Early stages of glass alteration in the coastal atmosphere

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Abstract

Urban and industrial atmospheres can produce several damages on glass façades and historical stained-glass windows; however, the effect of coastal atmosphere on them has been scarcely studied. This work presents the results of the exposure of soda-lime, potash-lime, and mixed-alkali silicate glass to the coastal atmosphere in Cape Vilán (Galicia, Spain) at different distances to the coast (inland) during twelve months. Crystalline deposits were observed on all the samples, although their quantity depended on the meteorological conditions. The samples located nearest to the coast presented alteration layers, while those ones located farthest from the shore presented several isolated pits. Regarding the chemical composition, potash-lime silicate glass, typical composition from medieval glass windows, was the most altered glass in contrast to soda-lime and mixed-alkali silicate glasses.

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**Graphical abstract:**

![Graphical abstract image](image.gov)

**Keywords:** Glass, degradation, marine aerosol, Atlantic coast.

**Highlights**

1. The glass alteration depended on the distance to the coast
2. The glasses exposed near the coast presented alteration layers on their surface
3. The glasses exposed far from the shore presented isolated pits
4. Potash-lime silicate glass was the most altered glass
1. Introduction

Degradation of glass façades and historical stained-glass windows in atmospheric environment has been an important field of study in the last years since many of medieval windows present a poor state of conservation due to environmental factors [1-6]. The most common weathering pathologies are dust accumulation, pits, and crusts. These pathologies decrease the glass transparency, diminishing the artistic value of the stained-glass windows [2-5, 7-12].

Soiling is the most common alteration pathology in glasses and it concerns the deposit of soot particles and salts over its surface. The nature of these deposits depends on the type of atmosphere to which the glasses are exposed (urban, industrial, rural...), although insoluble mineral dust and the soluble inorganic species $\text{SO}_4^{2-}$, $\text{NO}_3^-$, $\text{Ca}^{2+}$, $\text{Na}^+$ and $\text{Cl}^-$ are the most common ones [7-9]. Surface deposits produce the increase of the surface roughness and the hygroscopicity [7]. Water, both rain and condensation, is the main degradation agent because it produces the lixiviation of alkaline and alkaline-earth ions from the glass matrix and the hydrolytic attack of the siloxane bonds [2, 3, 5, 10, 11]. As a result of this attack, pits and alteration layers can be produced.

Environmental pollution produces also a very intense effect on glass. Urban environments induce more intense soiling and alteration pathologies than sub-urban areas due to the higher pollution in the former areas [3, 5, 12]. Moreover, the atmospheric gases ($\text{CO}_2$, $\text{SO}_2$, $\text{NO}_2$), solubilized in rainwater, can react with the alkaline and alkaline-earths ions ($\text{Na}^+$, $\text{K}^+$, $\text{Ca}^{2+}$) leached from the glass to form crystalline deposits [4]. The accumulation of these deposits on the glass surface can form thick crusts of carbonates or sulfates.

One of the least assessed environments is the coastal atmosphere. In this environment, the content of marine aerosols is high and they can induce a severe damage on different materials, such as metal or stone [13-19]. Nevertheless few studies on glass alteration have been carried out. In the frame of the “International Co-operative Programme on Effects on Materials including Historic and Cultural Monuments” (ICP Materials), several glass samples were spread around Europe in different atmospheres (industrial, urban, rural) [20]. The samples exposed in Athens, in a coastal environment, presented a more intense alteration process in comparison with other European cities with inland atmospheres [12]. Furthermore, some glasses exposed in Galicia (Spain) showed the formation of several crystalline phases (chlorides, carbonates...) and, even, alteration layers after being exposed during three months in a coastal atmosphere [21]. The fast alteration of these modern glasses proves the aggressiveness of the coastal atmospheres. High concentration of marine aerosols can be detected as far as $\sim$20 km from...
the coast (inland) depending on the local geography [22, 23]. For this reason, historical stained-glass windows without protection and glass façades exposed near the shore could be threatened.

The aim of this study is to determine the alteration mechanism of three type of glasses (soda-lime, potash-lime, and mixed-alkali silicate glasses) located at different distances to the shore. The samples were exposed unsheltered from rain during twelve months to characterize the real impact of a coastal atmosphere in the glass degradation. This research will be especially useful to evaluate the conservation strategies of coastal stained-glass windows and glass façades.

2. Experimental

2.1. Glass Samples

Three different glasses were formulated following the composition of the most common glasses from stained-glass windows and glass façades from different historical periods: soda-lime silicate (NCV), mixed-alkali silicate (MCV) and potash-lime silicate glasses (KCV) (Table 1). They were melted in the laboratory and blown into the form of a roundel with the same procedure than the historical production of crown window glass [24]. Ten samples of each glass were cut in slices of (1 x 1 x 0.2) cm³.

Table 1. Chemical composition of soda-lime silicate (NCV), mixed-alkali silicate (MCV) and potash-lime silicate (KCV) glasses analyzed by Particle Induced X-ray Spectrometry (PIXE).

<table>
<thead>
<tr>
<th>Chemical composition (wt. %)</th>
<th>Na₂O</th>
<th>MgO</th>
<th>Al₂O₃</th>
<th>SiO₂</th>
<th>P₂O₅</th>
<th>SO₃</th>
<th>Cl</th>
<th>K₂O</th>
<th>CaO</th>
<th>TiO₂</th>
<th>Cr₂O₃</th>
<th>MnO</th>
<th>Fe₂O₃</th>
<th>CuO</th>
<th>ZnO</th>
<th>SnO₂</th>
<th>PbO</th>
</tr>
</thead>
<tbody>
<tr>
<td>NCV</td>
<td>16.91</td>
<td>1.77</td>
<td>3.15</td>
<td>66.12</td>
<td>0.02</td>
<td>0.03</td>
<td>0.01</td>
<td>2.86</td>
<td>7.30</td>
<td>0.03</td>
<td>0.09</td>
<td>0.27</td>
<td>-</td>
<td>1.15</td>
<td>0.28</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>MCV</td>
<td>11.70</td>
<td>3.43</td>
<td>6.81</td>
<td>62.97</td>
<td>0.12</td>
<td>0.01</td>
<td>4.72</td>
<td>8.83</td>
<td>0.12</td>
<td>0.38</td>
<td>0.55</td>
<td>-</td>
<td>0.38</td>
<td>-</td>
<td>-</td>
<td>0.38</td>
<td>-</td>
</tr>
<tr>
<td>KCV</td>
<td>0.10</td>
<td>1.96</td>
<td>4.16</td>
<td>57.14</td>
<td>1.76</td>
<td>0.01</td>
<td>17.56</td>
<td>16.64</td>
<td>0.06</td>
<td>0.01</td>
<td>0.12</td>
<td>0.01</td>
<td>0.45</td>
<td>0.01</td>
<td>-</td>
<td>-</td>
<td>-</td>
</tr>
</tbody>
</table>

2.2. Corrosion tests

The samples were exposed in the Cape Vilán wind farm (Galicia, Spain) for periods of 3, 6, 9 and 12 months. They were placed in a vertical position unsheltered from the rain in three different corrosion stations located at different distances from the Atlantic shore (332, 710 and 1250 m). The deposition rate of atmospheric chloride ions (salinity) was measured by the Wet Candle Method [25] and depended on the distance from the corrosion stations to the shore (Table 2). Frequent heavy rainfall and high relative humidity levels were recorded by the AEMET (Spanish Meteorological Agency) at the test site (Table 2 and Table 3), which confirmed
the prolonged times of wetness on the glass surface. The atmospheric SO$_2$ content was negligible [26]. The Corrosion station 1 disappeared after 12 months of exposure due to strong winds in the area.

Table 2. Total precipitation, average wind speed and deposition rate of atmospheric chloride ions measured in the corrosion stations during the exposure periods with the Wet Candle Method [25].

<table>
<thead>
<tr>
<th>Exposure period</th>
<th>Total precipitation (mm)</th>
<th>Average wind speed (km/h)</th>
<th>Corrosion station / Cl$^-$(mg/m²-day)</th>
</tr>
</thead>
<tbody>
<tr>
<td>07/16 – 10/16</td>
<td>109.0</td>
<td>19.0</td>
<td>699.8 178.2 36.1</td>
</tr>
<tr>
<td>10/16 – 01/17</td>
<td>256.9</td>
<td>20.0</td>
<td>900.9 291.6 57.2</td>
</tr>
<tr>
<td>01/17 – 04/17</td>
<td>313.6</td>
<td>28.3</td>
<td>2214.1 620.7 73.8</td>
</tr>
<tr>
<td>04/17 – 07/17</td>
<td>95.2</td>
<td>20.0</td>
<td>- 288.1 28.9</td>
</tr>
</tbody>
</table>

- Not available

Table 3. Meteorological factors during the exposure in Cape Vilán wind farm (Galicia, Spain).

<table>
<thead>
<tr>
<th>Climate parameters</th>
<th>Average value</th>
</tr>
</thead>
<tbody>
<tr>
<td>Prevailing wind</td>
<td>Northeast wind</td>
</tr>
<tr>
<td>Temperature</td>
<td>14.6 ± 3.0 ºC</td>
</tr>
<tr>
<td>Humidity</td>
<td>85 ± 4 %</td>
</tr>
<tr>
<td>Days per month with precipitation &gt; 0.1 mm</td>
<td>13 ± 4 days</td>
</tr>
<tr>
<td>Days per month with precipitation &gt; 1 mm</td>
<td>8 ± 5 days</td>
</tr>
<tr>
<td>Total monthly precipitation</td>
<td>61.6 ± 49.9 mm</td>
</tr>
</tbody>
</table>

2.3. Characterization techniques

Glass samples were characterized by the following techniques: Optical microscopy (OM), Fourier Transform Infrared Spectroscopy in Attenuated Total Reflectance (FTIR-ATR) and Optical surface roughness (OSR). The static contact angle (water drop-glass surface) was also measured.

OM was carried out by a Zeiss Axioplan 2 microscope equipped with the objectives ×5, ×20, ×50 and ×100 in reflective light mode with bright field. The images were taken with a Nikon digital DXM1200F camera coupled to the microscope.

Fourier Transform Infrared spectroscopy (FTIR) was carried out with a 4300 Handheld FTIR spectrometer of Agilent Technologies. The measurements were obtained in Attenuated Total Reflection (FTIR-ATR) mode with a spectral range from 4000 to 650 cm$^{-1}$ and a spectral resolution of 4 cm$^{-1}$. Each spectrum was the product of 32 internal scans. In order to compare spectra, the horizontal baseline was corrected with the baseline subtraction in the range 3800-4000 cm$^{-1}$ and, then, spectra were normalized to the maximum peak with the software Spectragryph version 1.2.6.
The surface micro-roughness was measured with an optical roughness meter TRACEiT from Innowep GmbH. 3D topographical maps (5 × 5 mm area, and height in µm) were carried out with a resolution of 2.5 µm (in X, Y and Z axes). To compare the glass surfaces, three roughness maps (2 x 2 mm) were measured on each sample, they were flattened and the arithmetic average roughness, Ra [27], was calculated with the software Gwyddion version 2.32. The average of Ra was calculated for each sample.

The static contact angle (water drop-glass surface) was determined with a Drop Shape Analyzer Phoenix 300 Seo from Surface Electro Optics [28]. The diameter of the needle was 0.483 mm (interior) and 0.711 mm (exterior).

3. Results and Discussion

3.1. Surface pathologies

All glasses exposed to the environmental conditions of Cape Vilán (Spain) presented alteration pathologies, even after 3 months of exposure. The most common pathology was the presence of surface crystals. They were both deposited from environmental particles such as aerosols and dust, or formed by the alteration of the glass surface. The number of deposits located was related to the atmospheric condition during the days before the collection. If samples were collected after a drought period, deposits were spread on the surface; however, if the glasses were collected after a rainy period, some isolated deposits were observed on the glass surface because rainwater washed the surface.

The inland distance to the shore affected the deposits composition. Large cubic crystalline salts attributed to NaCl crystals were observed on the samples from the Corrosion station 1, the nearest to the shore (Fig. 1 a). In general, surface deposits favored the retention of water content on the surface, which was observed by optical microscopy as dark or iridescent haloes around the deposits (Fig. 1 b). This water can stay on the glass surface for long periods of time favoring the local alteration of the glass in the long term.
Some samples presented also mechanical impacts on their surface due to the collision of soil particles transported by the wind at high speed [29, 30]. The mechanical impact created conchoidal fractures in the impact point (Fig. 1 c), whose dimensions depended on size, form, composition, hardness, impact angle and velocity of the soil particles [30].

Regarding the chemical alteration of the glass, both alteration layers and pits were observed on the surface as result of their exposure to the coastal atmosphere. Alteration layers were observed in soda-lime silicate and mixed-alkali silicate glasses exposed during 9 and 12 months in the Corrosion stations 1 and 2. This alteration layer was observed by optical microscopy as a whitish layer (Fig. 2 a). This layer is very fragile and uses to appear fissured or with scratching marks due to the sample manipulation. Potash-lime silicate glasses showed interference colors, characteristic of the initial stage of the alteration layer, after just 3 months in the three corrosion stations (Fig. 2 b). After 6 and 9 months, the glasses exposed in the Corrosion station 1 presented also whitish areas, which were related to the separation of the most external alteration layers (Fig. 2 c).

Pits in an initial state of alteration were also detected on the glasses from the Corrosion station 3 after 9 and 12 months of exposure, independently of the chemical composition of the sample. The majority of the pits were formed by a rounded deposit surrounded by an alteration layer, with iridescent or whitish color (Fig. 3 a). The iridescence observed around the
pits is an optical phenomenon related to the formation of the alteration layer; however, the whitish color is due to the separation of the thin alteration layer. The latter is very fragile and it appears frequently fractured (Figs. 3 b and c).

Fig. 3. OM images of the pits in the sample a) NCV3 after 12 months, b) MCV3 after 12 months, c) KCV3 after 9 months.

Some samples presented also insect eggs on their surface (Fig. 4). The eggs were white round spheres, but under the microscope light they look dark, and some of them appeared hatched. It is interesting that neither pits nor other pathologies related to water retention were observed under them; either because the eggs were placed on the glasses a short time before the glass collection or because the eggshells were hydrophobic [31].

Fig. 4. OM image of insect eggs on the KCV3 sample.

3.2. Surface roughness and wetting properties

The surface of the blown glass was not as flat as industrial or polished glass because it was made by a traditional elaboration process (Section 2.1.). These irregularities, together with surface deposits, influenced the surface topography.

Most of the samples presented a similar roughness (Ra = 4-8 µm) (Table 4, Fig. 5 a), which was attributed to the glass surface topography instead of the corrosion process. The glasses exposed during three months in the Corrosion station 3 were the roughest ones (Ra ≈ 11 µm) due to the presence of large deposits on them (Fig. 5 b); and some glasses presented Ra <2 (Table 4) because they showed a homogeneous clean surface (Fig. 5 c), which could be related to a punctual cleaning of the surface (environmental, manual) or to the good wetting between
water and glass surface (Table 5) [21]. The scratching marks due to manipulation and/or losses of alteration layers (Fig. 2 a and 3 c) have a depth of few micrometers and they were not observed in the roughness maps.

The contact angle between water drops and glass surface before the exposure is similar in the three glasses independently from the glass composition (Table 5), because the three glasses were fire polished (Section 2.1). During their manufacture, volatile ions, such as alkaline ions, were slightly evaporated from the surface creating a nanometric layer enriched in SiO₂ similar in the three glasses [32, 33]. This layer also stabilizes the surface because alkaline ions, the most vulnerable ones, were protected by the silica lattice. After the exposure, the contact angle between water drops and glass surface increased their value possibly due to the exchange of H⁺-ions from glass surface and Na⁺-ions from marine aerosols (Table 5). In general, the presence of deposits on the surface did not influence the contact angle because they are soluble in the water drop.

Table 4. Average roughness (OSR) of the glass surface (2 x 2 mm) in function of the time and the distance to the shore. * Analyzed area: 1 mm².

<table>
<thead>
<tr>
<th>Samples</th>
<th>Time (months)</th>
<th>Station 1 (µm)</th>
<th>Station 2 (µm)</th>
<th>Station 3 (µm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>NCV</td>
<td>0</td>
<td>5.3 ± 0.3</td>
<td>5.4 ± 1.4</td>
<td>7.6 ± 0.5</td>
</tr>
<tr>
<td></td>
<td>3</td>
<td>5.3 ± 0.4</td>
<td>1.6 ± 0.1</td>
<td>11.4 ± 0.4</td>
</tr>
<tr>
<td></td>
<td>6</td>
<td>5.5 ± 2.3</td>
<td>6.5 ± 1.5</td>
<td>3.5 ± 0.6</td>
</tr>
<tr>
<td></td>
<td>9</td>
<td>7.0 ± 1.6</td>
<td>3.8 ± 0.9</td>
<td>7.6 ± 2.6</td>
</tr>
<tr>
<td></td>
<td>12</td>
<td>-</td>
<td>7.4 ± 1.9*</td>
<td>1.1 ± 0.1</td>
</tr>
<tr>
<td>KCV</td>
<td>0</td>
<td>3.4 ± 0.3</td>
<td>4.7 ± 0.9</td>
<td>6.7 ± 0.6</td>
</tr>
<tr>
<td></td>
<td>3</td>
<td>2.1 ± 0.7</td>
<td>5.6 ± 1.6</td>
<td>11.1 ± 2.6</td>
</tr>
<tr>
<td></td>
<td>6</td>
<td>3.9 ± 1.2</td>
<td>3.6 ± 1.5</td>
<td>4.1 ± 0.8</td>
</tr>
<tr>
<td></td>
<td>9</td>
<td>2.3 ± 0.4</td>
<td>3.4 ± 0.8</td>
<td>4.2 ± 1.0</td>
</tr>
<tr>
<td></td>
<td>12</td>
<td>-</td>
<td>6.0 ± 2.9*</td>
<td>7.3 ± 0.6</td>
</tr>
<tr>
<td>MCV</td>
<td>0</td>
<td>5.9 ± 0.9</td>
<td>5.4 ± 1.3</td>
<td>5.9 ± 1.1</td>
</tr>
<tr>
<td></td>
<td>3</td>
<td>3.4 ± 1.3</td>
<td>7.9 ± 0.7</td>
<td>11.1 ± 1.6</td>
</tr>
<tr>
<td></td>
<td>6</td>
<td>4.0 ± 2.0</td>
<td>6.8 ± 2.6</td>
<td>4.7 ± 3.7</td>
</tr>
<tr>
<td></td>
<td>9</td>
<td>4.3 ± 1.3</td>
<td>6.0 ± 1.9</td>
<td>4.3 ± 0.8</td>
</tr>
<tr>
<td></td>
<td>12</td>
<td>-</td>
<td>6.5 ± 2.5*</td>
<td>7.7 ± 1.3</td>
</tr>
</tbody>
</table>

- Not available
Fig. 5. Roughness maps from the surface of the samples a) MCV1, b) KCV3, c) NCV3, after 3 months.

Table 5. Average of contact angle values (in degrees) and standard deviation measured for drops of distilled water on the glass surfaces before and after the exposure.

<table>
<thead>
<tr>
<th>Contact angle (°)</th>
<th>Before exposure</th>
<th>Station 1 (9 months)</th>
<th>Station 3 (9 months)</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>With deposits</td>
<td>Without deposits</td>
<td>With deposits</td>
</tr>
<tr>
<td>NCV</td>
<td>26.8</td>
<td>59.9</td>
<td>60.2</td>
</tr>
<tr>
<td>KCV</td>
<td>29.6</td>
<td>42.7</td>
<td>48.5</td>
</tr>
<tr>
<td>MCV</td>
<td>29.6</td>
<td>51.3</td>
<td>68.6</td>
</tr>
</tbody>
</table>

3.3. Chemical alteration

The evolution of the surface alteration in the coastal atmosphere was characterized by FTIR-ATR. The spectra of soda-lime silicate glasses were similar in the three corrosion stations during the 12 months of exposure (Figs. 6 a, b and c). The modification of the infrared bands was almost negligible; however, the sample NCV1 after 9 months presented a low increase in the band of the asymmetric stretching of the Si-O-Si (Fig. 6 a). This increase can be related to the formation of the silica gel layer on the surface [34]. The band attributed to the bending of the molecular water adsorbed on the surface ($\delta_{1210}$) was also detected in some samples (Fig. 6 a).
Fig. 6. FTIR-ATR spectra of soda-lime silicate glasses from the a) Corrosion station 1, b) Corrosion station 2, and c) Corrosion station 3.

Mixed-alkali silicate glasses showed a similar behavior than soda-lime glasses (Figs. 7 a, b and c). Glass samples were almost stable in the Corrosion stations 2 and 3; however, the samples exposed in the Corrosion station 1, the nearest to the shore, presented the progressive formation of the alteration layer on the glass surface (Fig. 7 a). The band attributed to the asymmetrical stretching of the siloxane bonds ($\nu_{as}$ (Si-O-Si) ~1010 cm$^{-1}$) increased its height after the glass being exposed 6 and 9 months as result to the formation of the alteration layer (Fig. 7 a). It was observed also the bending band of the molecular water adsorbed on the glass surface (Figs. 7 a, b and c).
Contrary to soda-lime and mixed-alkali silicate glasses, potash-lime silicate glass was the most unstable glass in the Cabo Vilano conditions because all the samples, even after 3 months of exposure, were altered (Figs. 8 a, b and c). It was observed the progressive increase in the IR bands related to the asymmetric stretching of the siloxane bond (~ 1010 and ~ 1200 cm\(^{-1}\)) and a decrease in the stretching band of the silanol groups (~ 870 cm\(^{-1}\)) (Figs. 8 a, b and c). Hydration water was also detected in all the samples (Figs. 8 a, b and c).
Some spectra presented the bands attributed to the carbonate groups, mainly in the samples from the Corrosion station 3 (Fig. 8 c). The carbonates could be environmental aerosols or formed as result of the reaction between the leached ions from the glass with the environmental CO$_2$. The most probable compound is CaCO$_3$, which is the most stable deposit and with less solubility according to the evolution of the crystalline phases on glass alteration [21].
4. Discussion

4.1. Influence of the environment

The corrosion stations were located in the Cape Vilán wind farm, in the west coast of Galicia (Spain), in the “Costa da Morte” (“Coast of Death”). This area presents a temperate climate with dry and temperate summer (Csb) according to the Köppen–Geiger climate classification [35, 36] and a West European Maritime Climate according to the Climatic regions of the Iberian Peninsula [37]. This region is characterized by a warm temperate climate with dry summers (T (temperature)$_{\text{max}}$ < 22 °C, T$_{\text{mean}}$ > 10 °C during more than 4 months per year, P (precipitations)$_{\text{min\_summer}}$ < P$_{\text{min\_winter}}$, P$_{\text{max\_winter}}$ > 3 P$_{\text{min\_summer}}$, and P$_{\text{min\_summer}}$ < 40 mm) [38].

Marine atmosphere predominates at a maximum distance of ~20 km from the coast [22, 23], as in the Cape Vilán wind farm (< 1.5 km). The prevailing wind of the area is the northeast wind which brings marine aerosols from the Arneliña inlet, in the Atlantic Ocean. It was observed that higher wind speed, faster deposition rate of atmospheric chloride ions on the corrosion stations (Table 2). The coarse particles from marine aerosols are mainly composed by sea salt, predominantly sodium and chlorine ions [39, 40]. The main route of Cl$^-$- and Na$^+$-ions deposition is by wet deposition [23] and they can form NaCl crystals on the glass surface (Fig. 1a).

During the exposure period, the temperature was around 15 °C and the relative humidity was very high (~85 %) (Table 3). Under these environmental conditions, glass surfaces were covered by few monolayers of liquid water [41]. This water can induce the hydrolytic attack (Reaction 1) and the dealkalinization of the glass surface (Reaction 2). The silanol bonds formed as result of these reactions can polymerize between them to form a porous layer enriched in SiO$_2$ (Reaction 1), which is commonly called silica gel layer [42].

\[ \equiv\text{Si-O-Si} + \text{H}_2\text{O} \rightleftharpoons 2 \equiv\text{Si-O-H} \]  (1)

\[ \equiv\text{Si-O-M} + \text{H}_3\text{O}^+ \rightarrow \equiv\text{Si-O-H} + \text{M}^+ + \text{H}_2\text{O} \]  (2)

This silica gel layer presents a more open structure than bulk glass and permits the diffusion of water molecules and other chemical species through it [43, 44]. Both hydrolysis and dealkalinization can occur simultaneously in the bulk glass below the silica gel layer, continuing the dealkalinization process and increasing the thickness of the alteration layer.

The dealkalinization reaction releases alkaline and hydroxyl ions to the medium (Reaction 2). If the concentration of these basic ions is high (pH > 9), they can break the oxygen bridge bonds...
inducing the irreversible disruption of glass network (Reaction 3). The accumulation of negative charges in the alteration layer also increases the repulsion forces which tend to open the glass network, accelerating the alteration process [45, 46]. Rainwater can wash the leached ions (alkaline, alkaline-earth and hydroxyl ions) from the glass surface. The removal of hydroxyl ions favors the maintenance of a neutral pH in the surface glass, which prevents the basic corrosion mechanism (Reaction 3).

$$≡\text{Si-O-Si}≡ + \text{OH}^- \rightarrow ≡\text{Si-OH} + ≡\text{Si-O}^-$$  (3)

Precipitations during the exposure period were recurrent, mainly during the winter (Table 2). Around 13 days per month rained in Cape Vilán, 8 days/month rained moderately, and they presented an average of 61.6 mm/month (Table 3). This intermittent rain produced a degradation mechanism alternating between active and static state. During the rainy period, the glass surface was washed and the basic ions and deposits were removed [5, 11]. However, when the rain stopped, a thick layer of water stayed on the glass surface for a long time due to the high environmental humidity. This water can cause a hydrolytic attack favoring the formation of the alteration layer (Reaction 1 and 2). This water can also dissolve aerosols and atmospheric gases, which can react with the leached ions from the glass to form deposits when water evaporates [4]. The most common deposits in the coastal atmosphere, according to their Gibbs free energy of formation and solubility, are CaCO$_3$, Na$_2$CO$_3$ and NaCl [21]. These deposits can retain water on the glass which can induce punctual alterations (Fig. 1 b) [10, 47]. The number of deposits on the surface depended on the atmospheric condition before and during the collection. If rainwater washed the surface, fewer deposits were observed on it.

NaCl aerosols also induce the increase of the hygroscopicity on the glass surface. The hygroscopic behavior of single crystals of NaCl has a well-defined deliquescence point at ~76 % RH, but halite particles aggregated with other soluble phases decrease the deliquescent point at slightly lower than 74 % RH [48]. At higher humidity, such as in Cape Vilán (85 ± 4 %) (Table 3), these particles will deliquesce to form droplets. If several droplets were located in the glass, they could join together covering homogeneously the glass surface due to the high wettability of the glasses (Table 5).

4.2. Influence of the exposure period

The glasses exposed in Cape Vilán (Galicia, Spain) showed a progressive alteration related to the exposure period (Section 3.1 and 3.3) [3]. The intermittent rain in the corrosion stations induced successive cycles of wetting and drying (Table 3). After the rain, the water on the glass
surface induced the hydroxyllic attack and the dealkalization process (Reactions 1 and 2). The leached ions could also react with the atmospheric gases dissolved in the hydration water to form deposits. However, the following rainy episode would wash the glass surface removing the deposits and leached ions, and leaving new fresh water on the glass surface.

After the successive cycles, pathologies such as pits or hydration layers were formed on the glasses (Section 3.1 and 3.3). The damage produced by these pathologies is permanent even after being washed with the rain [3, 5, 7, 11]. This damage can also accelerate the degradation reaction because pits can retain water favoring the formation of alteration layers around them (Figs. 3 a, b and c) [47], and the silica gel layers also retain water and leached ions, which can induce the basic mechanism of corrosion (Section 4.1) [42, 49].

4.3. Influence of the glass composition

Glasses presented a different behavior to the exposure depending on their chemical composition. Potash-lime silicate glass was the most altered glass, instead of soda-lime silicate and mixed-alkali silicate glasses which were “almost” stable glasses (Figs. 9 a, b and c).

Fig. 9. OM images of the glass samples a) NCV3, b) MCV3, and c) KCV3, after 9 months of exposure.

Soda-lime silicate and mixed-alkali silicate glasses presented several deposits on the surface (Figs. 1, 9 a and 9 b). Their presence was related with both atmospheric aerosols (marine and soil particles transported by wind) and neocrystallizations formed between the leached ions from the glass and the atmospheric gases solubilized in the surface water (Section 4.1). The number of surface deposits depended on the environment more than on the chemical composition of the glass. As result of the exposure, these glasses showed a thin alteration layer in the most harmful environment and isolated pits in the samples located at more distance to the shore. The alteration in these glasses is mainly produced by surface soiling [2], deposits retain water on the glass surface inducing punctual attacks (isolated pits) [2]. Moreover, previous works [21, 50, 51] showed that soda-lime silicates present a slow and long hydration process in aqueous media, which could be even slower in atmospheric environments.
On the contrary, potash-lime silicate glasses formed surface alteration layers in all the samples, independently of their exposure location (Figs. 8 a, b and c). These alteration layers presented different thickness depending on the degree of alteration. This alteration layer is formed by the dealkalization of the glass network (Reaction 2) followed by a condensation reaction (Reaction 1) [1, 52]. For this reason, the increase of the intensity of the FTIR-ATR bands attributed to the bridging bonds of the silica (asymmetric stretching bands of SiO$_2$, $\nu_{as \ Si-O-Si}$) followed by the relative decrease of the band attributed to the non-bridging bonds (asymmetric stretching bands of SiO$_-$, $\nu_{as \ Si-O}$) pointed to the formation of a thick surface alteration layer [53]. Higher is the relation between the bands, thicker is the alteration layer. It was observed that the formation of a thick alteration layer was faster in the Corrosion Station 2, instead of the Corrosion Station 1 (Fig. 8 a and b). This is because the alteration layer in the samples exposed near to the coast (KCV1) could be lost due to environmental conditions. On the contrary, the formation of alteration layers on the samples exposed in the Corrosion Station 3 presented the slowest process.

Potash-lime silicate glasses are more susceptible to alteration because the ionic radius of K$^+$-ions (0.133 nm) is higher than Na$^+$-ions (0.097 nm) [54], which departure forms a more open structure in comparison with soda-lime silicate glasses. Regarding the chemical stability, the bond strength of groups [$\equiv$Si-O-K] is lower than groups [$\equiv$Si-O-Na] which favors the hydrolytic attack of the former bonds. Rainwater and environmental humidity are the most important factors for glass degradation in potash-lime silicate glasses and produce alteration layers [2, 3, 42]. The evolution of the alteration layer in the samples from the Corrosion stations 1 and 2 is similar and faster than the evolution of samples from the Corrosion station 3 (Figs. 8 a, b and c). This different alteration rate was not due to the meteorology but to the content of marine aerosols (Section 4.4). In mixed-alkali silicate glasses, the moderate content of sodium ions closed the structure around the potassium ions which increased its chemical stability.

The surface topography also affects to the glass corrosion because the irregularities inherent to the artisanal blown glass retained the water in some areas (Section 3.2.), which favored its faster alteration (Fig. 2 b).

4.4. Influence of the distance to the coast

The glasses exposed in Cape Vilán (Galicia, Spain) showed a different behavior in function of the distance from the corrosion station to the Atlantic coast. In general, the samples exposed in the Corrosion station 1 and 2, the nearest to the shore, presented alteration layers (Figs. 10 a and b); while those samples from the Corrosion station 3 showed isolated pits (Fig. 10 c).
The glasses presented the same chemical composition and they were exposed to the same atmospheric environment, so this different behavior is related to the atmospheric concentration of marine aerosols in each corrosion station (Table 2 and 3).

The coarse particles from marine aerosols are mainly composed by sea salt, predominantly sodium and chlorine ions [39, 40]. They can form NaCl crystals on the glass surface, as it was observed on the surface of the samples from the Corrosion station 1 (Fig. 1 a). NaCl aerosols deliquesce at environmental humidity to form droplets of saline solutions (Section 4.1). The samples exposed in areas with a higher deposition rate of atmospheric chloride ions, such as Corrosion station 1 and 2 (Table 2), could be “homogeneously” covered with a layer of saline solution favoring the alteration of the entire surface at the same time. On the contrary, the glasses in the Corrosion station 3, with a lower deposition rate of atmospheric chloride ions (Table 2), presented punctual attacks (pits). This latter pathology has been previously observed as result of accelerated tests in alteration chambers without the influence of marine aerosols [55] and they are the result of surface soiling because surface deposits retained water on the glass producing punctual attacks [10, 47, 55].

The influence of marine aerosols was also observed in the evolution of potash-lime silicate glass. The thickness of the alteration layer on the samples from the Corrosion station 3 increased slowly in comparison to the samples from the Corrosion station 1 and 2 (Figs. 8 a, b and c), showing that the high content of marine aerosols accelerated the alteration mechanism.

Previous studies [50, 51, 56] have shown that high content of Na\(^+\)-ions in an aqueous medium increase the alteration rate of silicate glasses because they can be interchanged with the H\(^+\)-ions from the silanols groups \(≡\text{Si-O-H}\). As consequence of the ionic exchange and the steric hindrance of the Na\(^+\)-ions, \(≡\text{Si-O-Na}\) groups were situated perpendicularly to the surface, favoring the opening of the structure and accelerating the corrosion mechanism [50]. The
results of this study are in accordance with the mechanism proposed because, for the same
time of exposure, those samples located near to the shore presented an advance state of
alteration in contrast to those ones located far (Section 3.3). The high content of marine
aerosols could be partially washed away by the rainwater; nevertheless, it is clear that they
can modify the surface during the dry period (Table 5).

5. Conclusions

Glass alteration depended on both the content of marine aerosols (distance to the coast) and
the chemical composition of the glasses. Those samples exposed in the corrosion stations
nearest to the shore presented alteration layers, while those ones located farthest from the
coast showed several isolated pits. The higher content of marine aerosols could increase the
hygroscopicity of the glass surface and, also, open the glass structure favoring the alteration.
While the samples exposed far from the shore showed the isolated pits related to the soiling.

Regarding the chemical composition of the glass, mixed-alkali silicate glass showed a similar
behavior to soda-lime silicate glass since sodium ions closed the glass network around the
potassium ions avoiding the hydrolytic attack. Nevertheless, potash-lime silicate glass was the
most altered glass because its open lattice favors the hydrolytic attack.

This study proves that the stained-glass windows or glass façades located in areas with high
concentration of marine aerosols, such as a few hundred meters from the coast or in areas
where the wind transports a high content of sea/ocean aerosols, are threatened because the
glasses in this environment can experiment a very fast alteration, even higher than in polluted
environments.

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