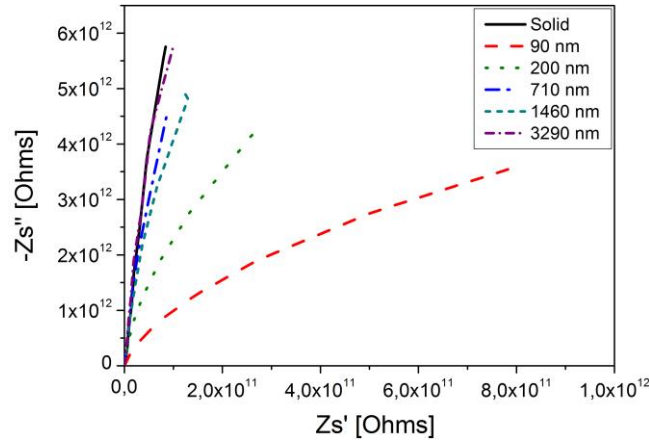


Figure 2. Nyquist plot for solid, microcellular, and nanocellular PMMA

Finally, the thermal stability of these effects was analysed as a function of the cell size. In order to obtain this information, measurements in the temperature domain were also conducted. In the case of the shape parameters alpha and beta, no differences between microcellular and nanocellular foams were detected as a function of temperature. This behaviour must be further studied due to the previous evidences found in the $\tan \delta$. On the other hand, a decrease of the normalized dielectric strength, $\Delta\epsilon/\Delta\epsilon^{solid}$, that was stable in the temperature range measured (from -20 to 110 °C) was observed as a function of cell size (Figure 3).

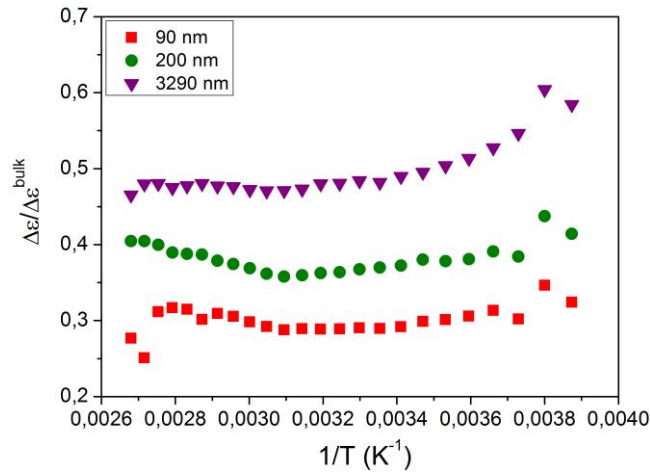


Figure 3. Evolution of the normalized dielectric strength as a function of the inverse of temperature for materials with different cell sizes.

This reduction of $\Delta\epsilon/\Delta\epsilon^{solid}$ was previously observed only at room temperature [14] and it was related to a progressive decrease of the number of free dipoles able to rotate, i.e., it is related to the confinement effect of the solid phase in the nanocellular system. Therefore, this results proves that the modifications of the dielectric behavior of PMMA foams induced by the pore size are stable in the temperature domain; a relevant result from an applied point of view.

4. CONCLUSIONS

The dielectric properties of microcellular and nanocellular PMMA foams have been studied, finding a clear evolution from a capacitive behavior to a combination of a resistive and a capacitive response when the cell size is reduced from the microscale to the nanoscale. In particular, it has been found that the dielectric constant of nanocellular PMMA is strongly reduced over that of solid PMMA, showing that the development of nanocellular polymeric materials is a promising way to achieve the next-generation of dielectric materials with a very low thickness (as long as unexpected phenomena like the MWS do not appear at high frequencies). Furthermore, experimental results have confirmed the confinement of the polymer chains along the temperature domain (from -20 to 110 °C), showing a great stability up to temperatures close to the glass transition temperature (T_g) of the solid material. Finally, some dynamic mechanical properties previously published of nanocellular materials as well as the dielectric spectrum of PMMA foams have been verified and confirmed [14, 20]. Nowadays, the unexpected conductive mechanism found at low frequencies in nanocellular foams is not a problem for the implementation of nanocellular polymeric systems as low dielectric constant materials (since they have to work at high frequencies), but it has to be taken into account to study these systems in more detail and to develop new models to predict its complex behavior.

Acknowledgments

Financial support from FPI grant BES-2013-062852 (B. Notario) from MINECO and FEDER program (MAT 2012 – 34901) MINECO, FEDER, UE (MAT2015-69234-R) and the Junta de Castile and Leon (VA035U13) are gratefully acknowledged.

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Supporting Information

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Solid state foaming process:

Neat PMMA was injected into pieces $50 \times 15 \text{ mm}^2$ with 3 mm thickness, using a small scale injection molding machine developed by DSM Xplore. The working temperature was fixed at 240 °C, whereas mold temperature was set at 60 °C. The injection pressure was fixed at 1 MPa. All samples were transparent and showed a good surface appearance as well as a good injection behavior, without air bubbles inside the parts.

Foaming experiments were carried out in a high pressure vessel provided by TOP Industry (France), with a capacity of 300 cm³ and capable of operating at maximum temperature of 250 °C and maximum pressure of 40 MPa. The reactor is equipped with an accurate pressure pump controller provided by Teledyne ISCO, and controlled automatically to keep the temperature and pressure at the desired values. The CO₂ vessel temperature and pressure were monitored in the course of the process. Thus a collection of experiments was performed in a modified solid state foaming process. The usual solid state foaming process with amorphous polymers has three stages: the saturation (under fixed gas pressure and temperature), gas desorption during and after the pressure release (to room pressure and temperature), and foaming of the sample (at a temperature over or around the T_g of the plasticized polymer). However, the actual glass transition temperature of PMMA-CO₂ systems can reach values close to room temperature, even below room temperature in some particular conditions, so in this work the desorption (at room temperature) and foaming stages (also performed at room temperature) are not clearly separated.

In this study, samples were saturated at different pressures, from 10 to 30 MPa, and at room temperature during 24 h to assure the complete dissolution of CO₂ in the polymer. After this saturation process, foaming was triggered by releasing the pressure inside the vessel at a pressure drop rate between 10 and 30 MPa/min and carried out at room temperature.

SEM micrographs:

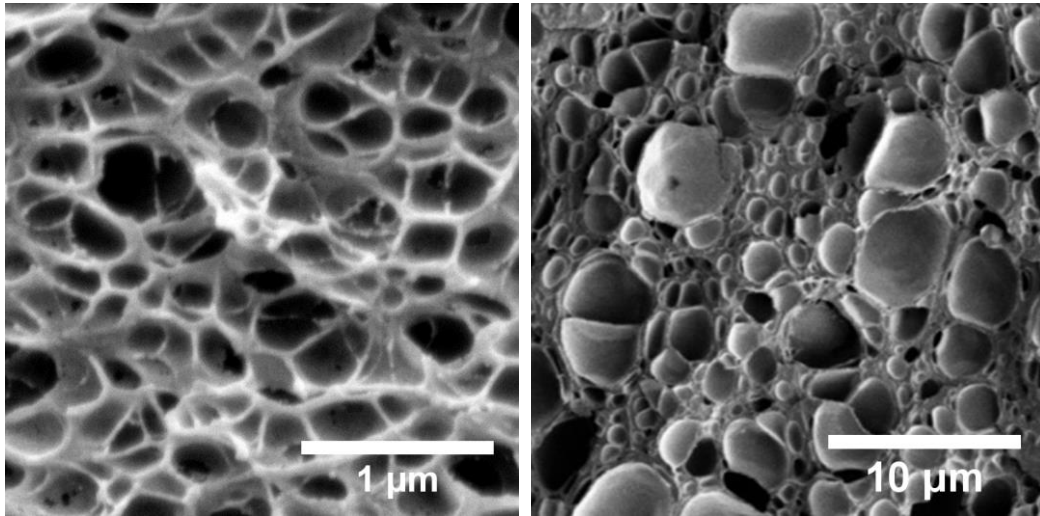


Figure S.1. Representative SEM micrographs of nanocellular (left) and microcellular (right) PMMA foams.

Tan δ :

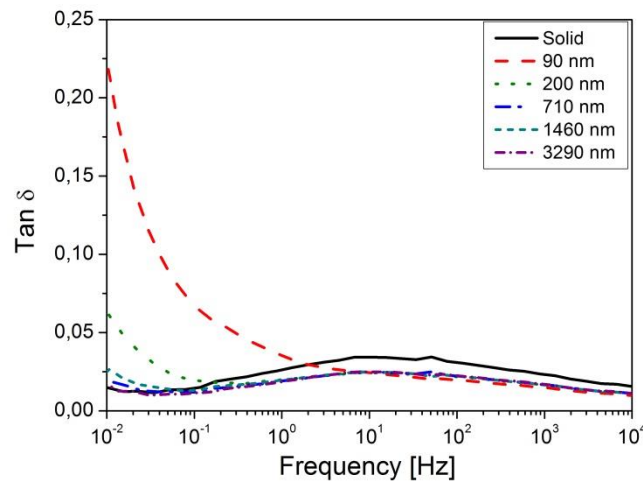


Figure S.2. Evolution of $Tan \delta$ at room temperature vs frequency for the solid, microcellular, and nanocellular PMMA

Figure S.2 illustrates the evolution of the $Tan \delta$ at room temperature as a function of the frequency for the solid, microcellular, and nanocellular PMMA foams. It can be observed how at low and medium frequencies $Tan \delta$ increases considerably when cell size is reduced to the nanometer range, remaining unaltered for solid and microcellular foams in the whole frequency range. This increase in $Tan \delta$ is related to the so called confinement effect [14].